

Absorption Enhancement for Ultra-Thin Solar Fuel Devices with Plasmonic Gratings

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Abstract

We present a concept for an ultra-thin solar fuel device with a nanostructured back contact. Using rigorous simulations we show that the nanostructuring significantly increases the absorption in the semiconductor, CuBi_2O_4 in this case, by 47% (5.2 mAc m^{-2}) through the excitation

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of plasmonic modes. We are able to attribute the resonances in the device to metal-insulator-metal plasmons coupled to either localised surface plasmon resonances or surface plasmon polaritons.

Rounding applied to the metallic corners leads to a blueshift in the resonance wavelength while maintaining absorption enhancement, thus supporting the possibility for a successful realization of the device. For a 2D array, the tolerance of the polarization-dependent absorption enhancement is investigated and compared to a planar structure. The device maintains an absorption enhancement up to incident angles of 75° .

The study highlights the high potential for plasmonics in ultra-thin opto-electronic devices such as in solar fuel generation.

1 Introduction

The conversion of sunlight to storable fuel is a challenge of paramount importance to modern society. Photoelectrochemical devices, consisting of semiconductor photoelectrodes immersed in aqueous solution, are a particularly interesting way to achieve this solar-to-fuel conversion. In recent years many different device designs based on various materials have seen intensive research [1, 2]. Among these, metal oxides are particularly interesting since they possess general aqueous stability and are relatively inexpensive [3, 4]. However, they share a common drawback, that of the discrepancy between carrier transport and light absorption. Due to the poor transport properties, the carrier diffusion length in oxides is typically less than 100 nm [5, 6, 7, 8, 9, 10, 11]. On the other hand, they normally have an indirect band gap; relatively thick films (>500 nm) are needed to absorb enough light. This mismatch often severely limits the performance of a metal oxide photoelectrode.

In the present work we focus on the metal-oxide semiconductor CuBi_2O_4 , which is an emerging p-type semiconductor for solar fuel applications. It has a band gap of around 1.8 eV, which is ideal for a top absorber in a tandem configuration [12, 13, 14, 10]. It also has a suitable band position; the conduction and valence band edges straddle both water reduction and oxidation potentials. As a result, the photocurrent onset potential for CuBi_2O_4 has been reported to be ~ 1 V vs reversible hydrogen electrode (RHE), which is beneficial for a tandem configuration [12, 13, 14]. Previous reports of CuBi_2O_4 synthesis deposition have shown highly porous and irregular surface structures [10, 13, 15]. Alternative synthesis methods such as spray-pyrolysis have been used to make dense, homogeneous CuBi_2O_4 thin films [16] and new methods such as pulsed laser deposition (PLD) could be used to obtain a highly uniform CuBi_2O_4 ultra-thin film. PLD has been demonstrated as a viable option for other metal oxide photoelectrode materials including resonant light trapping Ti-doped $\alpha\text{-Fe}_2\text{O}_3$ [17].

However, the material is limited by the optical absorption vs. carrier transport mismatch as was previously mentioned. The diffusion length is in the range of ~ 50 nm, while a thickness of more than 500 nm is needed to absorb 90% of the incident light [10]. This is especially true for the longer wavelengths (>

450 nm), where the quantum efficiency has been reported to be very low.

In order to combat this challenge, we propose to reduce the semiconductor thickness to 100 nm which should ensure efficient carrier collection by shortening the length photo-generated carriers have to travel. Simultaneously we use light management to obtain a sufficient absorption of incident sunlight.

Various light management strategies applied to the field of solar fuels have been presented in the literature. Photonic crystal structuring of the active layer has been used to enhance absorption [18]. Other approaches use dielectric particles and gratings to localize light inside the active layer [19, 20]. Structuring the active material into nanorods can also increase absorption while maintaining short carrier diffusion lengths [21]. A further proposed light management strategy is that of plasmonics [22, 23]. Plasmonic metallic nanoparticles act as optical antennas, allowing light to be concentrated in the vicinity of the semiconductor material, thereby enhancing absorption [24, 25]. Furthermore metallic particles themselves may have beneficial catalytic properties [14, 26]. Despite this, certain challenges are present for particles, such as quenching the photocatalysis process through recombination [27, 28].

In this paper we circumvent the challenges of using particles by considering a 100 nm thick CuBi_2O_4 layer on a grating consisting of laterally alternating Ag and SiO_2 on top of an Ag layer. Ag and SiO_2 are less positive than CuBi_2O_4 vs. RHE. In order to circumvent a Schottky at the rear interface, an additional back contact layer or heavy doping layer of the CuBi_2O_4 may be necessary [29]. Since we focus on the optical device design, such a layer is not taken into account in the current work. The unit cell of this periodic structure is shown in the inset of figure 1(b). We will refer to the SiO_2 region as a nanoslot since it forms a slot in the Ag. Plasmonic gratings have been realized for multiple applications including photovoltaic absorption enhancement [30] and biosensing [31]. For the current application, the nanoslot grating serves as the metallic back contact to transport photo-generated holes to the anode side as well as to enable better light management. A similar structure has been applied to infrared absorption enhancement in photovoltaics [32]. Through careful device design, we are able to shift the operational frequency to visible wavelengths.

2 Results and Discussion

2.1 1D Grating

Figure 1 shows the absorption and losses for the ultra-thin photoelectrode with a planar Ag back reflector with no nanoslots (a) and a nanostructured back reflector (b). The inset shows a schematic of each structure. Both of the devices have a 100 nm thick CuBi_2O_4 layer.

The catalytic reaction, in this case water reduction, occurs at the interface between H_2O and the inorganic semiconductor photocathode, CuBi_2O_4 . Light is incident through the H_2O .

The nanostructured device has a unit cell width (pitch) of 112 nm, the

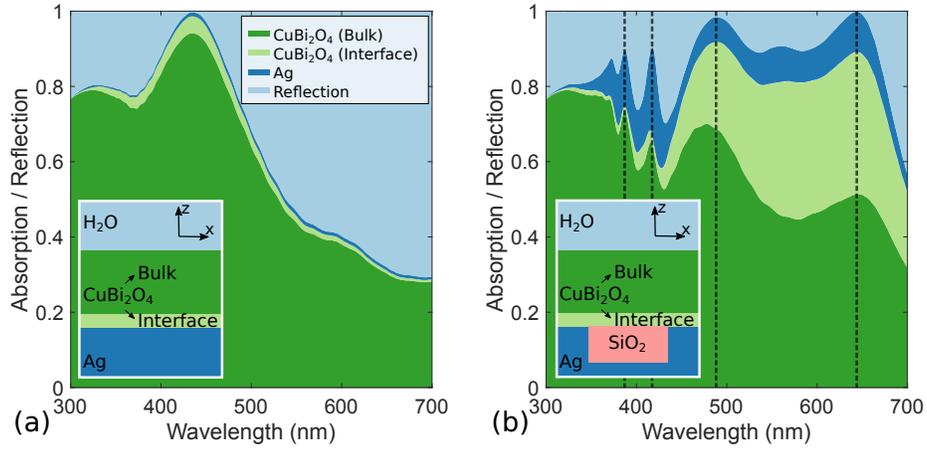


Figure 1: The absorption in the 100 nm thick CuBi₂O₄ layer (divided into bulk [first 95 nm] and interface [last 5 nm] contributions), and the losses in Ag and Reflection for a solar fuel device with a flat Ag back contact (a) and a nanostructured Ag back contact (b). The dotted lines indicate resonance wavelengths shown in figure 2. Insets show schematic drawings of the periodic unit cell for each case.

SiO₂ slot is 70 nm wide and 60 nm deep, it is infinitely extended in the plane perpendicular to the page, therefore defining a 1D grating. These values were obtained from an optimization (figure ??). We split the semiconductor into two regions. Firstly, the upper 95 nm of the semiconductor material (dark green), which is in contact with the water. Secondly, the final 5 nm of the semiconductor material (light green), which is in contact with the Ag back contact. These regions will be referred to as the 'bulk' and 'interface' regions respectively. Analogously, the absorption in each region will be referred to as 'bulk' and 'interface' absorption, respectively.

For the first case of the simple back reflector (no SiO₂ nanoslots), the interface absorption of the semiconductor is very small compared to the bulk. This is due to the exponential damping of light while traversing the first 95 nm of the material (Lambert-Beer law) and also due to the volume of the region in question being much smaller than that of the bulk region.

In order to quantify this absorption for solar fuel applications, we calculate the photocurrent density (J_{abs}) from the absorption curve and the solar spectrum. This assumes that all absorbed photons can contribute to water reduction. This assumption should be interpreted as the upper limit on photocurrent density for the proposed photoelectrode architecture. When modeling practical devices, the various loss mechanisms (e.g. recombination) have to be taken into account through the absorbed photon-to-current efficiency (APCE). We note that there have been reports for nanostructured metal-oxides with APCE values close to 100% [21]. The proposed structure is able to reach a theoretical photocurrent density of 11.1 mAcm⁻² which is 54% of the maximum achievable short circuit current density for a material with a band-gap of 1.8 eV (20.5 mAcm⁻²).

Due to the low losses provided by Ag, the absorption in the back reflector is minimal, meaning that the main loss mechanism is reflection. This loss mechanism is eliminated entirely at the wavelength of 440 nm due to the presence of a Fabry-Perot resonance which eliminates the reflection. These kinds of resonances are clearly beneficial to absorption, however they cannot provide an arbitrarily broad absorption enhancement, since the only free parameter for tuning such a resonance is the film thickness.

Absorption is seemingly still present at wavelengths up to 700 nm. The complex refractive index used for CuBi₂O₄ also contains parasitic absorption at and below the band gap (figure ??). However analysis of the absorption coefficient has shown that the band gap for CuBi₂O₄ lies around 700 nm. Therefore we use this wavelength as the cutoff for absorption contributing to the photocurrent density.

Figure 1(b) shows the reflection and absorption for the nanostructured back contact (Ag with SiO₂ nanoslots). The polarization is oriented in the $x - z$ plane so that the plasmonic effects can be studied. In this case the total absorption is over 80% for most of the visible spectrum, particularly between 500 and 650 nm where the solar spectrum has peak intensity. The overall absorption increase leads to a maximum short circuit current density of 16.3 mAcm⁻² (80% of the maximum achievable). This large increase in absorption can be attributed to the

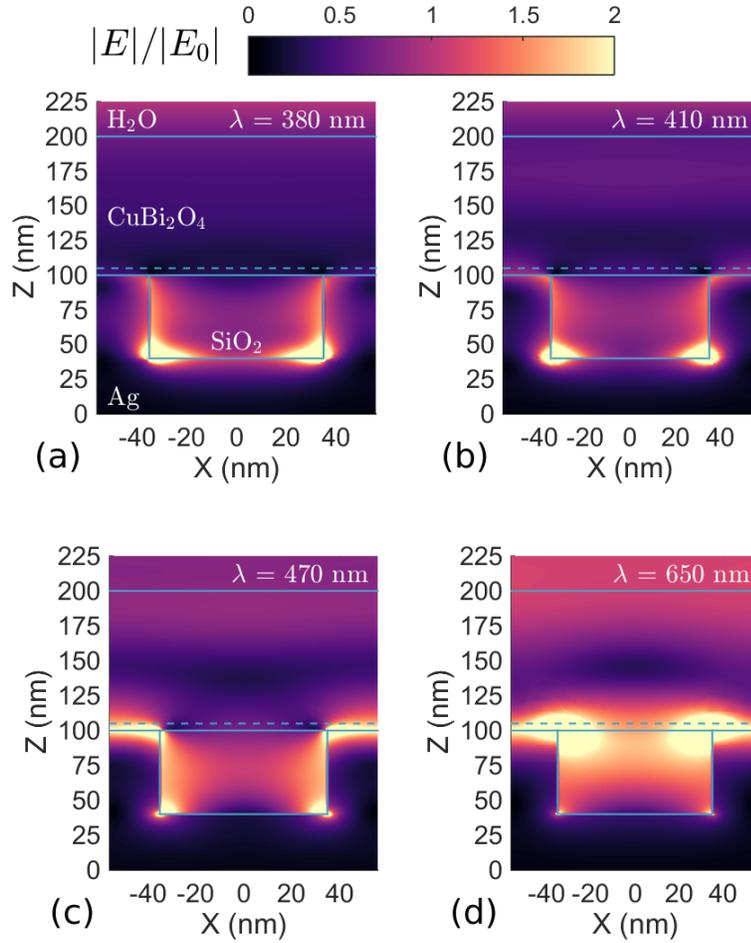


Figure 2: The electric field in the nanostructured solar fuel device for four different wavelengths taken from the peaks in figure 1. Wavelengths are 380, 410, 470, 650 nm for parts (a-d), respectively. Light is incident normally from above and polarized in the x-z plane.

excitation of resonant modes that exist at the Ag / CuBi₂O₄ interface, due to the large contribution of the interface absorption (light green). In this case the bulk absorption contributes 11.1 mAcm⁻² to J_{abs} while the interface absorption contributes 5.2 mAcm⁻² to J_{abs} . In addition the parasitic absorption in the Ag back contact also increases due to increased interaction with the Ag arising from the resonant modes.

In order to investigate the mechanism of the absorption enhancement further, we show in figure 2 the electric near field strength of the nanostructured back contact for the peak wavelengths shown in figure 1(b), namely 380, 410, 470 and 650 nm. The presence of a resonant mode is clearly shown in each case with strong localization of the electric field.

All four of the modes can be associated with metal-insulator-metal (MIM) plasmon resonances of the slot [33, 34]. This can be seen in the variations of the resonance wavelength with respect to the slot length which varies with a wavelength close to the analytical MIM mode wavelength (figures ?? and ??).

The four modes can be grouped into two categories. At the shorter wavelengths of 380 and 410 nm (figures 2(a) and (b)), the resonance is mainly confined to a localized mode at the bottom corners of the slot. In contrast, the resonances at the longer wavelengths of 470 and 650 nm are mainly located at the upper corners of the slot and at the Ag / CuBi₂O₄ interface.

An analysis of the variation of these modes with the device pitch (keeping the slot size constant) reveals that at wavelengths between 450 and 700 nm, the field localized at the upper corners acts as a source of surface plasmon polariton (SPP) modes (figures ?? and ??). Since the SPP modes have significant field strength in the CuBi₂O₄ they are able to increase the absorption there. In contrast the shorter wavelength modes remain localized in the bottom of the slot and therefore contribute mainly to losses in the Ag.

The maximum absorption enhancement is observed when an antinode of the MIM mode resonance is at the slot opening and when the SPP mode which is excited constructively interferes with itself. This holds true for the device at a wavelength of 650 nm. Slot length variations (figure ??) and pitch variations (figure ??) which affect the MIM and SPP resonances, respectively, show a peak for 60 nm slot length and 112 nm pitch. For the peak at 470 nm wavelength, the MIM mode is at maximum enhancement while the SPP is slightly off resonance. Conversely, at 575 nm wavelength, the MIM mode is off resonance while the SPP mode shows maximum constructive interference causing the increase in interface absorption visible at this wavelength.

Due to the coupling to SPP interface modes, the total absorber layer thickness can, in principle, be reduced while maintaining a high absorption. However, the total absorption obtained will still drop with decreasing layer thickness, as the short wavelength light (< 350 nm) still needs to be absorbed conventionally since the inter-band transition losses in Ag prevent any beneficial resonances from forming at these wavelengths. Therefore, although we have presented absorption curves for the device with a 100 nm thick absorbing layer, we stress that the presented nanostructured device could also conceivably provide absorption enhancements for much thinner layers.

2.2 Effect of Corner Rounding

A further consideration for the implementation of these nanostructures in realistic devices is the ability to fabricate precise geometries. As the near field pictures from figure 2 show, there is a strong localization of electric field in the vicinity of the sharp edges of the Ag grating. Such perfectly sharp edges are difficult to fabricate, therefore a certain amount of rounding on the edges should be expected.

In figure 3 we show the absorption in the 100 nm of CuBi_2O_4 for the nanostructured 1D grating for three cases of corner rounding radius (R_c) at both the upper and lower grating corners. The definition of R_c is shown in the inset of figure 3. The values of R_c presented are 0 (same absorption curve as shown in figure 1(b)), 2 nm and 10 nm. It can be seen that even for $R_c = 2$ nm, the resonances are blueshifted and this becomes more pronounced with increasing corner rounding.

The blueshift seen is a combination of multiple factors. The MIM mode resonance wavelength tends to decrease with decreasing slot width. This may be more relevant to the shorter wavelength resonances since they are localised to the bottom of the slot. Furthermore, due to the inhomogeneous width of the slot, the length of slot which has the necessary width for the supporting the MIM resonance will be effectively shorter. Since the MIM mode provides an absorption enhancement when an antinode is present at the slot opening, the wavelength of MIM mode necessary for this may be shifted to shorter wavelengths. Finally, as the corner rounding increases, more of the SPP resonance will be located inside the slot which has a lower refractive index than CuBi_2O_4 . This lower refractive index will tend to redshift the SPP resonance. Due to competing factors, the effect of corner rounding on the exact resonance position is difficult to predict, necessitating further numerical study for an optimum to be found.

Despite the blueshifting the core resonant modes are all still present. We can conclude that the resonant absorption enhancement is not reliant on an unphysical singularity at the corners. This is important for the physical realization of the device.

The maximum photocurrent density obtained for the device with no corner rounding was previously shown to be 16.3 mAcm^{-2} . When a corner rounding of 2 nm is imposed, the photocurrent density lowers slightly to 15.8 mAcm^{-2} . As the corner rounding is increased to 10 nm, the photocurrent density further decreases to 14.6 mAcm^{-2} . All of these values show a significant improvement over the planar value of photocurrent density of 11.1 mAcm^{-2} . It should be further noted that the geometry can be reoptimized with corner rounding. If the corner rounding is 10 nm then a new optimum absorption enhancement can be found for a pitch of 200 nm, slot width of 100 nm and slot length of 50 nm. In this case the photocurrent density is increased to 16.5 mAcm^{-2} . The absorption profile for the optimized structure with corner rounding is shown in figure 3.

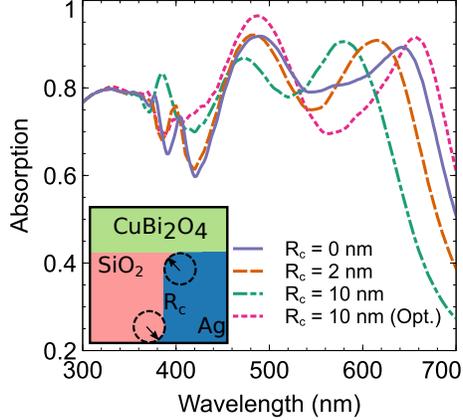


Figure 3: The absorption inside the absorbing semiconductor (CuBi_2O_4) as a function of the wavelength for three different values of corner rounding R_c . The first three structures have the same geometrical parameters as in figure 1(b), while the fourth has the geometrical parameters optimized for the corner rounding. Inset shows the definition of R_c . All other aspects of the geometry are the same as in figure 1(b). Light is incident normally from above and polarized in the x - z plane.

2.3 2D Grating

For integration into a solar fuel device, the proposed nanostructured grating has to continue to provide a strong enhancement in the presence of unpolarized light at different angles of incidence. In order to enhance the unpolarized response, the 1D grating can be extended to a 2D grating while keeping the same dimensions as the 1D grating. The resonance conditions found previously can be maintained for either a Ag grating with SiO_2 nanoslots (1), or a SiO_2 matrix containing Ag cubic nanoparticles connected to a Ag back contact (2). A schematic of configuration (2) is shown in figure ???. Configuration (2) was found to be more optically beneficial than configuration (1) and was therefore chosen for the results presented in this section. For reasons of computational efficiency, corner rounding has not been used for the 2D grating.

The periodic unit cell of the 2D grating has a 3 fold mirror symmetry. The mirror planes lie along the x axis, the y axis and at 45° to the x and y axes. Therefore it is sufficient to use azimuthal angles $0^\circ < \phi < 45^\circ$ to obtain the azimuthally averaged response. We chose 5 azimuthal angles equally spaced between 0° and 45° . The difference in absorption between averaging over 3 and 5 angles was smaller than 10^{-3} , therefore it was concluded that 5 angles are sufficient to obtain an averaged response. For each polar angle (θ in figure ??) of incidence > 0 we can define two orthogonal polarizations: p polarization where the electric field orientation lies in the scattering plane and s polarization

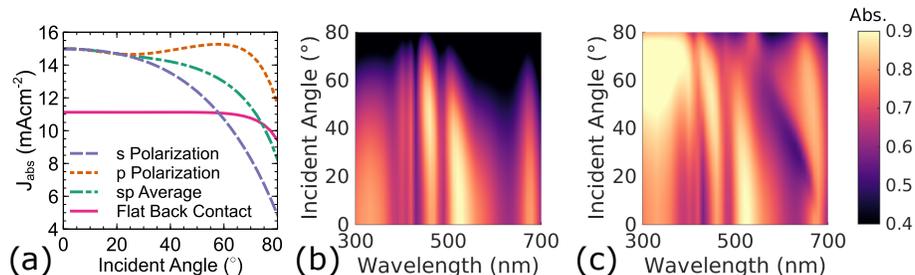


Figure 4: The absorbed photocurrent density as a function of the incident polar angle θ (a). The absorption in the semiconductor (CuBi_2O_4) for a 2D grating for polarization s (b) and p (c) as a function of both wavelength and polar incidence angle.

where the electric field orientation is perpendicular to the scattering plane, as shown in figure ??.

Figure 4(a) shows how the maximum photocurrent density J_{abs} for the structure depends on the incident polar angle. For oblique incidence p polarization provides a higher current density than for s polarization. Taking the unpolarized response into account, the grating outperforms a planar stack for angles up to 75° which is highly beneficial to solar fuel applications.

Figure 4(b-c) shows the wavelength resolved polar angular response for s and p polarization. The 2D grating, even at normal incidence, does not show the same behavior as the 1D grating. This is to be expected as they are physically different systems, however the main resonances present in figure 1(b) are also present in figure 4, namely a strong absorption enhancement between 500 and 600 nm and a slightly weaker enhancement between 600 and 700 nm. In general the absorption for the case of p polarization is higher as we move towards higher angles. This is partly due to reflection at the initial $\text{H}_2\text{O}/\text{CuBi}_2\text{O}_4$ interface being lower for p polarization.

The dependence of the plasmonic resonances on incident angle differs for the p and s polarizations. In figure 4(b) we see that for s polarization, the resonance wavelengths remain constant with increasing angle. The resonance positions do not change due to the electric field remaining normal to the vertical sides of the Ag cuboids for all incident angles. For the case of p polarization shown in figure 4(c), the resonances broaden and increase for higher angles. This is due to electric field component no longer being purely normal to the vertical sides of the Ag cuboids, thereby changing the resonance condition for excitation of MIM modes. Broader resonances will be beneficial to the broadband functioning of the device.

3 Conclusion

We have presented a nanostructured back contact for use in ultra-thin solar fuel devices. The nanostructuring was shown to significantly increase the absorption in the absorbing semiconductor, CuBi_2O_4 in this case, by 47% (5.2 mAcm^{-2}) through the excitation of plasmonic modes.

By varying the length of the SiO_2 nanoslot and the device pitch the resonances could be classified as MIM modes which remain isolated in the nanoslot or couple to SPP modes dependent on the wavelength.

By simulating the effect of corner rounding, it could be confirmed that the presented results do not rely on unphysical singularities at material interfaces. This means that translation of the simulated results to an experimental reality is promising. Moreover, we demonstrated that the detrimental effect of corner rounding, which is unavoidable in practical systems, can be fully compensated by adjusting the pitch and slot dimensions of the SiO_2 .

For a 2D grating based on the optimal 1D nanoslot grating, the angular tolerance of the absorption enhancement was investigated for two different polarizations. A greater angular tolerance was shown for p polarization, with the absorption even increasing at higher angles. The unpolarized response was shown to outperform planar layers up to angles of 75° .

These results provide a clear pathway to overcome the mismatch between the optical absorption and carrier diffusion length, which is present in many semiconducting photoelectrodes. In the end, it is expected that higher solar-to-fuel conversion efficiency with metal oxide photoelectrodes, especially the promising photocathode material CuBi_2O_4 , is to be achieved with our proposed architecture.

4 Methods

All simulations were done using the commercial software JCMsuite [35], a finite element solver for Maxwell's equations. All simulations modelled the upper and lower half spaces with perfectly matched layers while periodic boundary conditions were used in the $x - y$ plane. A plane wave source was incident from the upper half space. For H_2O and SiO_2 a wavelength independent refractive index of 1.33 and 1.5 was used, respectively. For Ag the data of Johnson and Christy were used [36]. The complex refractive index of CuBi_2O_4 was obtained via spectroscopic ellipsometry of a single crystal using an M-2000D ellipsometer (193-1000 nm, J.A. Woollam Co., Inc.). Details of crystal synthesis are given in the supporting information. By using a finite element degree of 3 and mesh side length constraint smaller or equal to one tenth of the wavelength in the material (for dielectrics) or a constant value of 5 nm (for metal) we were able to ensure an accuracy of greater than 10^{-3} for the absorption and reflection.

5 ASSOCIATED CONTENT

Supporting Information Available: The complex refractive index of CuBi_2O_4 used in all simulations. Description of the procedure for obtaining absorption and photocurrent density. The effects of length and pitch variations of the nanoslot. Optimization of the nanoslot with respect to slot width, slot length and pitch ratio. The geometry of the 2D nanoslot array and the definition of the scattering angles.

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