

# Annealing-induced Intermixing and Passivation of the Front Contact in Cu(In,Ga)Se<sub>2</sub> Devices – A Spectroscopic View on CdS and GaO<sub>x</sub>

Donald Valenta,<sup>1</sup> Hasan Arif Yetkin,<sup>2</sup> Tim Kodalle,<sup>2</sup> Jakob Bombsch,<sup>1</sup> Raul Garcia-Diez,<sup>1</sup> Claudia Hartmann,<sup>1</sup> Shigenori Ueda,<sup>4</sup> Johannes Frisch,<sup>1,3</sup> Regan G. Wilks,<sup>1,3</sup> Christian A. Kaufmann,<sup>2</sup> and Marcus Bär<sup>1,3,5,6</sup>

<sup>1</sup> Interface Design, Helmholtz-Zentrum Berlin für Materialien und Energie GmbH, Berlin, Germany

<sup>2</sup> PVcomB, Helmholtz-Zentrum Berlin für Materialien und Energie GmbH, Berlin, Germany

<sup>3</sup> Energy Materials In-Situ Laboratory Berlin (EMIL), Helmholtz-Zentrum Berlin für Materialien und Energie GmbH, Berlin, Germany

<sup>4</sup> NIMS Beamline Station at SPRING-8, National Institute for Materials Science (NIMS), Kouto, Sayo, Hyogo, 679-5148 Japan

<sup>5</sup> Helmholtz-Institute Erlangen-Nürnberg for Renewable Energy, Berlin, Germany

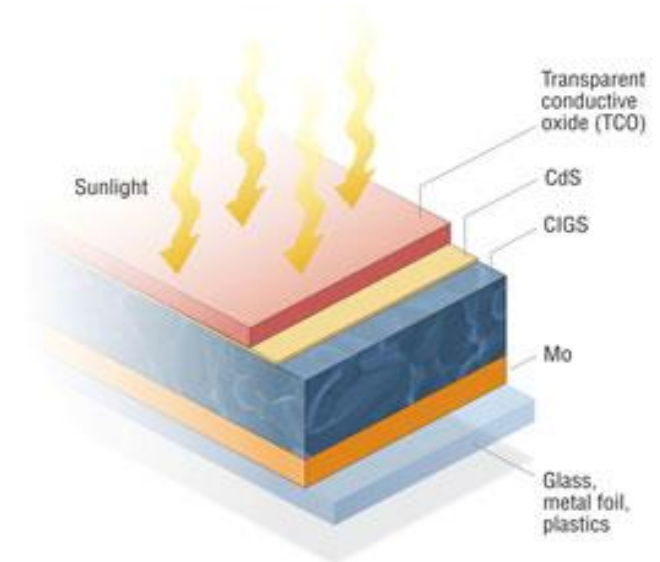
<sup>6</sup> Department of Chemistry and Pharmacy, Friedrich-Alexander-Universität Erlangen-Nürnberg, Erlangen, Germany

## Chalcopyrite $\text{Cu}(\text{In,Ga})\text{Se}_2$ - CIGSe thin-film solar cells

- High efficiency thin film solar cell
- Suitable for deposition on various substrates
- Can be flexible solar cells
- Future employment in ultra-thin and tandem solar cells

$$\eta = 23.4\%$$

Press release Solar Frontier (2019-01-17)



CIGS photovoltaic cell structure. Image by Alfred Hicks/NREL

## Further increase in efficiency of CIGSe solar cells:

- Depositing TCO at elevated temperatures to decrease optical losses
- For application in tandem devices, thermal stability of CIGSe stack is needed

Post-Deposition Annealing Treatment done on samples to simulate TCO deposition at elevated temperatures

## Post-Deposition Annealing Treatment (PDAT)\*:

- Effectively passivates defect states in CIGSe
- Decreased recombination centers density
- Increased effective acceptor density

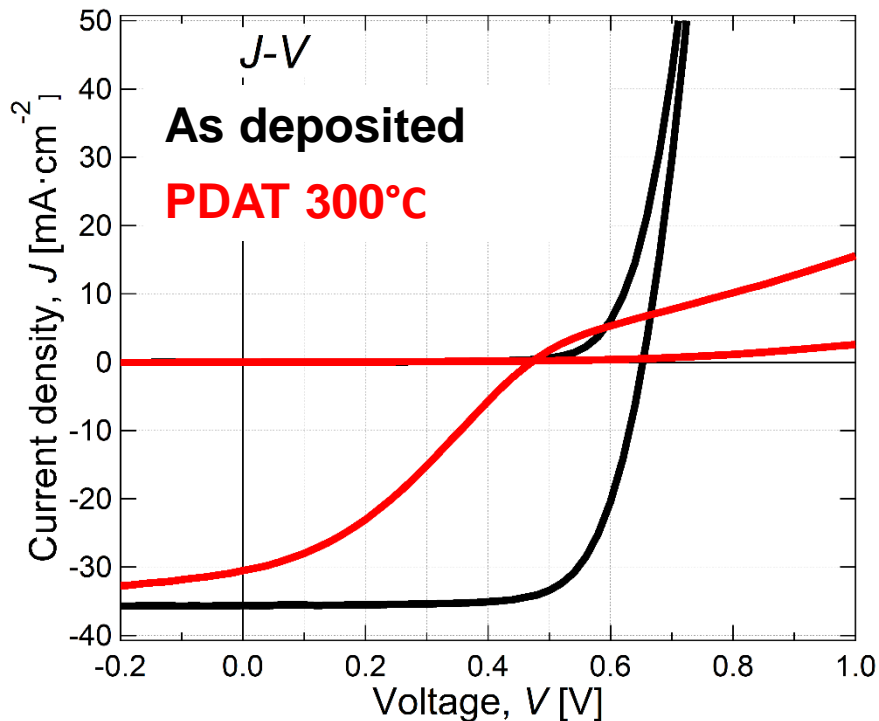


## Post-Deposition Annealing Treatment (PDAT)\*:

- Effectively passivates defect states in CIGSe
- Decreased recombination centers density
- Increased effective acceptor density



- Critical temperature 250 - 300°C → Deterioration of CdS/CIGSe stack



- PDAT treatments at 300 °C induce severe degradation in all cell parameters
- GD-OES shows: PDAT-induced diffusion of Na into the TCO and Cd diffusion into the CIGSe which leads to stack degradation

**Thermally robust buffer/absorber interface required !!!**

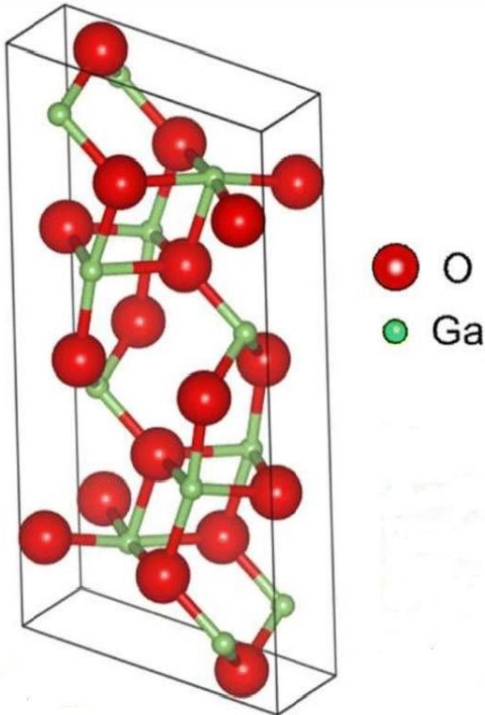
## Criteria for more thermally robust buffer layer\* :

- Diffusion barrier
- Energy band gap ( $E_g$ )  $> 3$  eV
- Buffer layer needs to provide beneficial energy level alignment

## Criteria for more thermally robust buffer layer\* :

- Diffusion barrier
- Energy band gap ( $E_g$ ) > 3 eV
- Buffer layer needs to provide beneficial energy level alignment

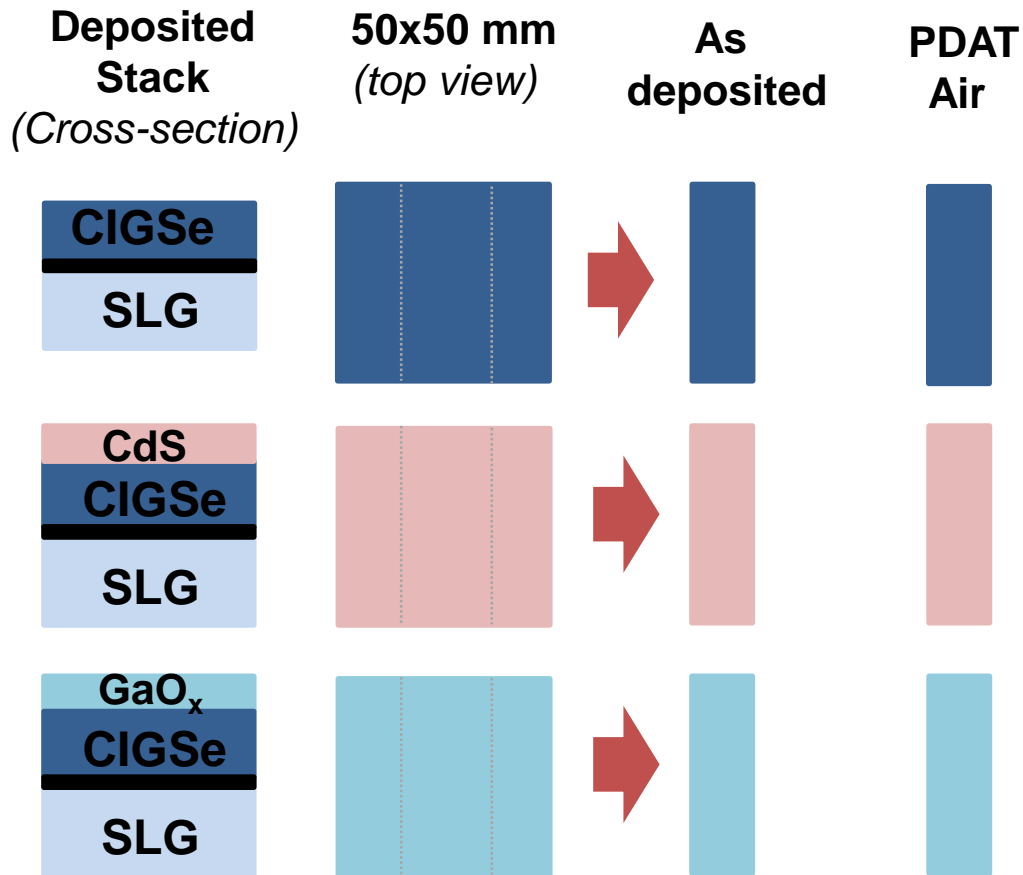
## Gallium oxide – $\text{Ga}_2\text{O}_3$



Xue et al. Nanoscale Research Letters (2018) 13:290

## $\text{Ga}_2\text{O}_3$ as a substitute for CdS buffer layer:

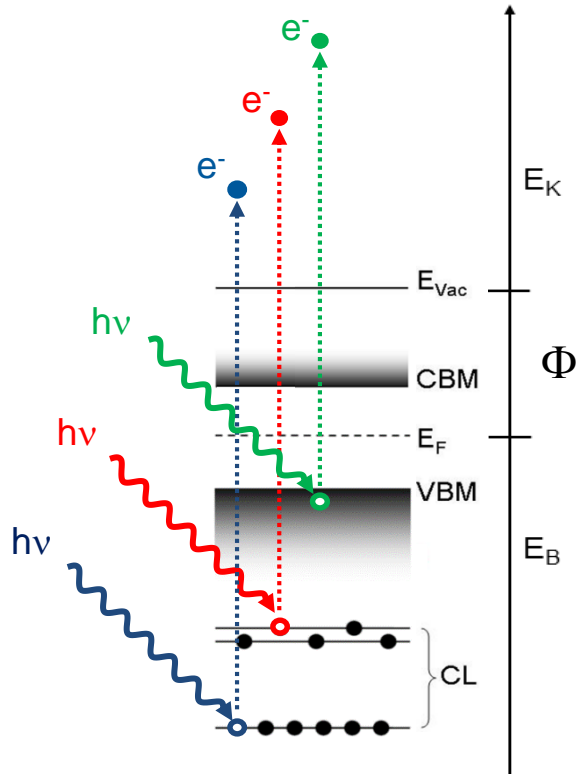
- Compatible with CIGSe
- Non-toxic, non-heavy metal compound
- Wide band gap material
- Prominent material for field effect passivation
- Thermally and chemically stable



Buffer layer	Nominal thickness (nm)
CdS	10, 20, 50
GaO <sub>x</sub>	5, 10, 20, 50

- CdS buffer layer deposited via CBD, GaO<sub>x</sub> deposition via RF- Magnetron Sputtering
- Annealed in air (PDAT – air) for 20 min at 300°C after buffer layer deposition

Photon In / Electron Out  
(Direct photoemission)



Kinetic energy of released electrons:

$$E_K = h\nu - E_B - \Phi$$

## HAXPES

$$h\nu = 5950 \text{ eV}$$

## XPS

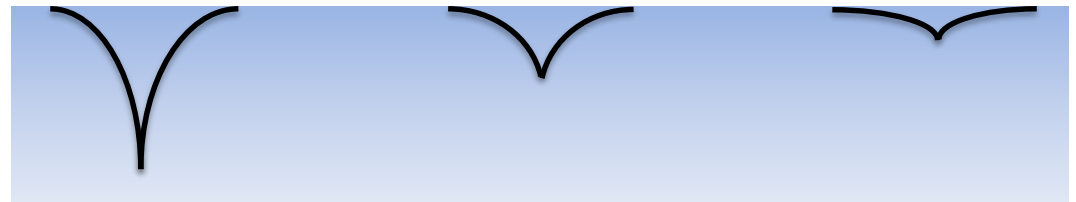
$$h\nu (\text{Mg } K_\alpha) = 1253.56 \text{ eV}$$

$$h\nu (\text{Al } K_\alpha) = 1486.58 \text{ eV}$$

## UPS

$$h\nu (\text{He I}) = 21.2 \text{ eV}$$

$$h\nu (\text{He II}) = 40.8 \text{ eV}$$



IMFP ~ 8.5 nm

IMFP ~ 2 nm

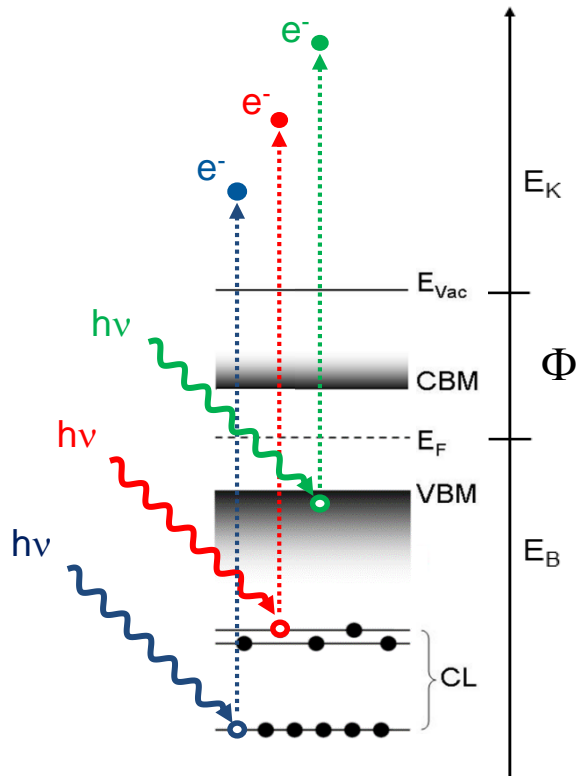
IMFP < 1 nm

- Inelastic Mean Free Path – IMFP calculated with Quases software\*



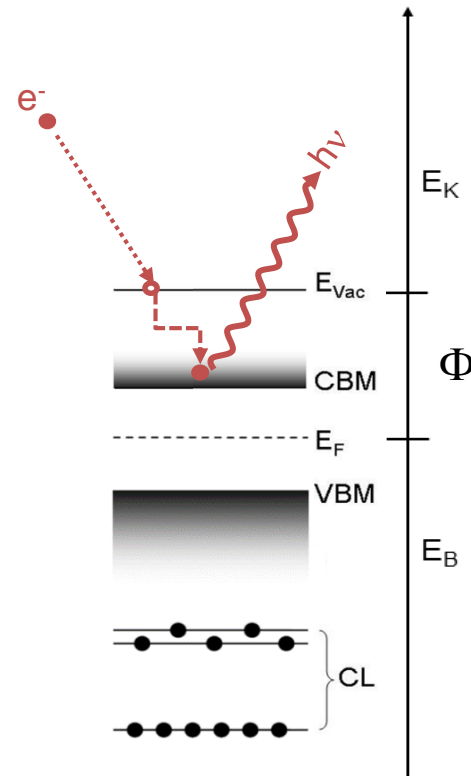
Photon In / Electron Out  
(Direct photoemission)

Electron In / Photon Out  
(Inverse photoemission)



Kinetic energy of released electrons:

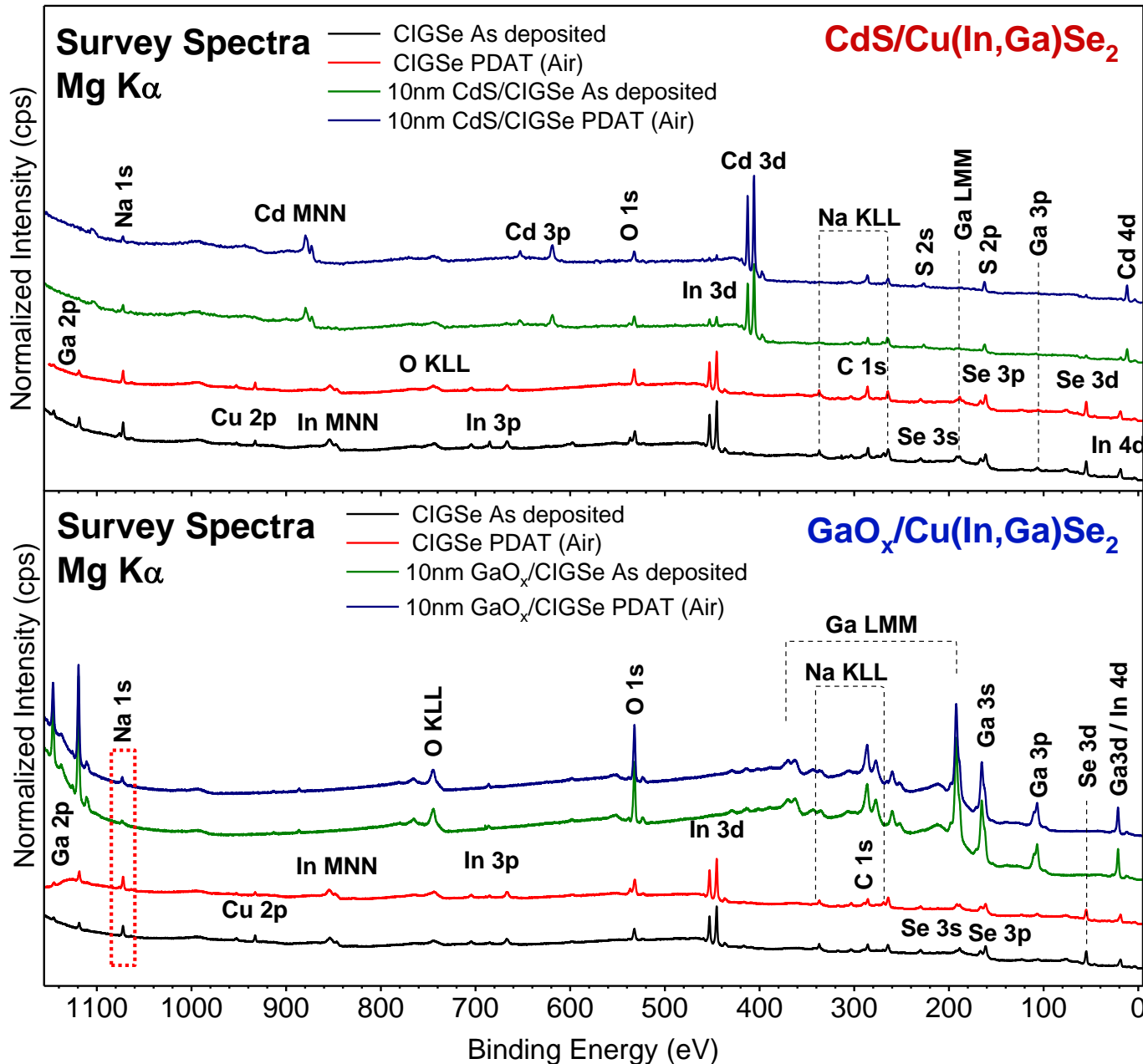
$$E_K = h\nu - E_B - \Phi$$



Detected photon energy:

$$h\nu = E_B + E_K + \Phi$$

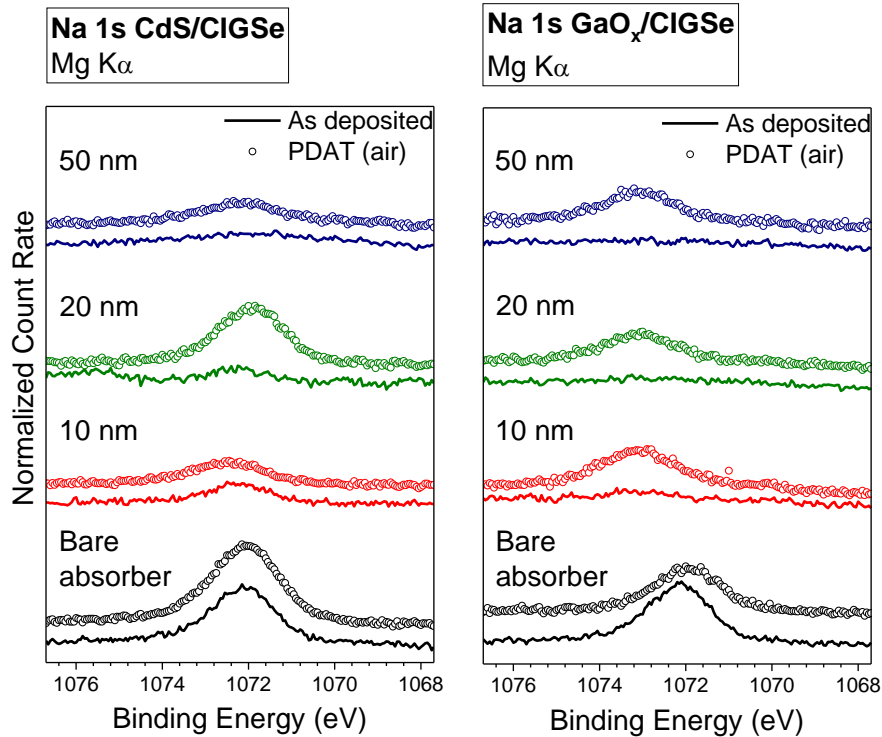
# CdS & GaO<sub>x</sub>/CIGSe Survey Spectra



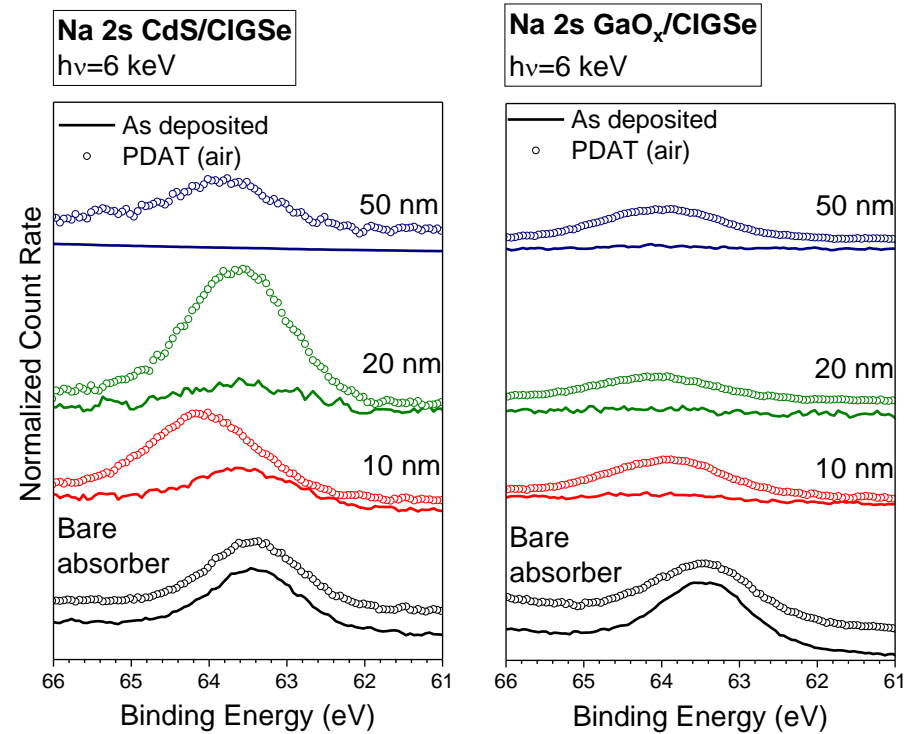
- From survey spectra visible attenuation of CIGSe peaks with buffer layer deposition
- GaO<sub>x</sub> related samples have better coverage, but Na peak still visible in spectra

# Na Diffusion Into Buffer Layer

## Na 1s



## Na 2s

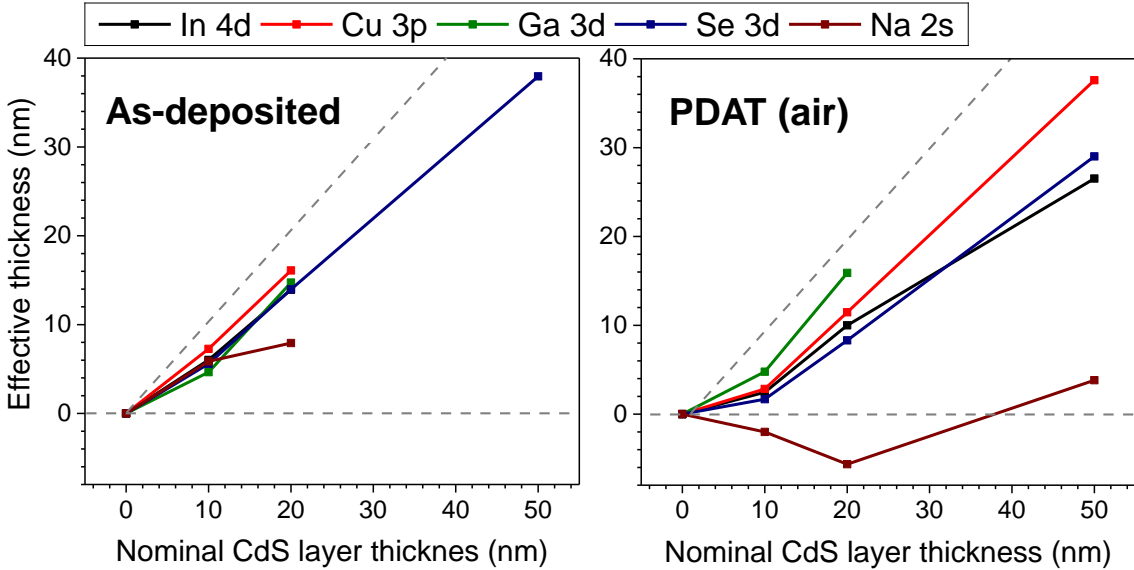


- PDAT induced Na diffusion present in all samples
- Na diffuses from SLG through whole stack
- GaO<sub>x</sub> acts as a Na barrier

Core level	Excitation energy (eV)	IMFP (nm)	Cross-section
Na 1s	1253.56	< 2	~ 100
Na 2s	5950	< 8.5	~ 0.39

# Effective Thickness Calculation

## CdS/CIGSe

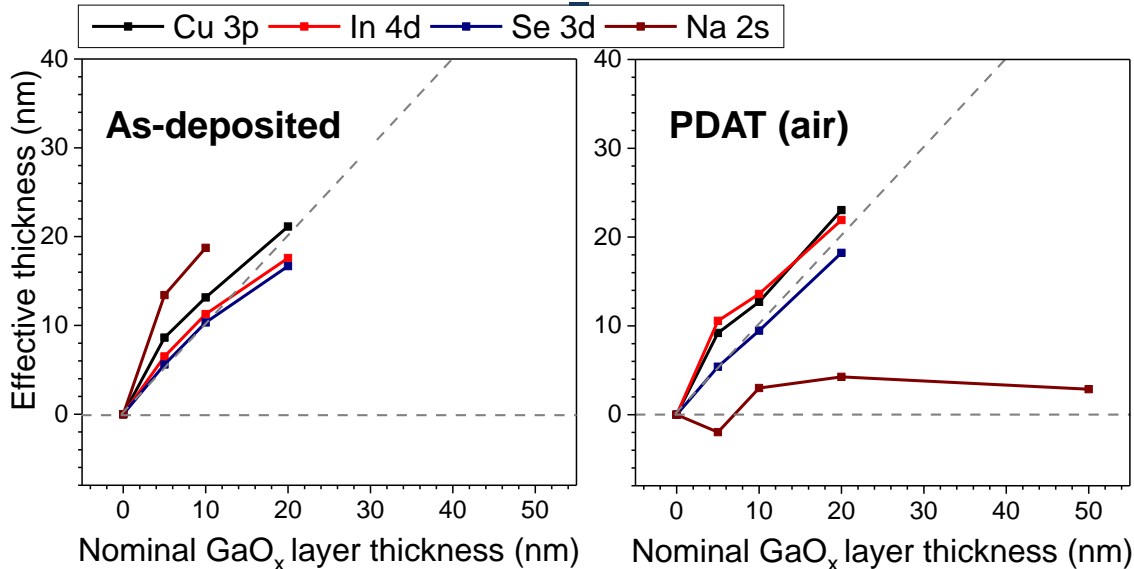


**Beer-Lambert law**

$$\frac{I}{I_0} = e^{-x/IMFP}$$

- Effective thickness calculated from shallow core levels signals via Beer-Lambert law
- Excitation energy  $h\nu = 6 \text{ eV}$ , IMFP  $\sim 8.5 \text{ nm}$
- Deposited CdS buffer layers thinner than nominal thickness

## GaO<sub>x</sub>/CIGSe



## CdS/CIGSe

## GaO<sub>x</sub>/CIGSe

As-deposited

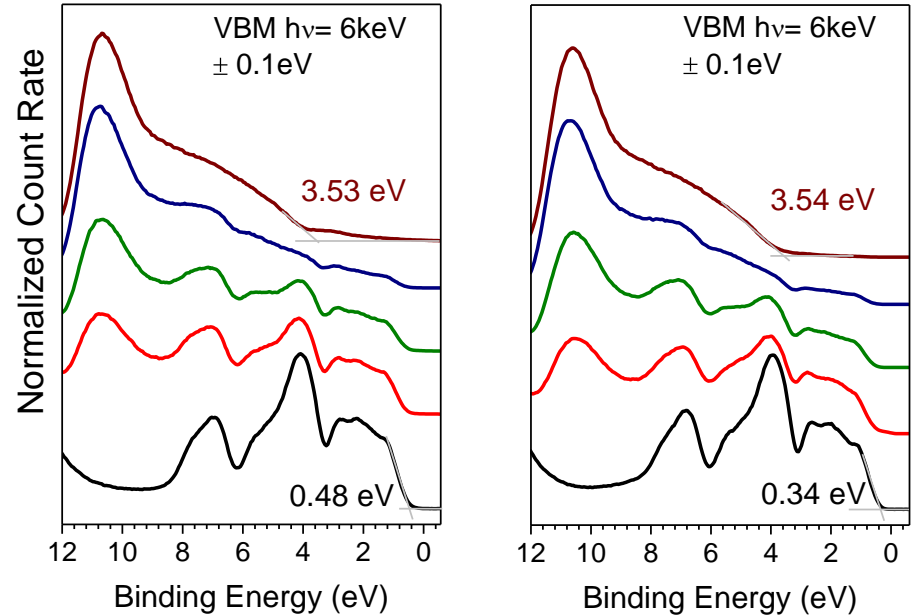
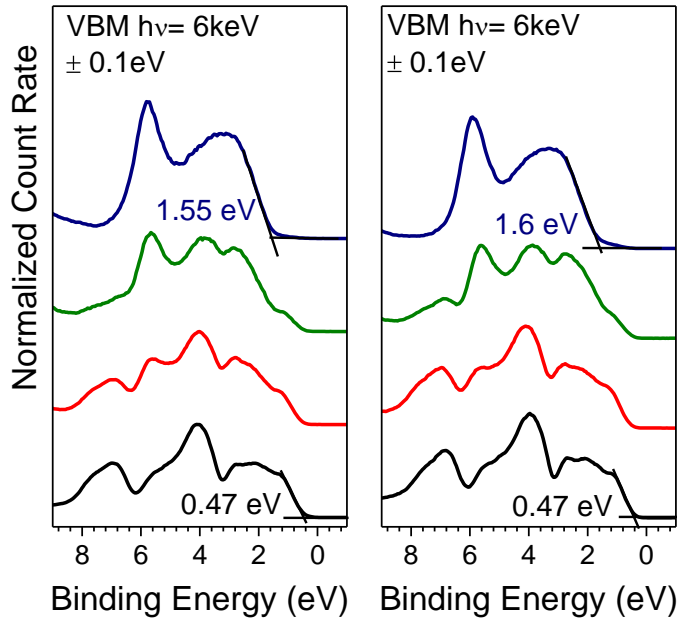
PDAT (air)

As-deposited

PDAT (air)

— Bare CIGSe — 10nm CdS/CIGSe  
— 20nm CdS/CIGSe — 50nm CdS/CIGSe

— Bare CIGSe — 5nm GaO<sub>x</sub>/CIGSe — 10nm GaO<sub>x</sub>/CIGSe  
— 20nm GaO<sub>x</sub>/CIGSe — 50nm GaO<sub>x</sub>/CIGSe



- VBM shape changes with increasing buffer layer thickness
- PDAT shows no major effect on VBM position

## CdS/CIGSe

## GaO<sub>x</sub>/CIGSe

As-deposited

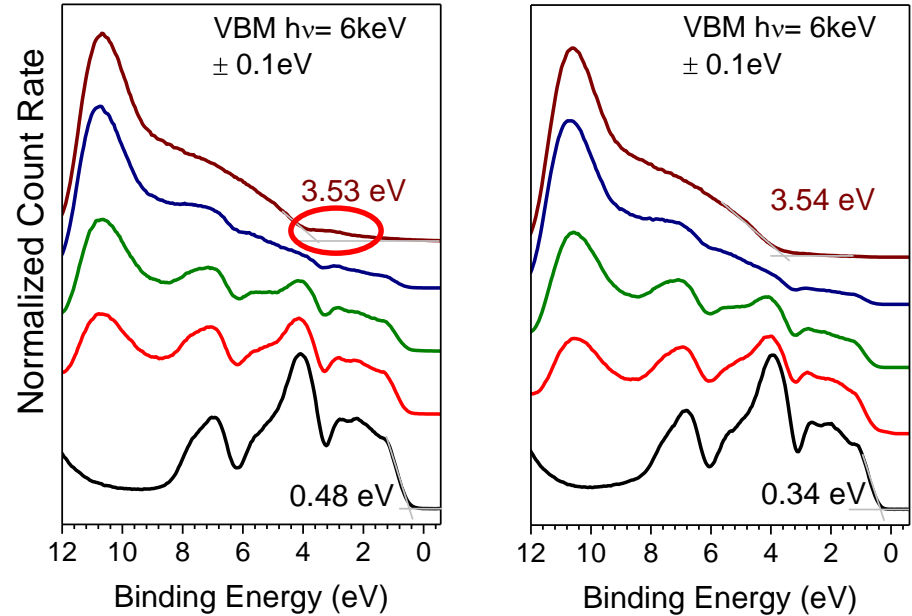
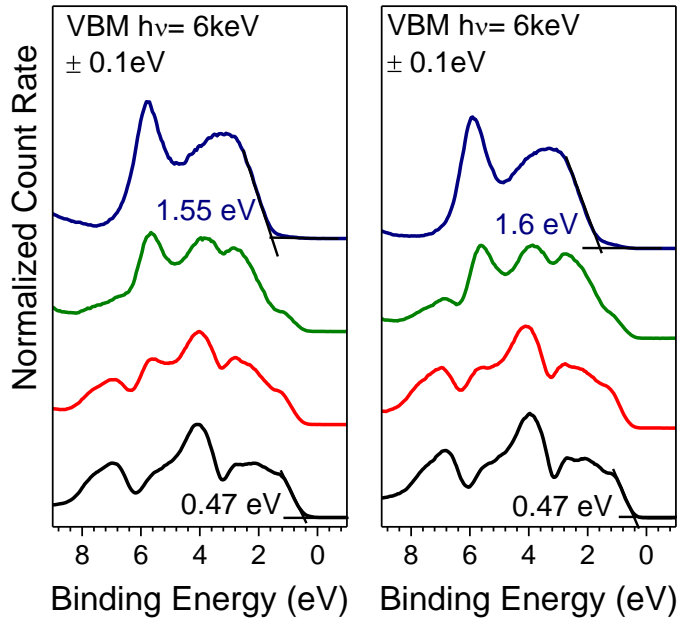
PDAT (air)

As-deposited

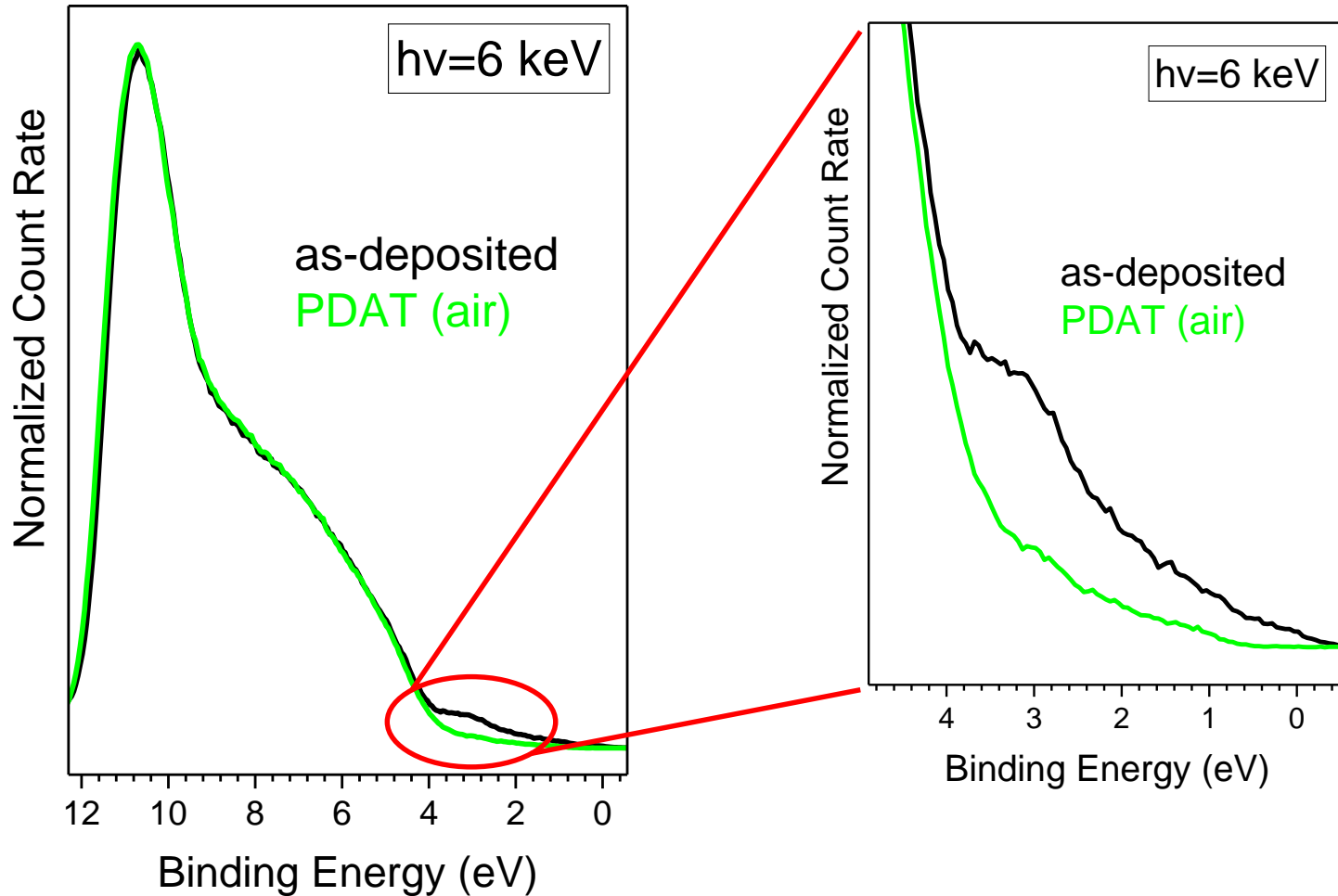
PDAT (air)

— Bare CIGSe — 10nm CdS/CIGSe  
— 20nm CdS/CIGSe — 50nm CdS/CIGSe

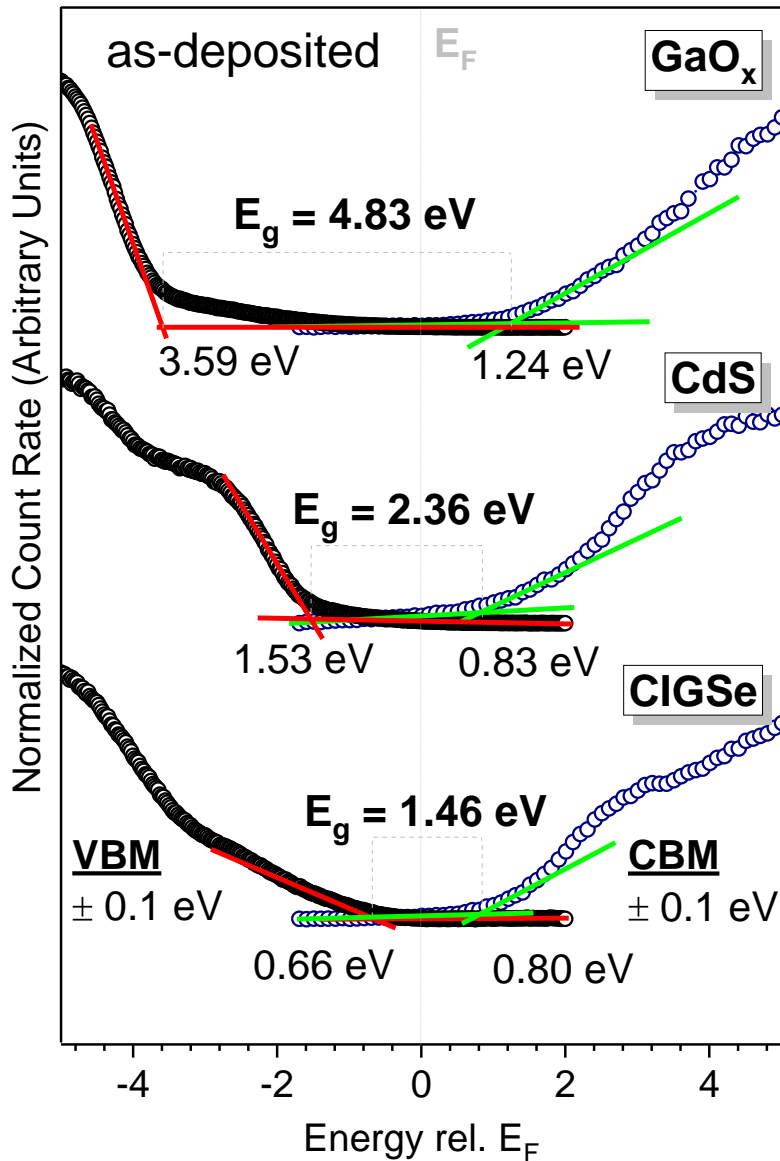
— Bare CIGSe — 5nm GaO<sub>x</sub>/CIGSe — 10nm GaO<sub>x</sub>/CIGSe  
— 20nm GaO<sub>x</sub>/CIGSe — 50nm GaO<sub>x</sub>/CIGSe



- VBM shape changes with increasing buffer layer thickness
- PDAT shows no major effect on VBM position
- Spectral intensity above VBM visible for thickest as-deposited GaO<sub>x</sub> layer



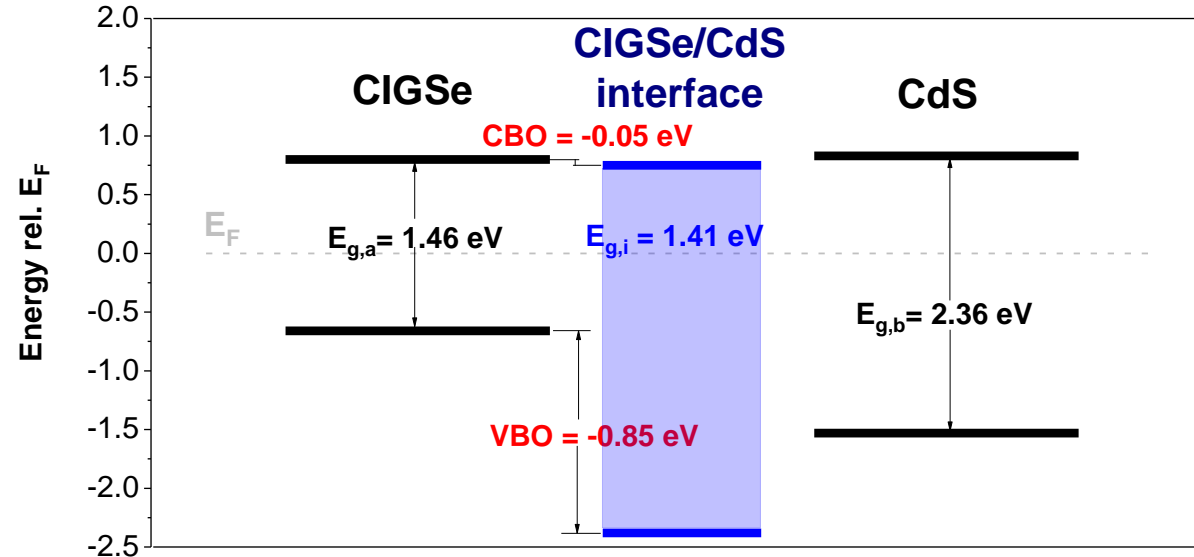
- Oxygen related near surface defect density visible in as-deposited GaO<sub>x</sub> sample
- Surface defect density decreases upon PDAT



Sample	$E_g$ (eV) UPS/IPES	Optical $E_g$ (eV)
CdS	2.36 eV	2.42 eV*
GaO <sub>x</sub>	4.83 eV	4.71 eV

- Positions obtained by linear extrapolation of leading edge in UPS and IPES data to estimate VBM, CBM values and energy band gap ( $E_g$ )

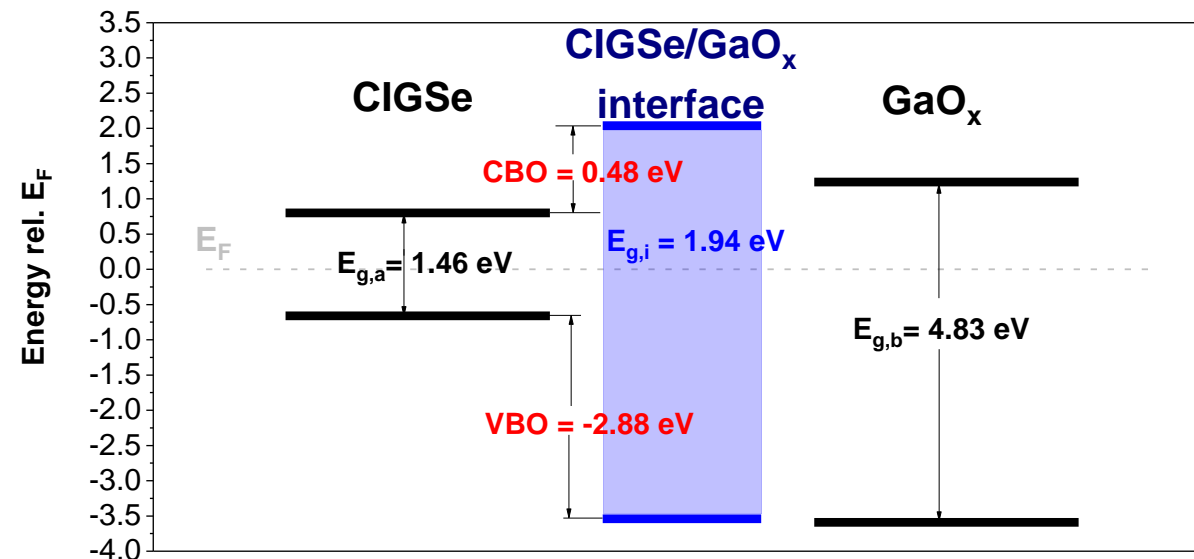




- CBM alignment at the CdS/CIGSe interface in alignment with achieving high efficiencies

- Increase of energetic barrier for charge carrier transport at CIGSe/ $\text{GaO}_x$  interface

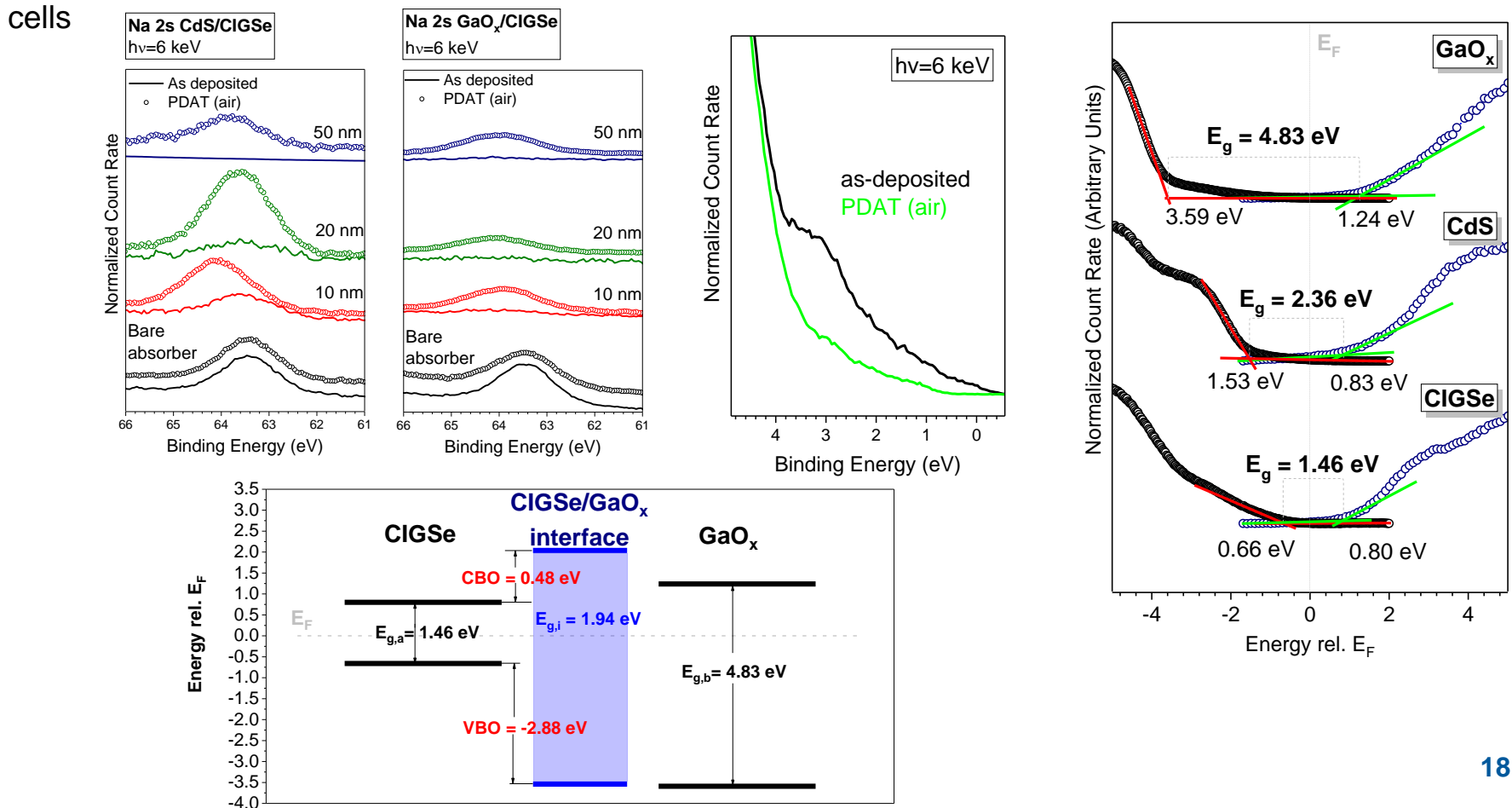
- Replacement of CdS with  $\text{GaO}_x$  increases interface band gap, the energy barrier for recombination across the (defect rich) buffer/absorber interface



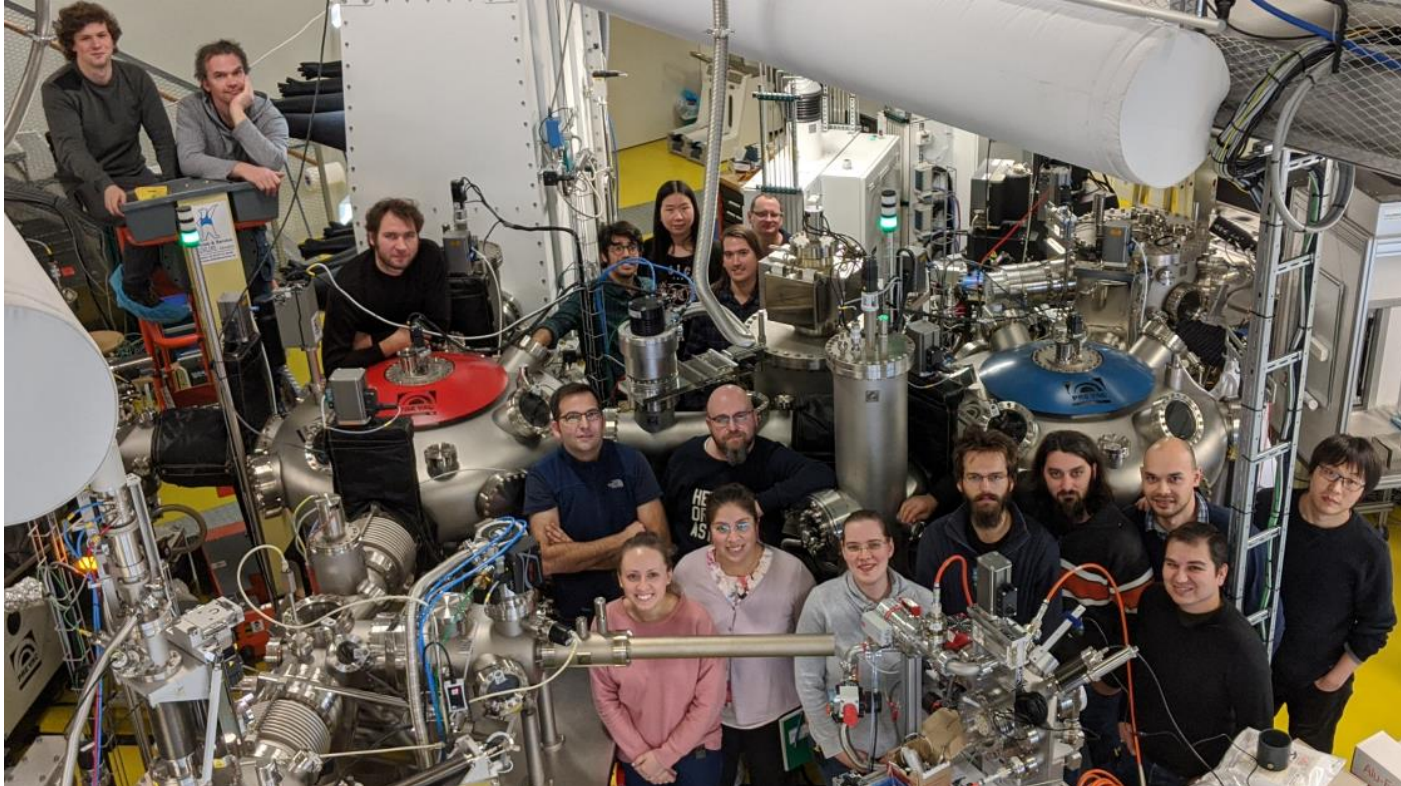
- $\text{GaO}_x$  (if designed properly) could be used as passivation layer in CIGSe-based solar cells\*

# Summary

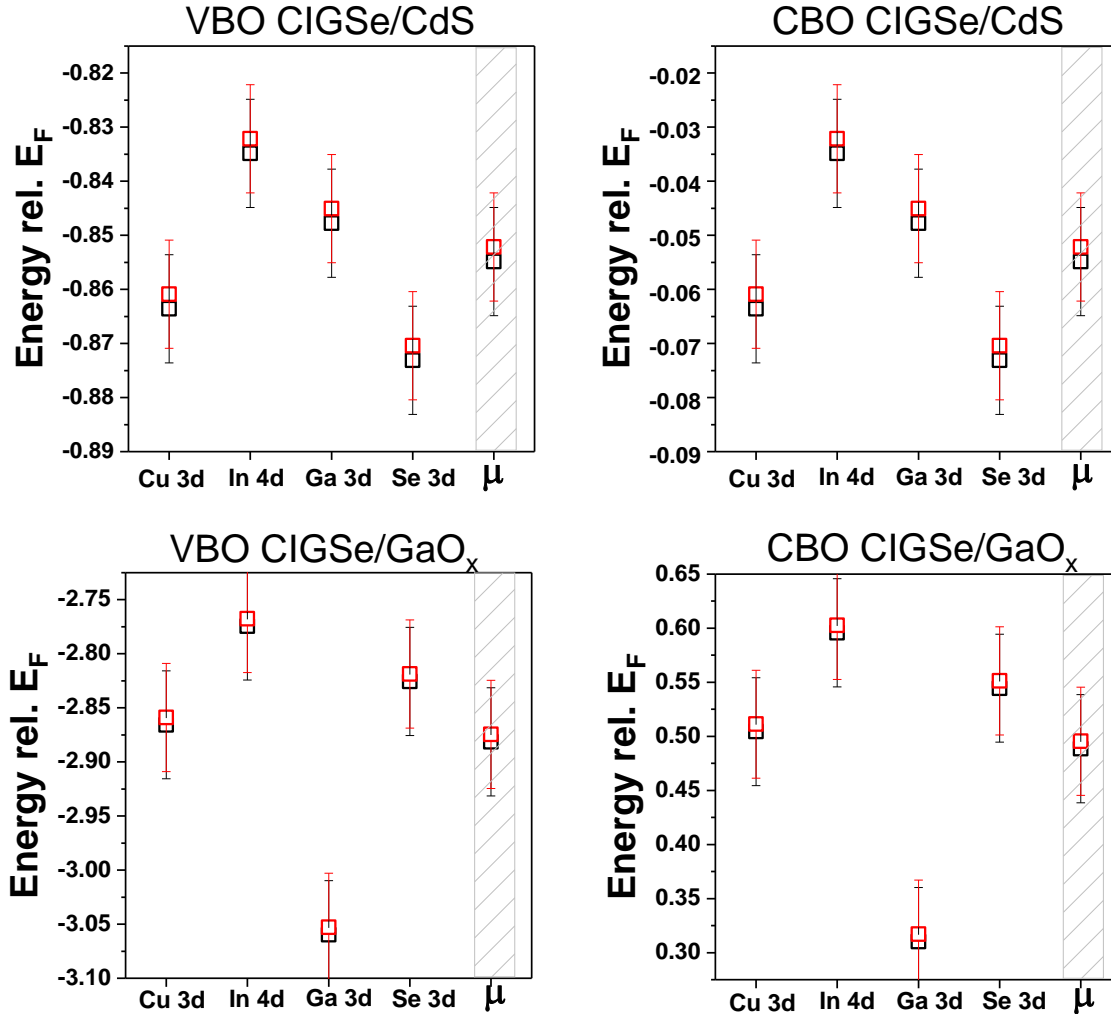
- Na diffusion in  $\text{GaO}_x$  less pronounced than in the CdS/CIGSe samples
- No prominent PDAT influence on VBM values except to the defect related states in  $\text{GaO}_x$
- PDAT passivates oxygen related defects  $\text{GaO}_x$  in VBM
- Derived energy level alignment suggest the application of  $\text{GaO}_x$  as passivation layer in CIGSe based solar cells



# Acknowledgment



**THANKS FOR YOUR ATTENTION!**



$$VBO = (CL_{cap} - VBM_{cap}) - (CL_{sub} - VBM_{sub}) - (CL_{cap}^{int} - CL_{sub}^{int}) \quad (1)$$

$$CBO = (E_g^{cap} - E_g^{sub}) + VBO \quad (2)$$