

TEM study of annealing effects on chemical and structural properties of solution-processed Zn(O,S) buffer layers in Cu(In,Ga)Se₂ solar cells

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Outline

1. **Motivation**
2. **Experimental**
3. **Chemical properties of the as-grown & annealed Zn(O,S) buffers**
4. **Bandgap energies of the Zn(O,S) buffers before & after annealing**
5. **Crystal structure of the Zn(O,S) buffers and corresponding phase analyses**
6. **Conclusions**

Motivation

■ Zn(O,S)

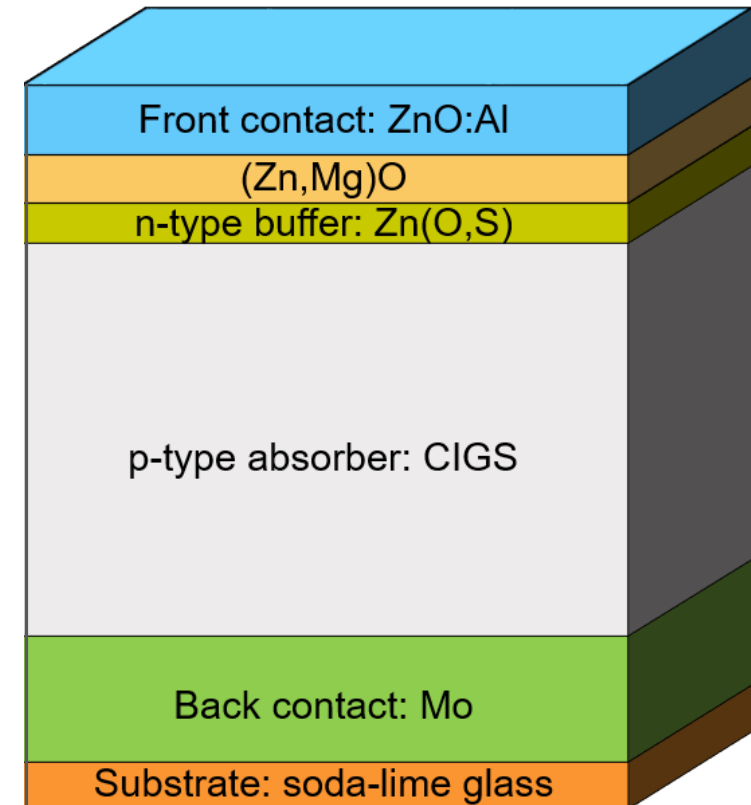
- non-toxic
- wide & tunable bandgap (2.6 - 3.6 eV) [1]
- Cu(In,Ga)Se₂ (CIGS) solar cells with solution-processed Zn(O,S) show high conversion efficiency above 23% [2]
- earth abundant, low cost, ...
 - promising buffer material to replace CdS

■ Post-growth annealing

- influences achieved performance

■ Transmission electron microscopy

- characterize chemical & structural properties
 - study influence of annealing on microchemistry & microstructure



[1] B.K. Meyer *et al.*, Appl. Phys. Lett. **85** (2004) 4929

[2] M. Nakamura *et al.*, IEEE J. Photovolt. **9** (2019) 1863

■ Fabrication of solar cells

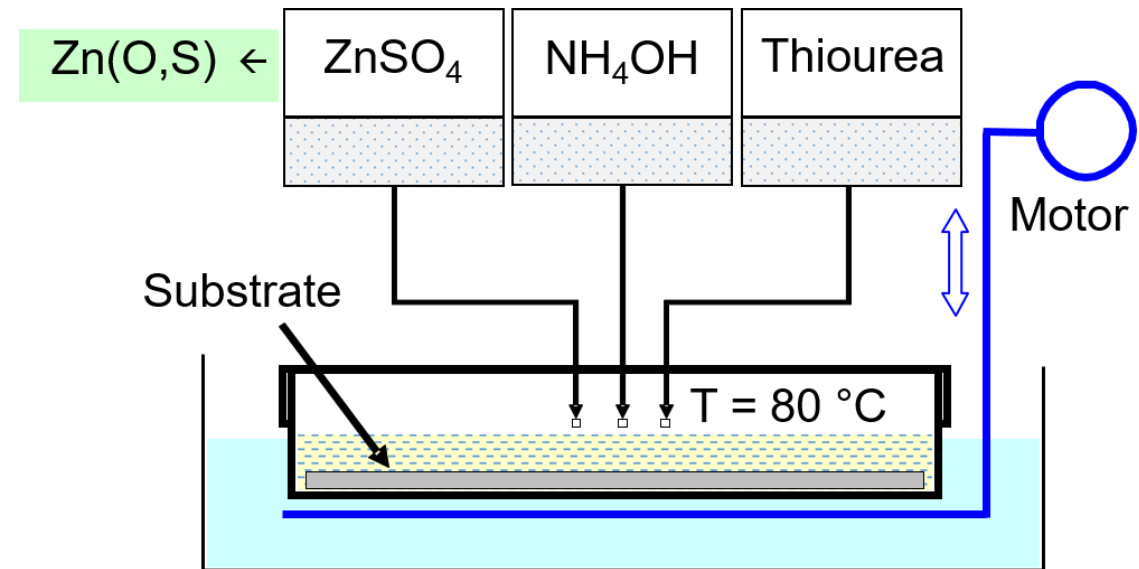
- CIGS: in-line multi-stage coevaporation
- Zn(O,S): chemical bath deposition (CBD), 20 nm
- (Zn,Mg)O: rf-sputtering
- ZnO:Al: dc-sputtering

■ Annealing

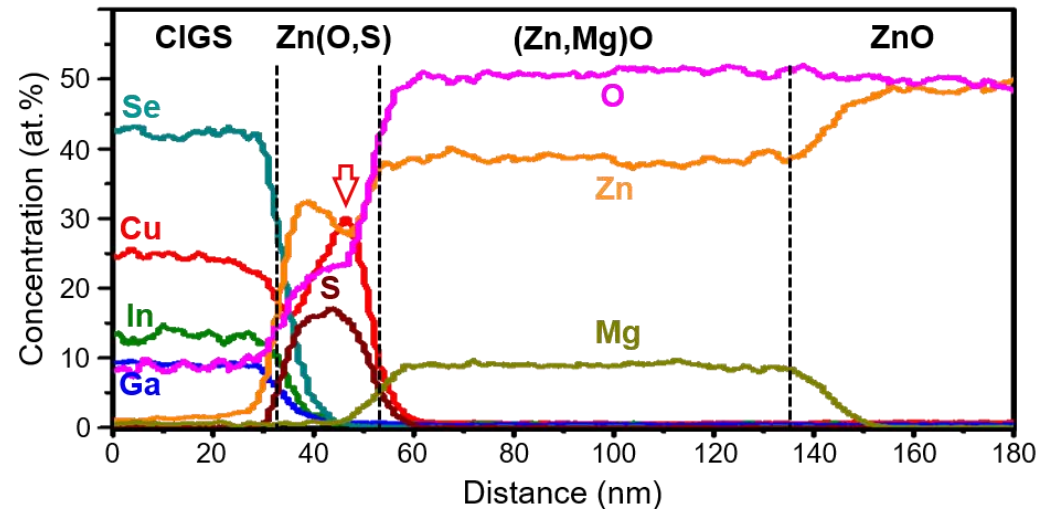
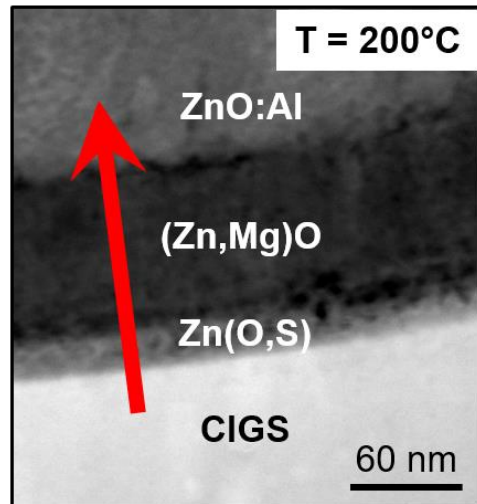
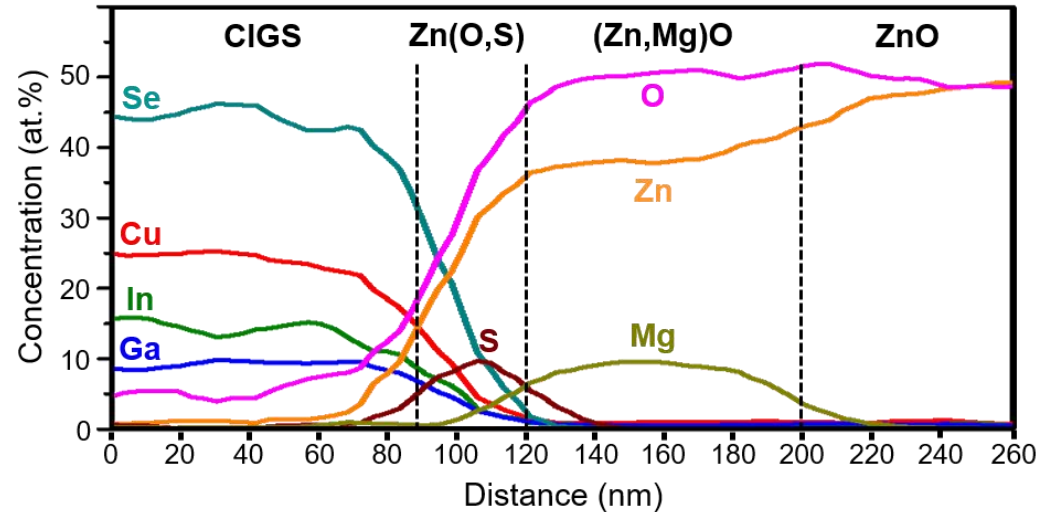
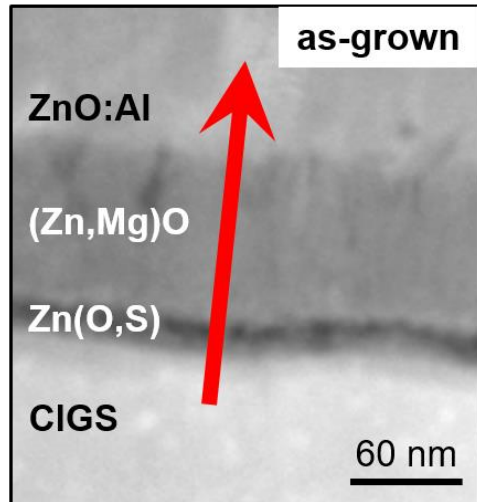
- performed in air on a hot plate, 30 min
- first @ 150°C, then @ 200°C

■ Characterization and phase analysis

- bandgap measurements: angle-resolved electroreflectance spectroscopy (ARER)
- chemical analysis: energy-dispersive X-ray spectroscopy combined with scanning transmission electron microscopy (STEM/EDXS)
- crystal structure: high-resolution transmission electron microscopy (HRTEM) and nanobeam electron diffraction (NBD)



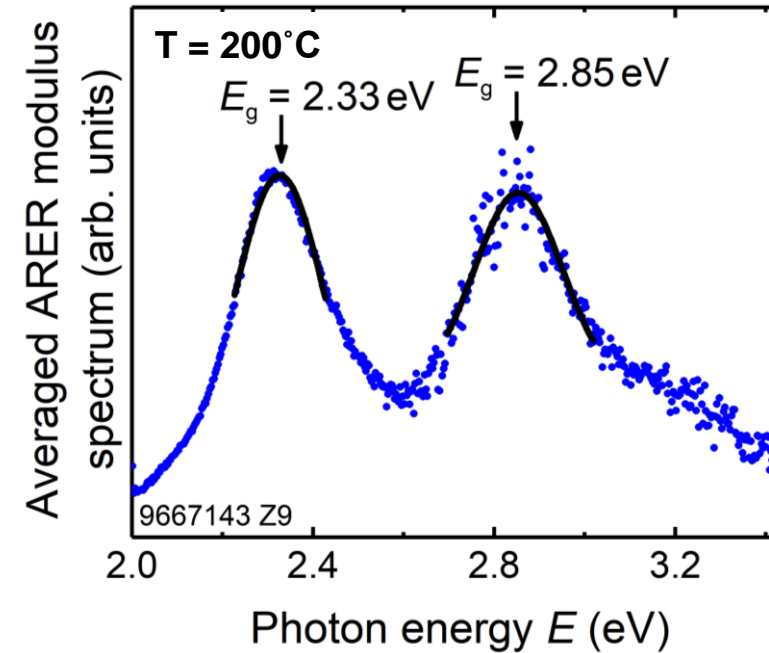
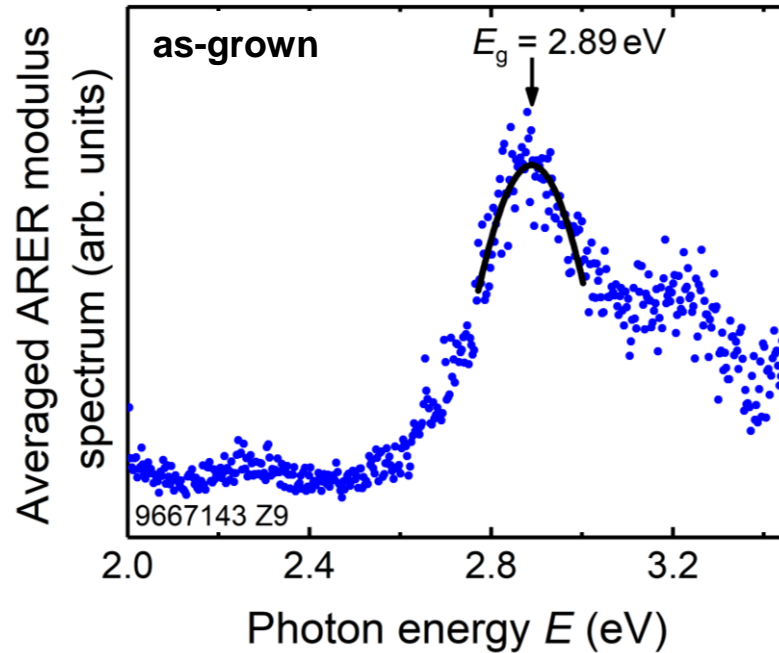
Chemical properties of the Zn(O,S) buffers



- In as-grown Zn(O,S) buffer, only Zn, O & S present, more O than S; other elements are artifacts
- In annealed Zn(O,S) buffer, the Cu concentration significantly increases, while the Zn concentration decreases
 → interdiffusion of Cu and Zn
- SSO* value increases after annealing

*SSO: $[S]/([S]+[O])$

Bandgap energies of the Zn(O,S) buffers

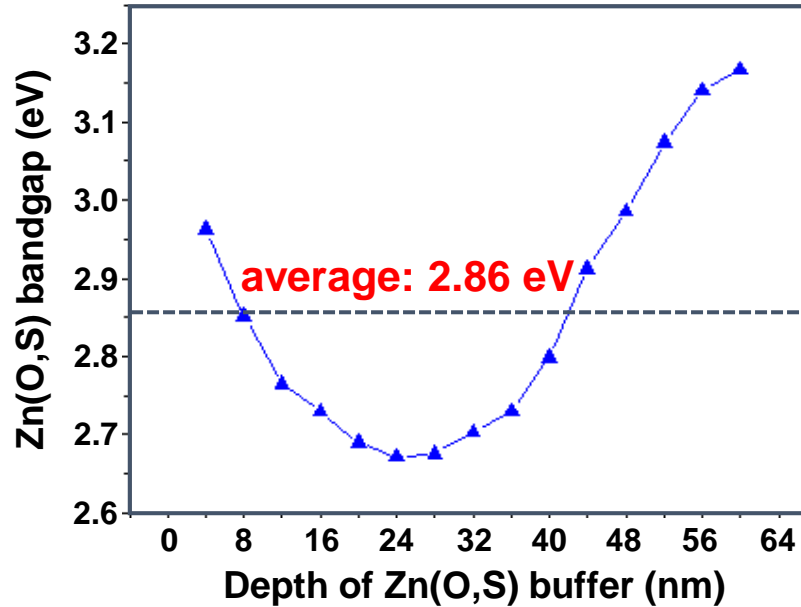
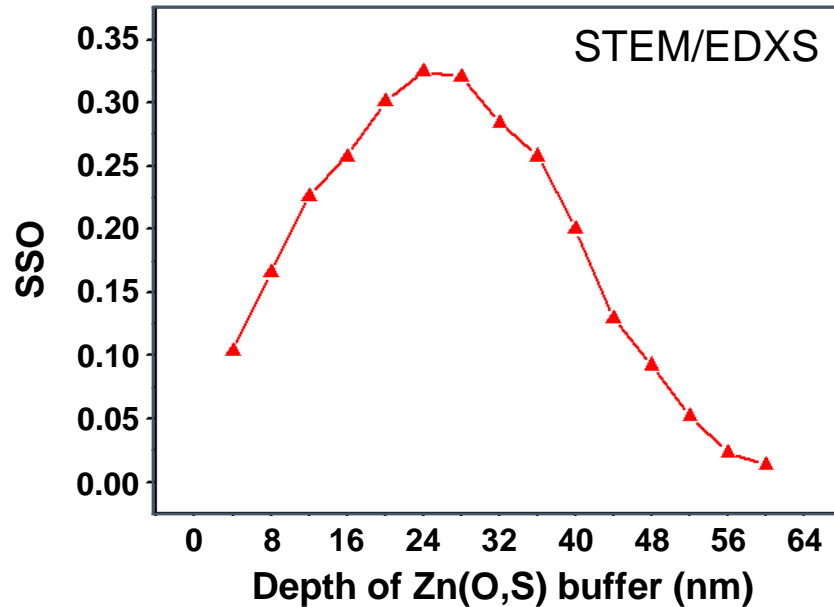


- Before annealing, one resonance at 2.89 eV is detected → bandgap energy of Zn(O,S) (2.6 - 3.6 eV)
- After annealing at 200°C for 30 min, bandgap of Zn(O,S) slightly changes (2.89 eV → 2.85 eV)
- An additional resonance at 2.33 eV → a secondary phase in the buffer → CuS (2.36 eV), Cu₂Se (2.23 eV), or Cu₃Se₂ (2.37 eV)

J. Seeger *et al.*, IEEE 46th PVSC, USA (2019) 0949

J. Seeger *et al.*, Appl. Phys. Lett. 115 (2019) 263901

Bandgap energies of the Zn(O,S) buffers

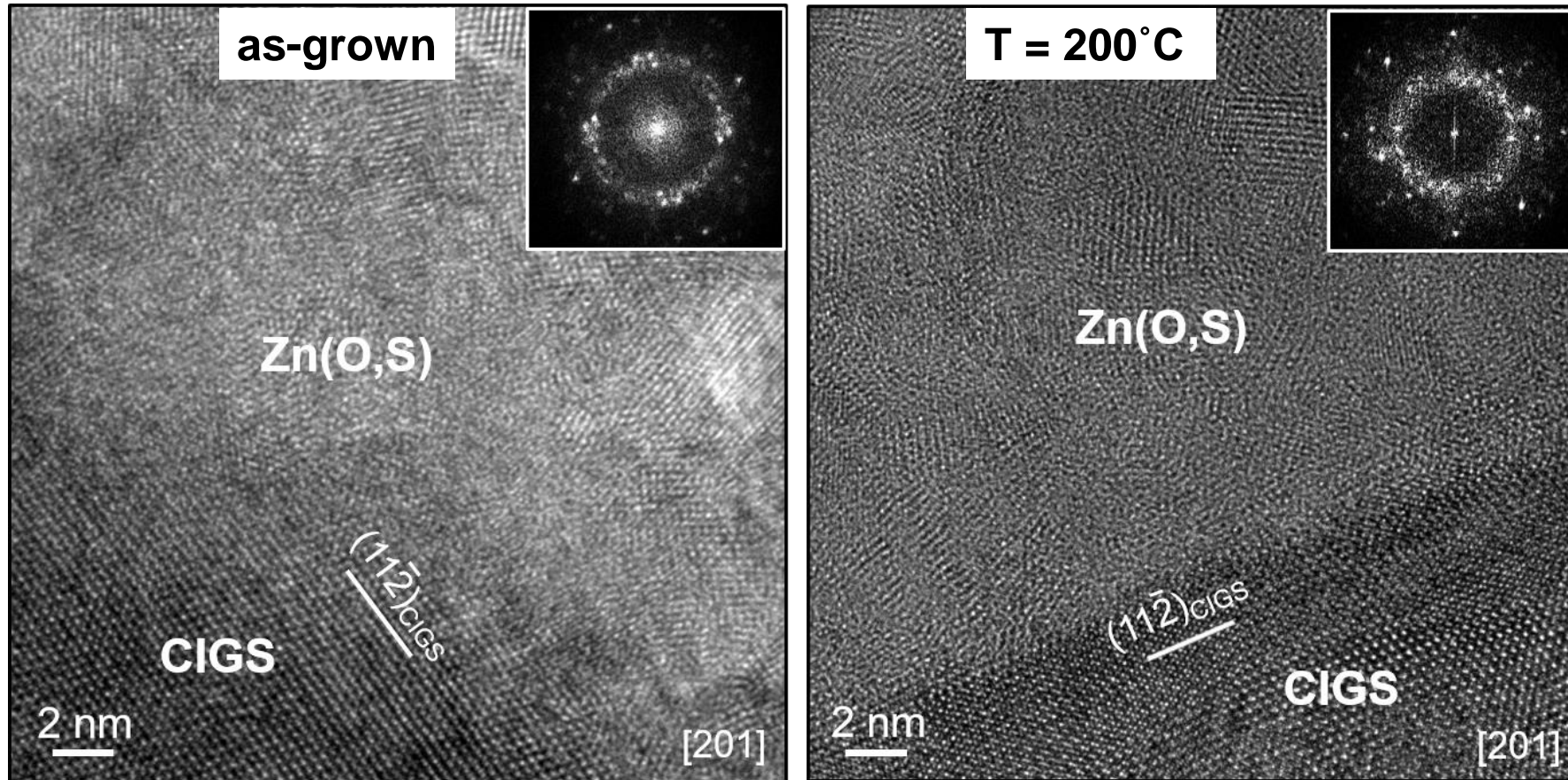


- $E_{\text{Zn(O,S)}}(x) = x E_{\text{ZnS}} + (1-x)E_{\text{ZnO}} - b(1-x)x$ [1]
 - $E_{\text{Zn(O,S)}}(x)$: bandgap of Zn(O,S)
 - E_{ZnS} : bandgap of ZnS (3.6 eV)
 - E_{ZnO} : bandgap of ZnO (3.2 eV)
 - x : SSO value in Zn(O,S)
 - b : a bowing parameter (~ 3.0 eV)

- Average calculated bandgap 2.86 eV from STEM/EDXS results and the measured bandgap 2.89 eV by ARER match each other
 - ➔ reliability of STEM/EDXS and ARER measurements

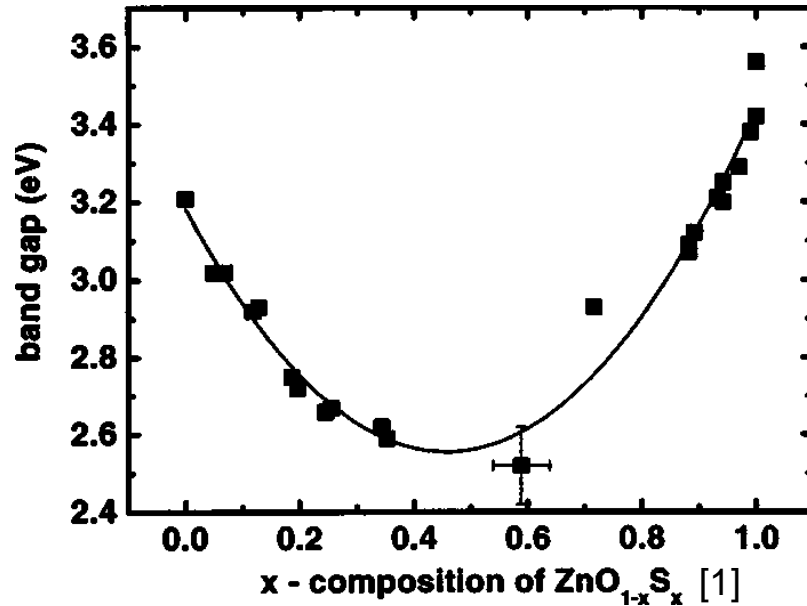
[1] B.K. Meyer *et al.*, Appl. Phys. Lett. **85** (2004) 4929

Crystal structure of the Zn(O,S) buffers

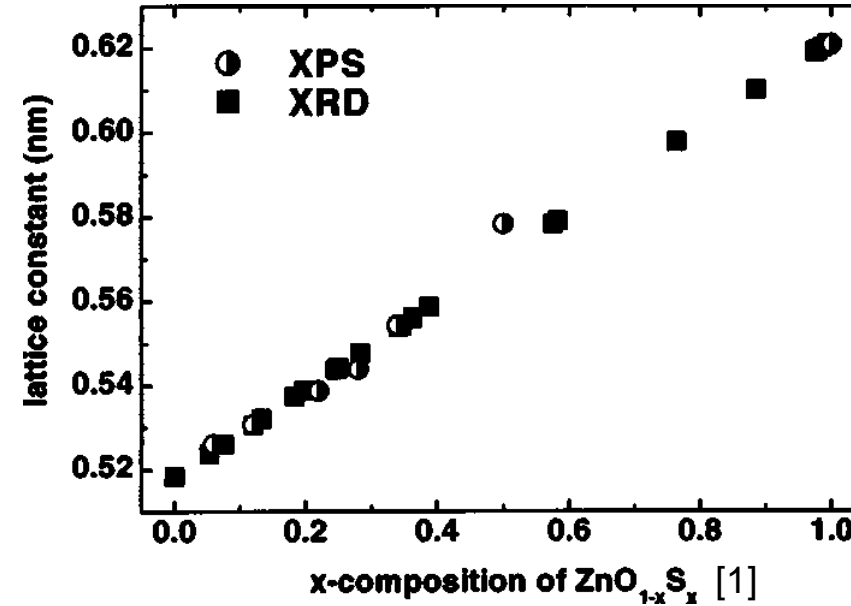


- Nanocrystalline structure of the Zn(O,S) buffer layer, independent of the orientation of the underlying CIGS grains
- Annealing @ 200°C does not significantly change the crystallinity of the buffer layer
- Detailed analysis of chemical phases within Zn(O,S) buffer layers would be performed by NBD

Phase analyses of the Zn(O,S) buffers



$$E_{\text{Zn(O,S)}}(x) = x E_{\text{ZnS}} + (1-x)E_{\text{ZnO}} - b(1-x)x$$

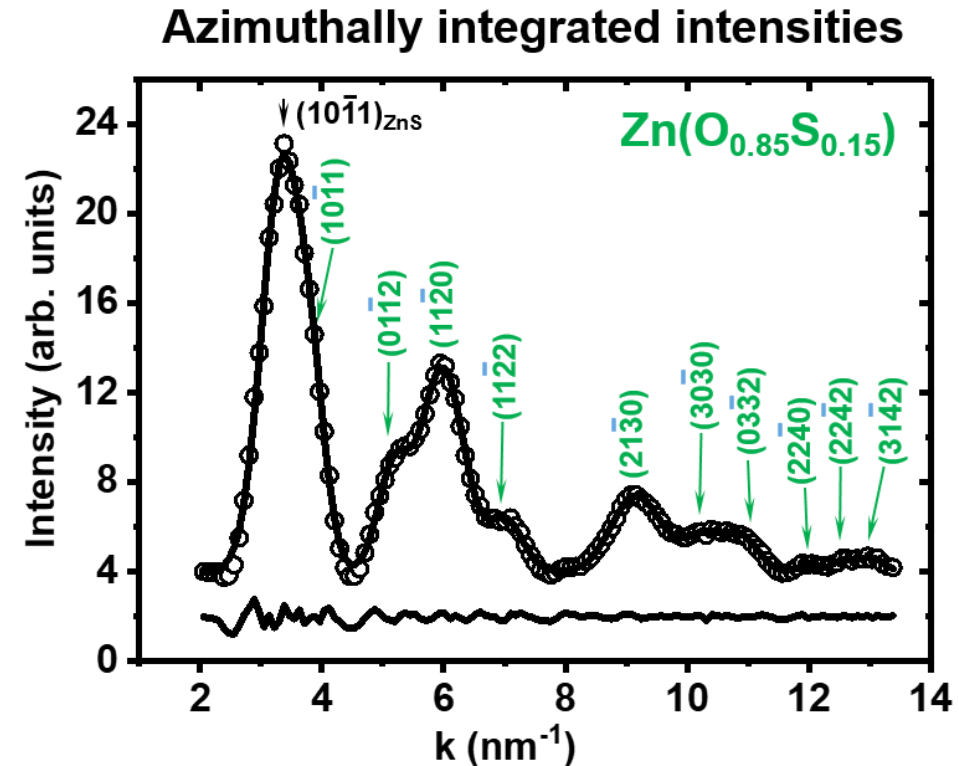
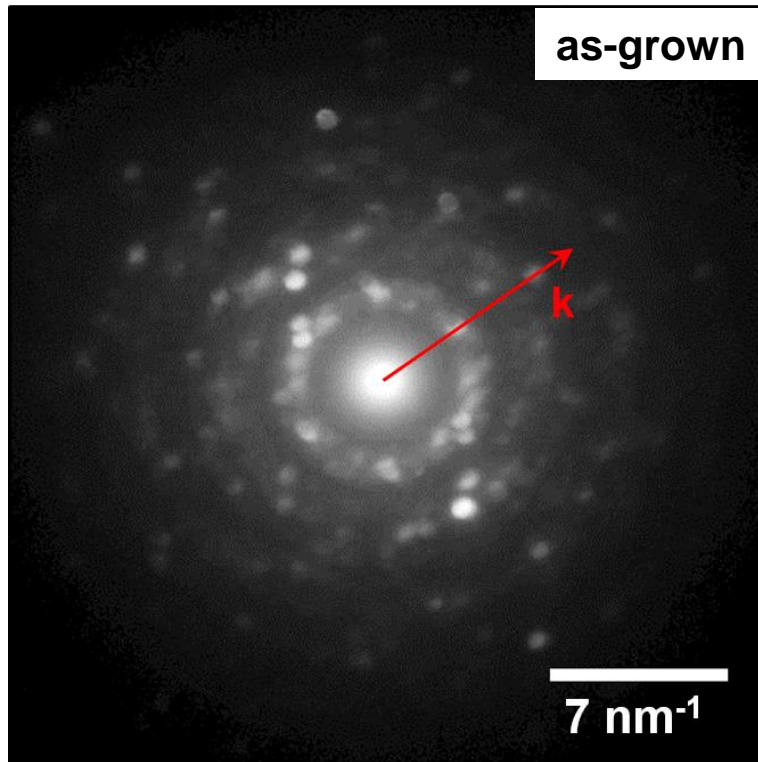


$$\text{Lattice constant} = 0.1x + 0.52$$

- As-grown Zn(O,S) layer $E_{\text{Zn(O,S)}}(x) = 2.89 \text{ eV} \rightarrow$ hexagonal $\text{Zn}(\text{O}_{0.28}\text{S}_{0.72})$ or $\text{Zn}(\text{O}_{0.85}\text{S}_{0.15})$
- Annealed Zn(O,S) layer $E_{\text{Zn(O,S)}}(x) = 2.85 \text{ eV} \rightarrow$ hexagonal $\text{Zn}(\text{O}_{0.30}\text{S}_{0.70})$ or $\text{Zn}(\text{O}_{0.83}\text{S}_{0.17})$
- Lattice parameters of the possible Zn(O,S) phases can be determined accordingly, which can be used to identify the present phase

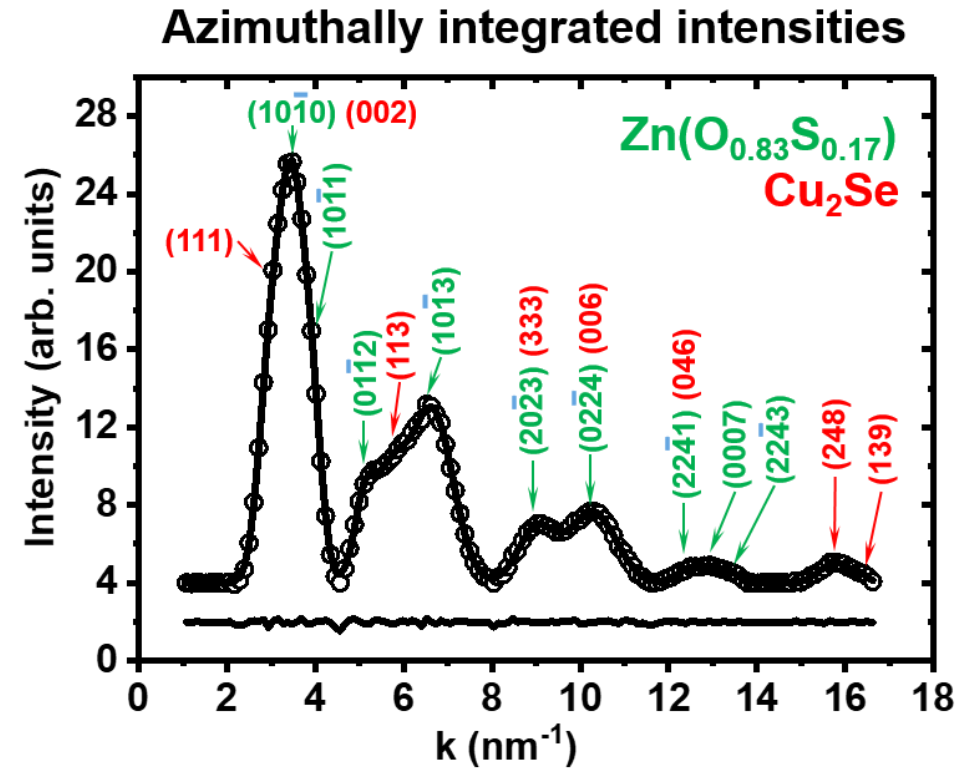
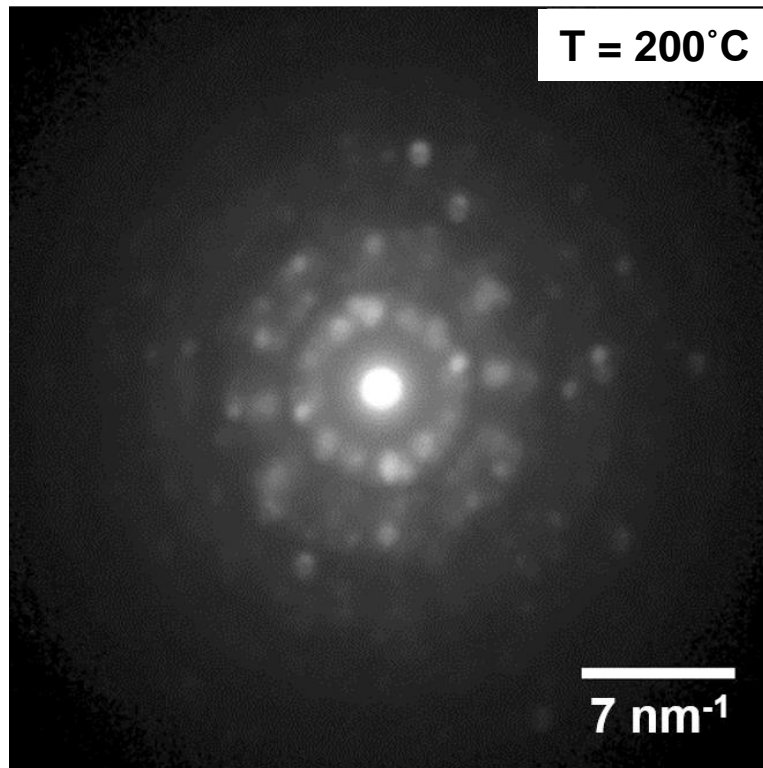
[1] B.K. Meyer *et al.*, Appl. Phys. Lett. **85** (2004) 4929

Phase analyses of the Zn(O,S) buffers (as-grown)



- Zn(O_{0.85}S_{0.15}) with wurtzite structure (P6₃mc, a = b = 3.3 Å, c = 5.3 Å) fits the NBD pattern better than Zn(O_{0.28}S_{0.72}), in agreement with the STEM/EDXS results
- ZnS with wurtzite structure (P6₃mc, a = b = 3.8 Å, c = 6.3 Å) may be also present in the buffer, whose bandgap (3.6 eV) cannot be detected by ARER (maximum 3.5 eV)

Phase analyses of the Zn(O,S) buffers ($T = 200^\circ\text{C}$)



- Zn(O_{0.83}S_{0.17}) with wurtzite structure (P6₃mc, $a = b = 3.3 \text{ \AA}$, $c = 5.4 \text{ \AA}$) fits the NBD pattern better than Zn(O_{0.30}S_{0.70})
- Cubic Cu₂Se (Fm-3m, $a = 5.7 \text{ \AA}$) fits the NBD pattern of this region; hexagonal CuS (P6₃mc, $a = b = 3.8 \text{ \AA}$, $c = 16.3 \text{ \AA}$) exists in another region as shown by HRTEM imaging → coexistence of Cu₂Se & CuS

Conclusions

- Annealing at 200°C leads to Cu diffusion from CIGS into the solution-processed Zn(O,S) buffer layer, forming a secondary phase
- Cubic Cu₂Se and hexagonal CuS are the most likely Cu-containing secondary phases with a bandgap energy of ~2.3 eV
- Due to the limitations of ARER, HRTEM and NBD, other phases cannot be excluded
- Annealing at 200°C does not significantly change the crystallinity (nanocrystalline) of the Zn(O,S) buffer layer, and slightly changes the stoichiometry of Zn(O,S)

Acknowledgement

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