

# X-ray CoreLab Facilities at HZB

HyPerCell/HySPRINT Workshop, 12. 10. 2017

## X-Ray CoreLab at HZB

- Methods and instruments
- Registration and booking

Susan Schorr  
Chair of the X-Ray CoreLab Steering Committee

## Mission statement

The **mission** of the X-Ray CoreLab is to use and to anchor the methods of lab-scale X-ray diffraction on an institutional and cross-cutting level in the HZB's strategy.

The X-ray CoreLab is supervised by a **Steering Committee**

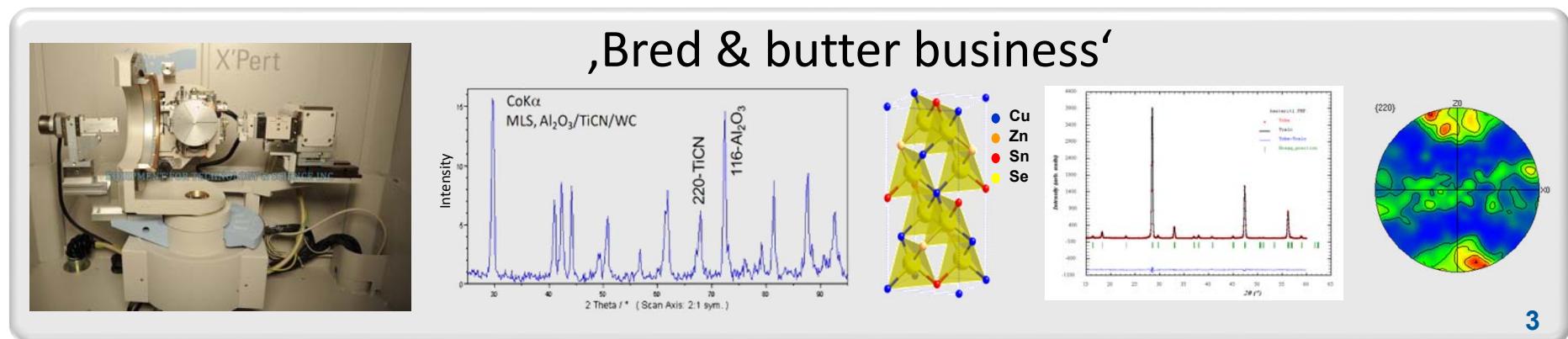
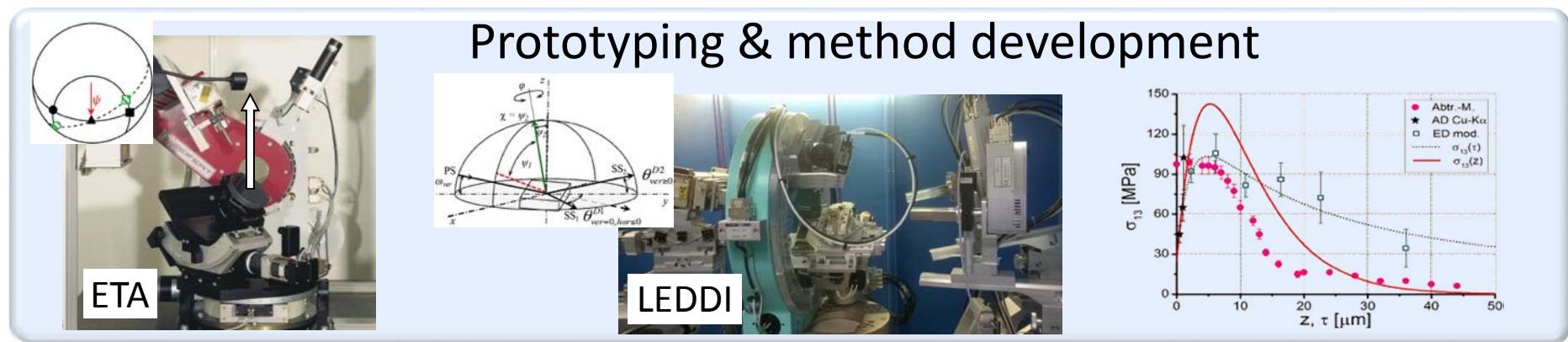
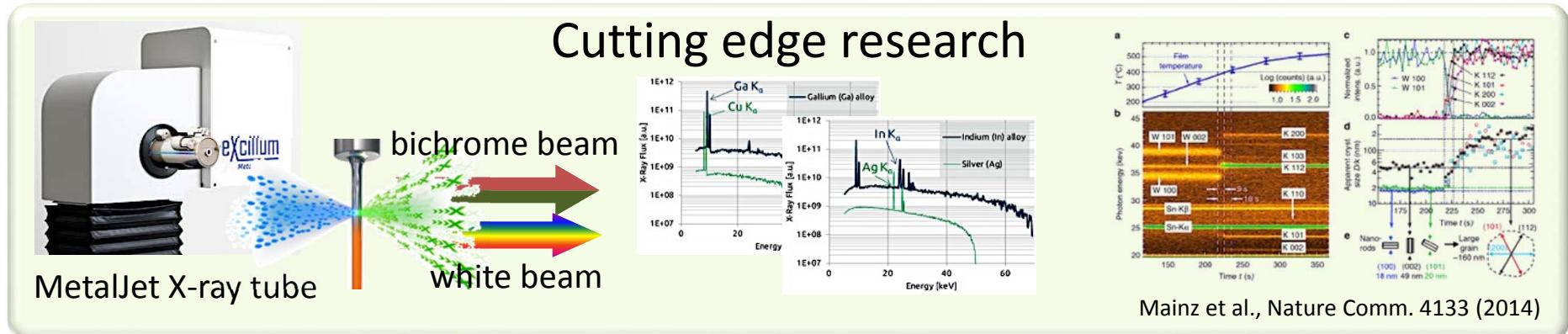
Susan Schorr, chair (EM-ASD)  
Christoph Genzel (EM-AME)  
Roel van de Krol (EE-IF)  
Bella Lake (EM-AQM)

LMC  
Michael Tovar  
(9 instruments)

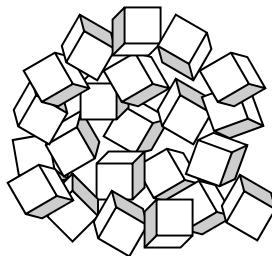


WCRC  
Christoph Genzel  
(3 instruments)

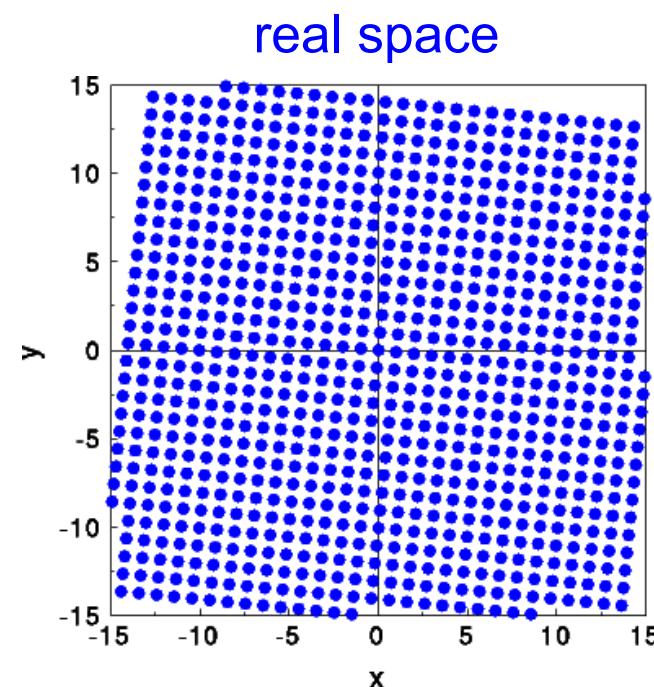
# The fundamental pillars of the X-ray corelab



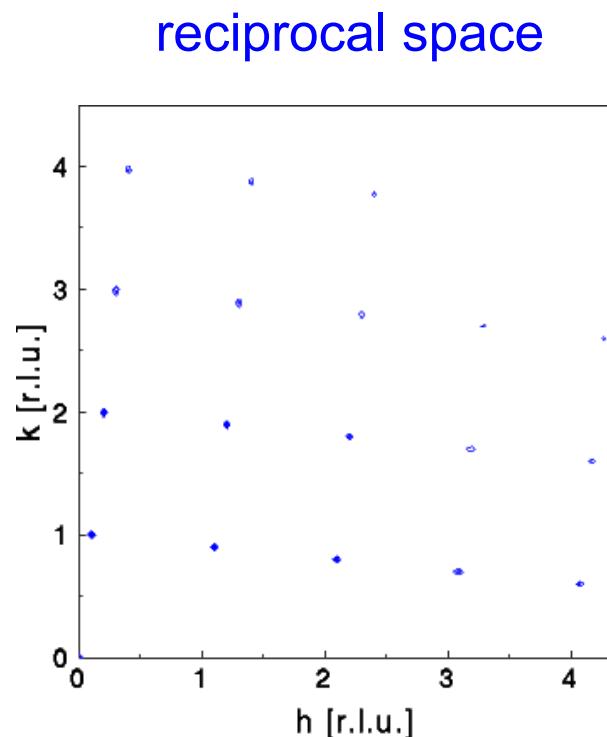
# Investigation of polycrystalline samples



What is a polycrystalline material?

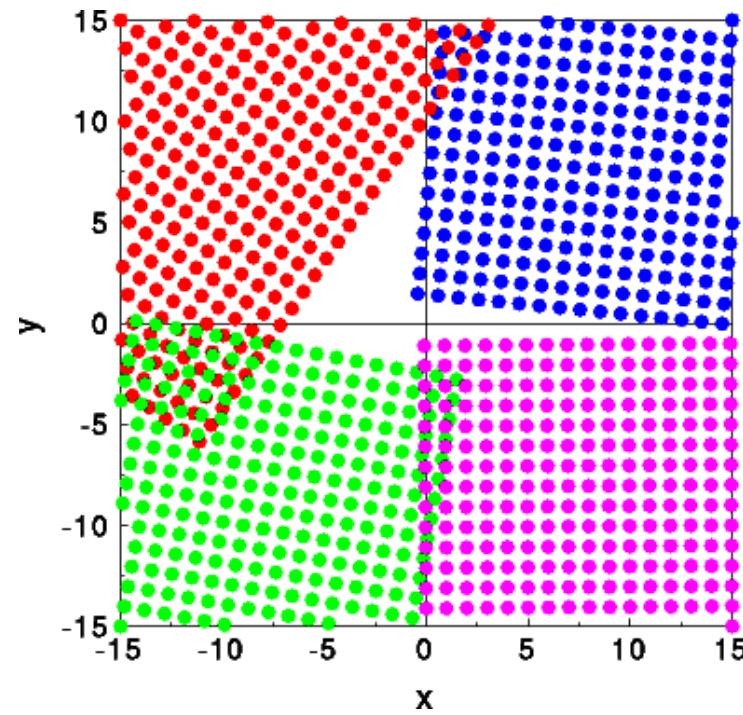


3D periodic arrangement  
of atoms/ions/molecules

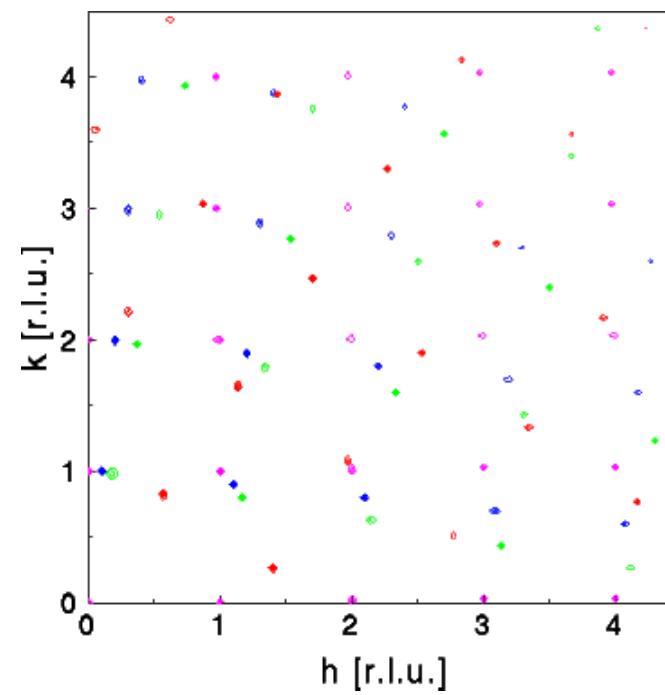


single crystal

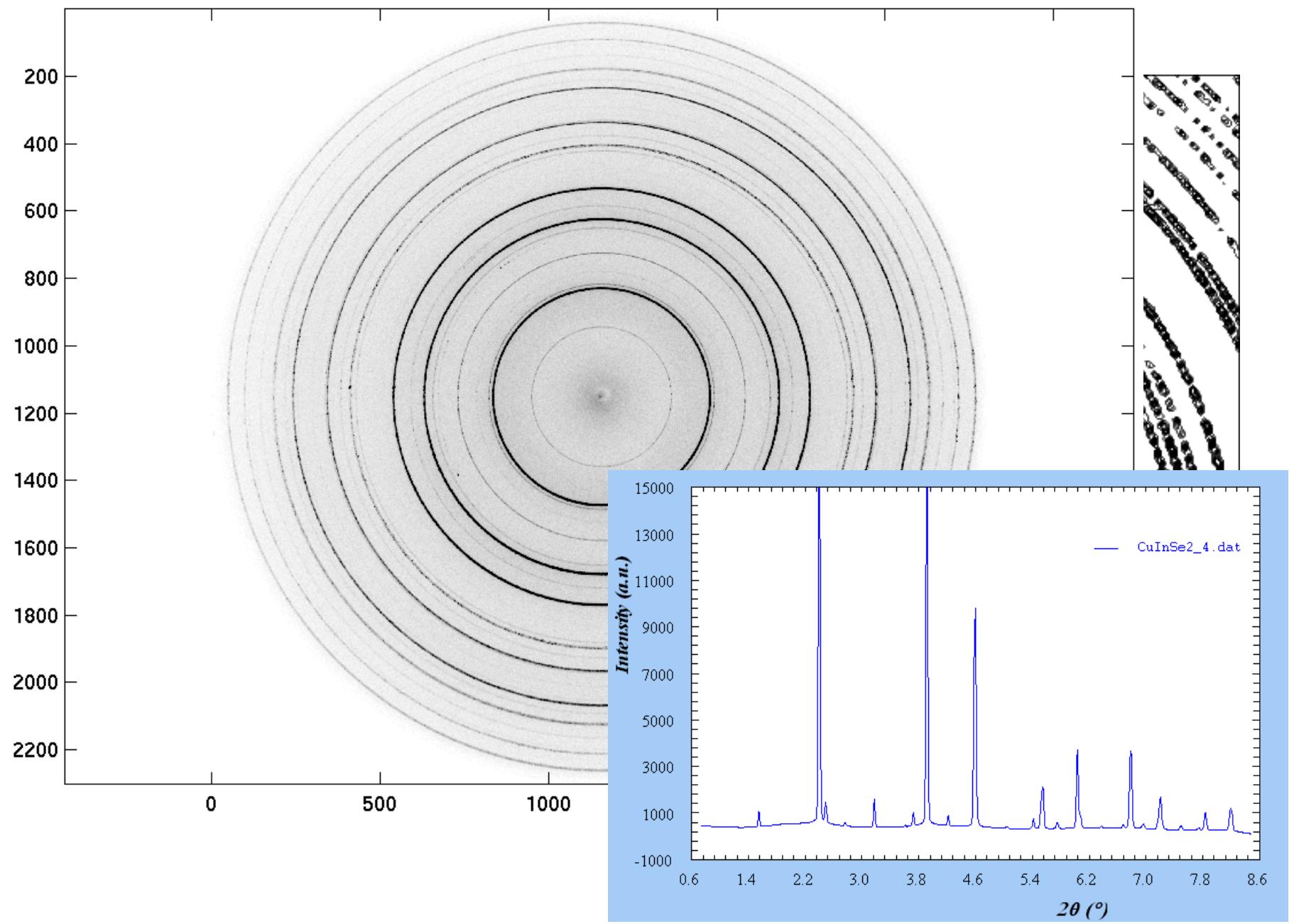
real space



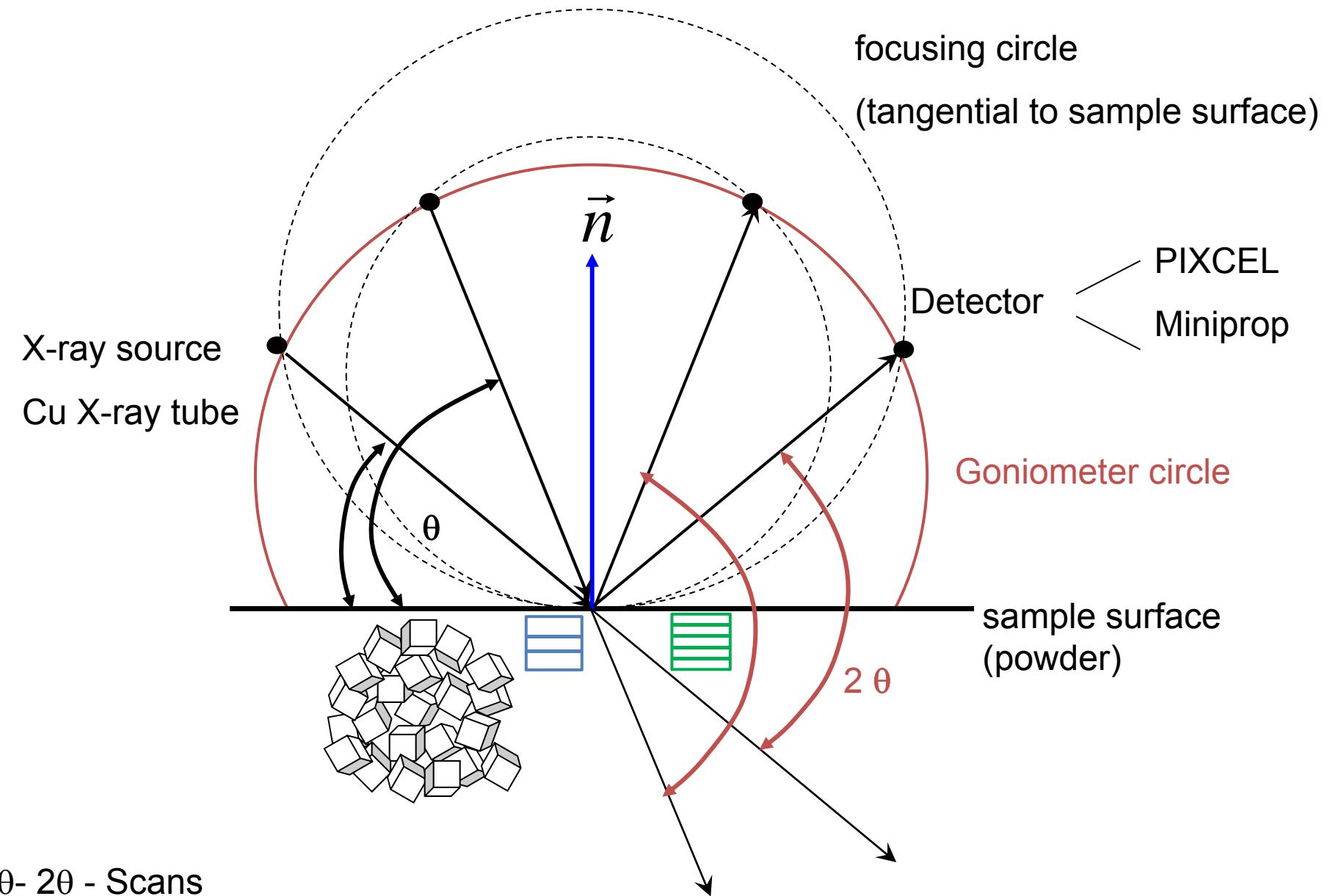
reciprocal space



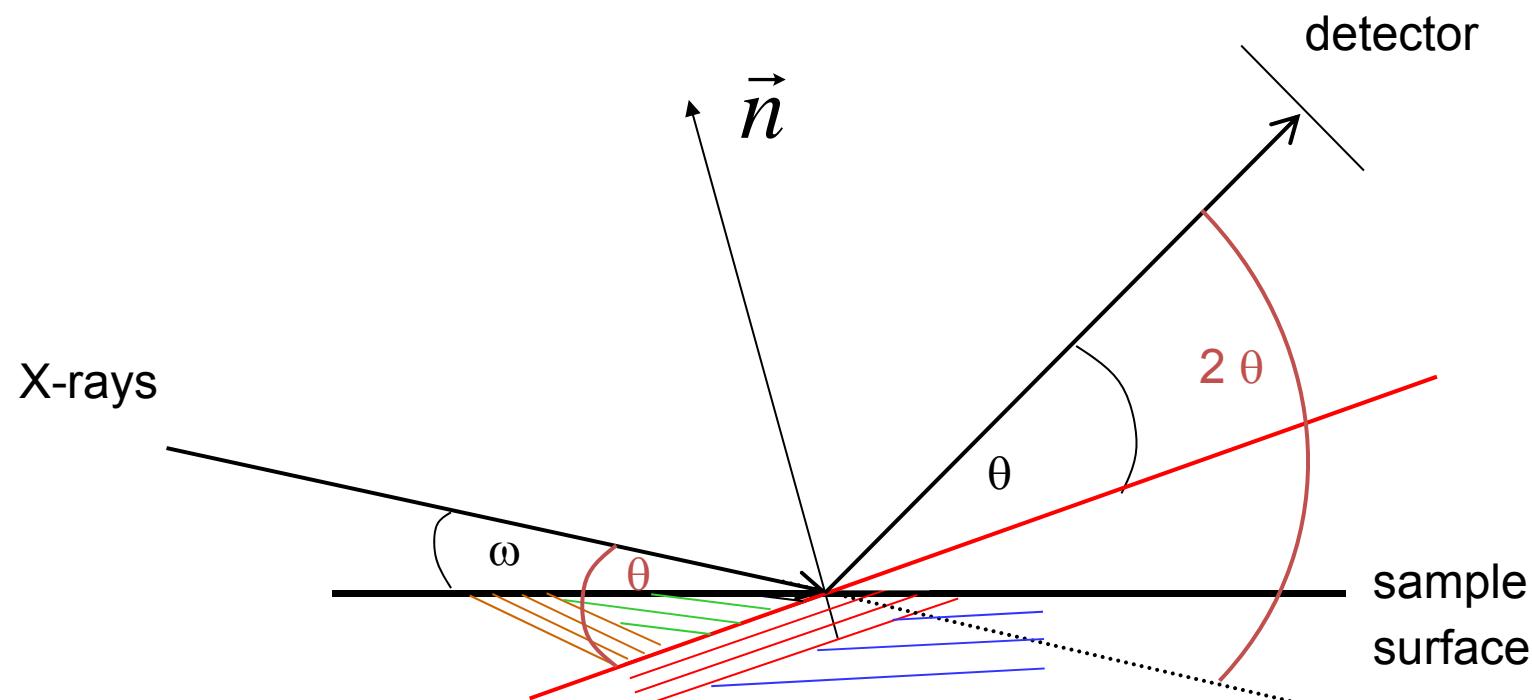
four single crystals



# Powder Diffraction (Bragg – Brentano – Geometry)



# grazing incidence X-ray diffraction - GIXRD



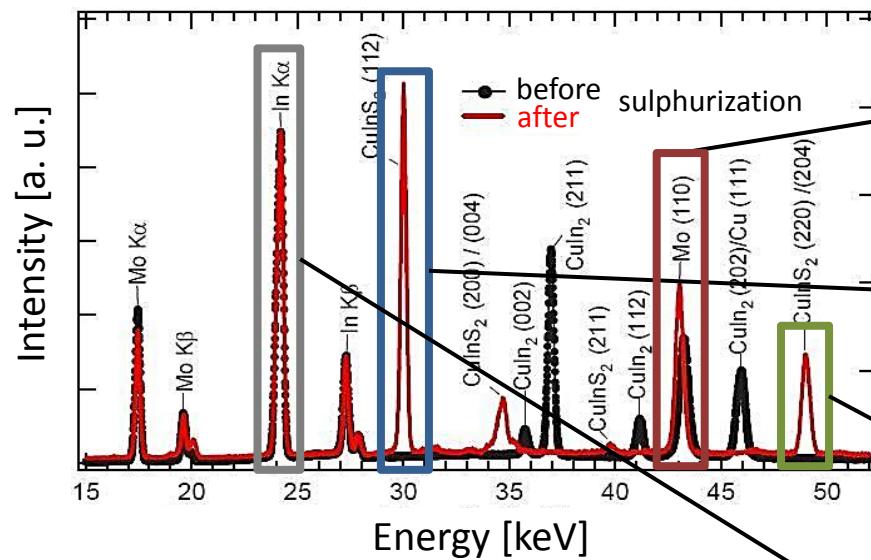
sample: polycrystalline thin film

- parallel beam
- fixed incidence angle  $\omega$
- detector - scan

# Information content of X-ray diffractograms

Features of diffraction methods:

- non-destructive
- phase selective
- information depth nm ... cm



## Line position and line shift:

- crystal structure
- residual stress

## Line width and line shape

- micro strain, defects

## Line intensity:

- crystallographic texture
- reaction kinetics

High energies >20 keV for:

- ... High information depth
- ... XRF close to K-edges of many elements

## Fluorescence lines:

- element distribution



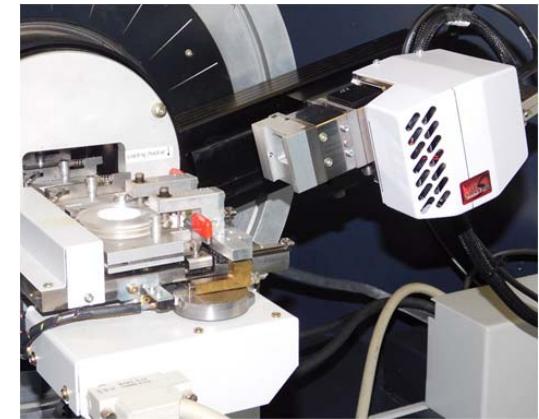
## Instrumentation @ LMC

Bruker D8 Advance for analysis of thin films (l.) and powders (r.)



## powder diffraction

- fast scans with LYNX Eye 1D detector (~5-10 min)
- sample changer for high throughput
- *in situ* high temperature sample environment
- *Bruker EVA* and *TOPAS* for phase analysis
- ICDD-PDF-2 for phase analysis,  
upgrade to PDF-4 in progress
- web access to FIZ-ICSD

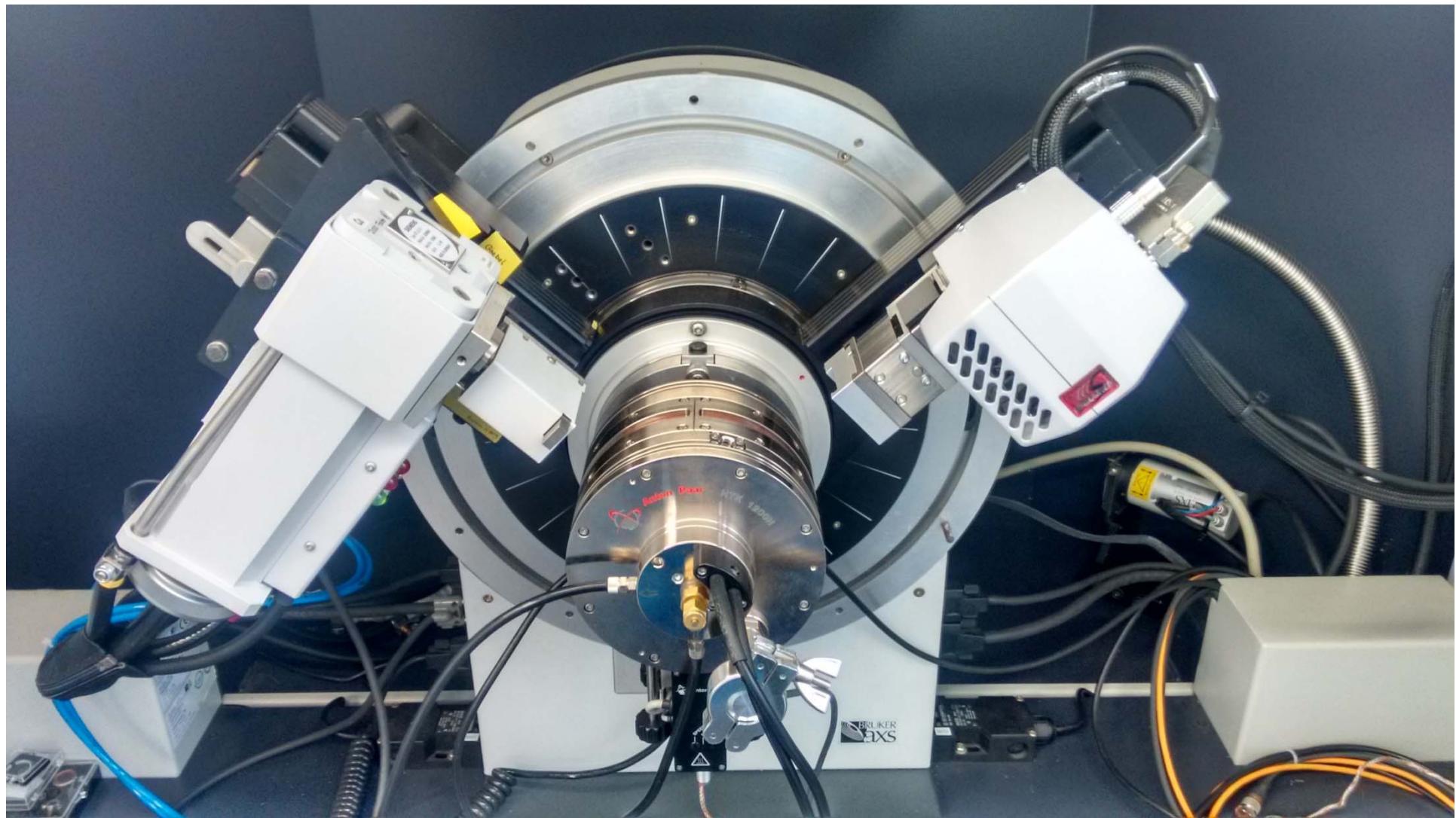


## What can be done?

- qualitative phase analysis (search/match with database)
- quantitative phase analysis (Rietveld refinement with e. g. TOPAS)
- single peak fits: lattice parameter (rectangular crystal system)  
peak width (FWHM)
- structure refinement
  - LeBail refinement (lattice parameter)
  - Rietveld refinement (all structural parameters)



# Anton Paar HTK 1200N High-Temperature Furnace-Chamber

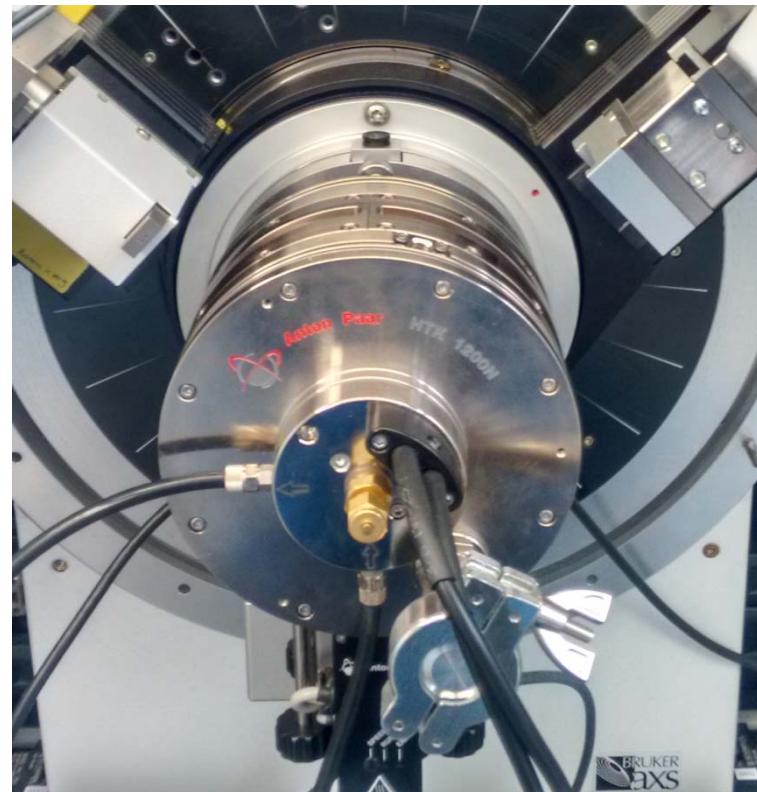




# Anton Paar HTK 1200N High-Temperature Furnace-Chamber

## Specifications:

- $RT \leq T \leq 1200^{\circ}\text{C}$ ,  $dT/dt \approx 1\text{ K s}^{-1}$
- oscillating sample holder for enhanced grain statistics
- motorized z-alignment stage to compensate for sample thicknesses and thermal expansion
- $p_{\min} = 10^{-4}$  mbar, air and inert gas atmosphere\*
- sample carriers for powders and thin films ( $\emptyset_{\max} = 20$  mm)
- X-ray window: graphite/Kapton (10 mm width)

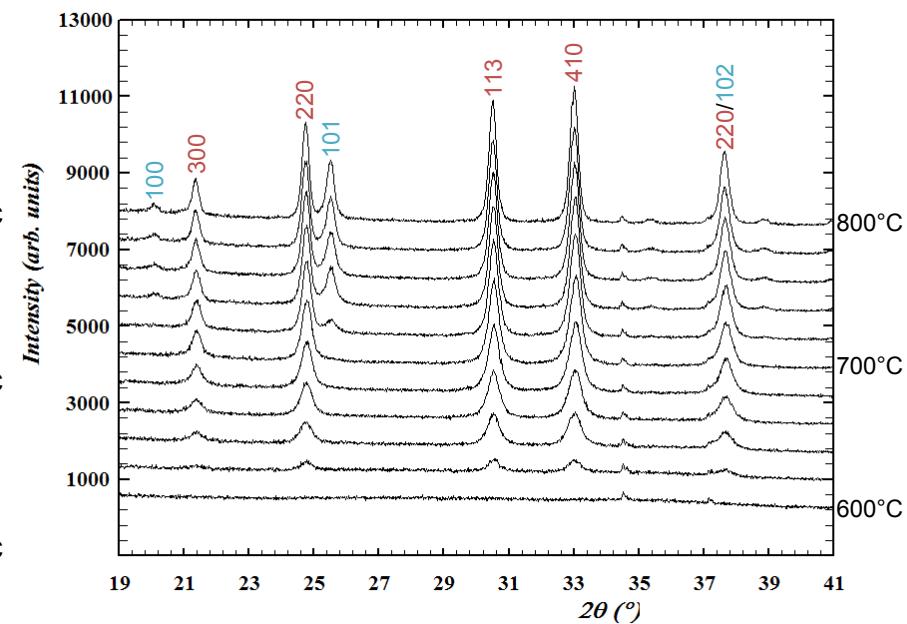
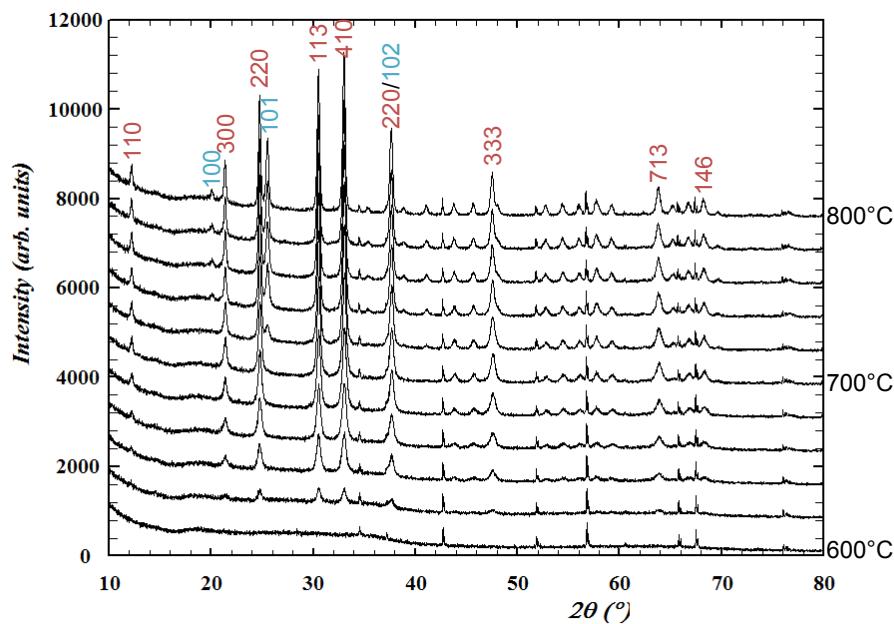


\* vacuum (scroll pump,  $\approx 10^{-4}$  bar) and Inert gas atmosphere (e.g. N<sub>2</sub> or user supplied gas mixtures) available

# *in situ* temperature-dependent diffraction

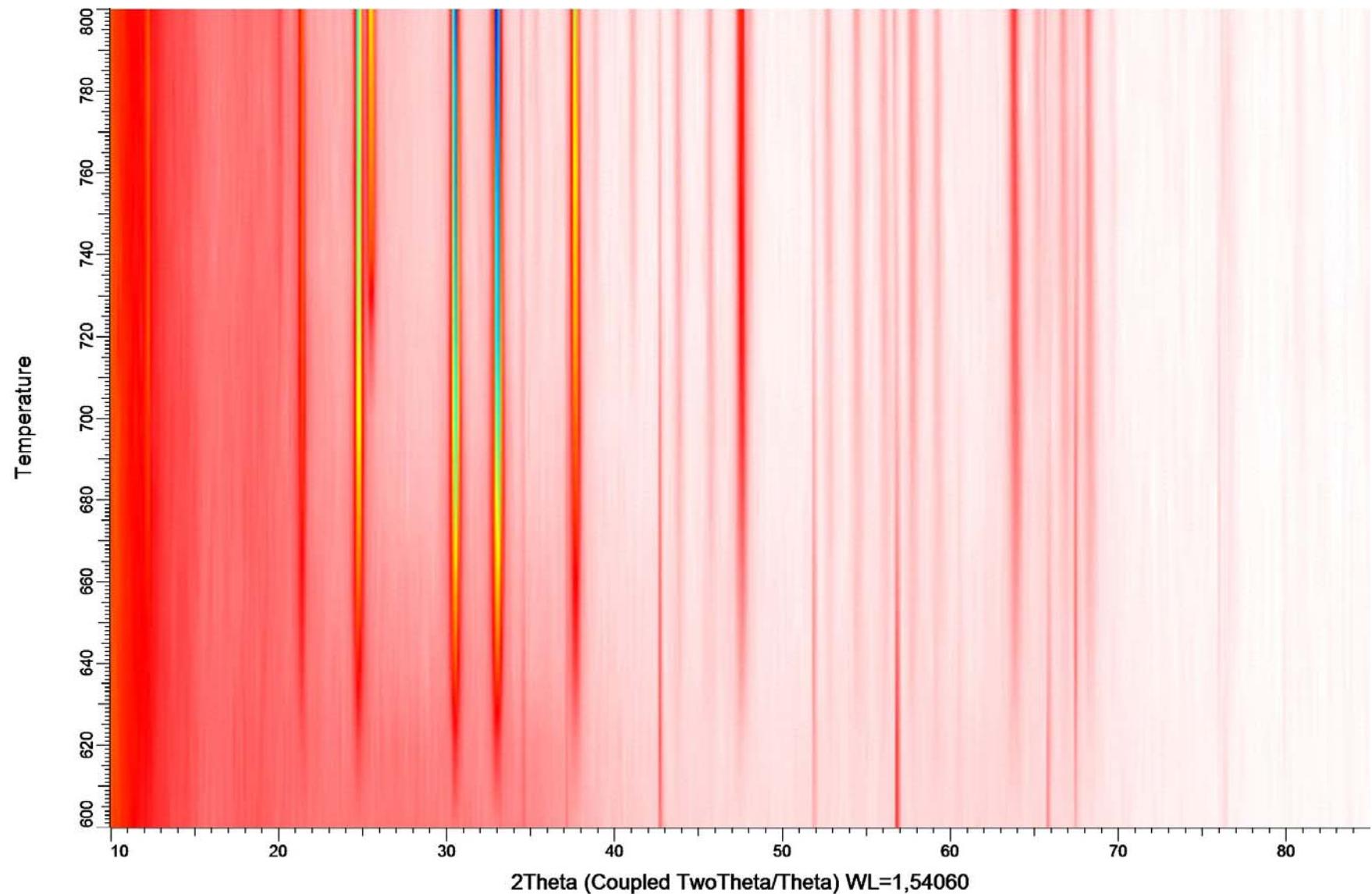
## monitoring of crystallization of $\text{ZnGe}_2\text{O}_4$

- *in situ* XRD measurements performed on Bruker D8 equipped with Anton Paar HTK1200N
- $2\theta$  range =  $10^\circ$  –  $80^\circ$
- temperature range =  $600$  –  $800^\circ\text{C}$ ;  $20\text{ K}$  steps;  $dT/dt \approx 1\text{ K s}^{-1}$ ;  $20\text{ min}$  delay before measurements
- air atmosphere
- phases:  $\text{ZnGe}_2\text{O}_4$ ,  $\text{GeO}_2$  ( $\alpha$ -quartz-type)



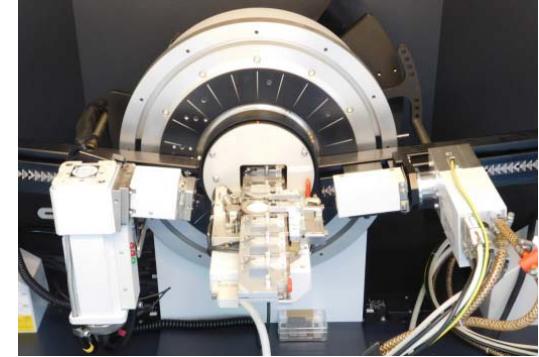
# *in situ* temperature-dependent diffraction

## monitoring of crystallization of $\text{ZnGe}_2\text{O}_4$



## grazing incidence diffraction (GIXRD)

- low background energy dispersive SOL-X detector
- sample changer for high throughput
- *Bruker EVA* for phase analysis
- sample height cannot be adjusted (necessary for pattern refinement)



## What can be done?

- qualitative phase analysis (search/match with database)
- single peak fits: lattice parameter (rectangular crystal system)  
peak width (FWHM)



## Instrumentation @ LMC

PANalytical MRD (l.) and MPD (r.) for analysis of thin film and powders, for texture and epitaxy analysis and micro-diffraction





## PANalytical MPD (multi purpose diffractometer)

- precise **GIXRD** measurements of thin films: parallel X-ray beam (X-ray mirror and Xe single counter), sample height can be adjusted (z-scan)
- sample table for x-y scans allows scanning
- **reflectivity** option for film thickness and roughness
- **fast powder diffraction** with 1-D PIXcel detector



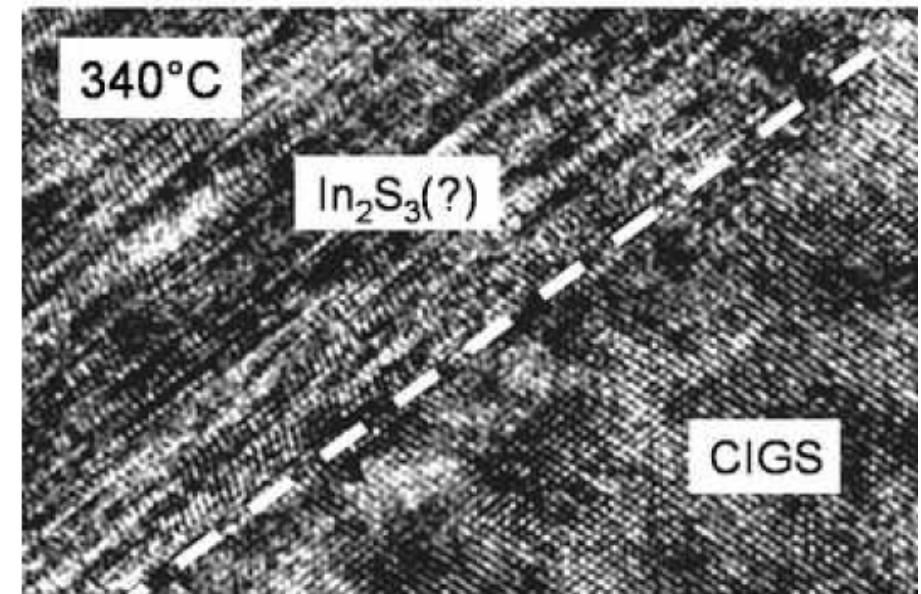
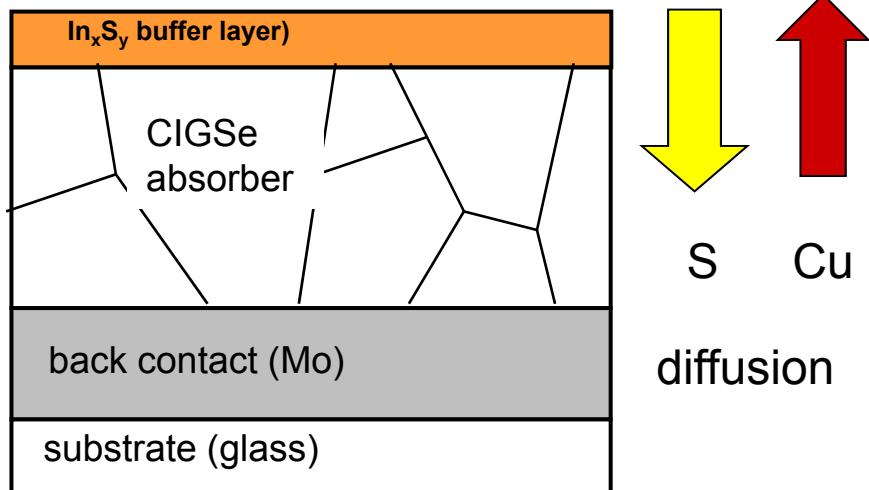
## What can be done?

- qualitative phase analysis (search/match with database)
- quantitative phase analysis (Rietveld refinement with e. g. HIGHSCORE)
- single peak fits: lattice parameter (rectangular crystal system)  
peak width (FWHM)
- structure refinement
  - LeBail refinement (lattice parameter)
  - Rietveld refinement (all structural parameters)

With both XRD and  
GIXRD  
measurements!

## Example: Depth-resolved phase analysis (qualitative)

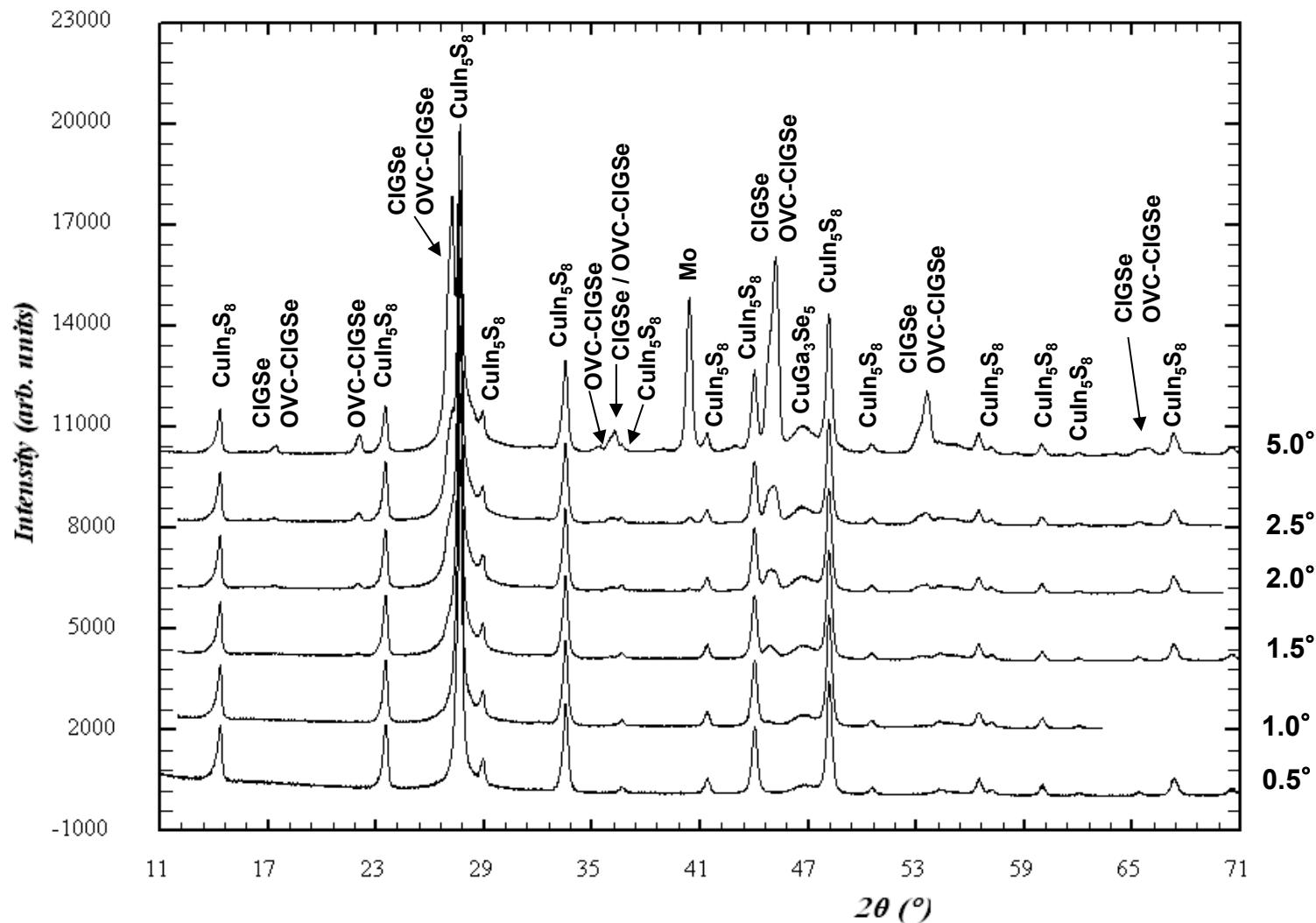
sputtered  $\text{In}_x\text{S}_y$  layer  
 $T_{\text{sub}} = 230^\circ\text{C}, 340^\circ\text{C}$ , no heating



What happened with the sputtered  $\text{In}_x\text{S}_y$  layer?

# $\text{In}_x\text{S}_y$ / CIGSe ( $T_{\text{sub}} = 340^\circ \text{ C}$ )

diffusion of Cu and Ga from CIGSe into the buffer ( $\text{In}_x\text{S}_y$ ) → formation of vacancy compounds



# $\text{In}_x\text{S}_y / \text{CIGSe}$ ( $T_{\text{sub}} = 340^\circ \text{ C}$ )

→ layer stacking:



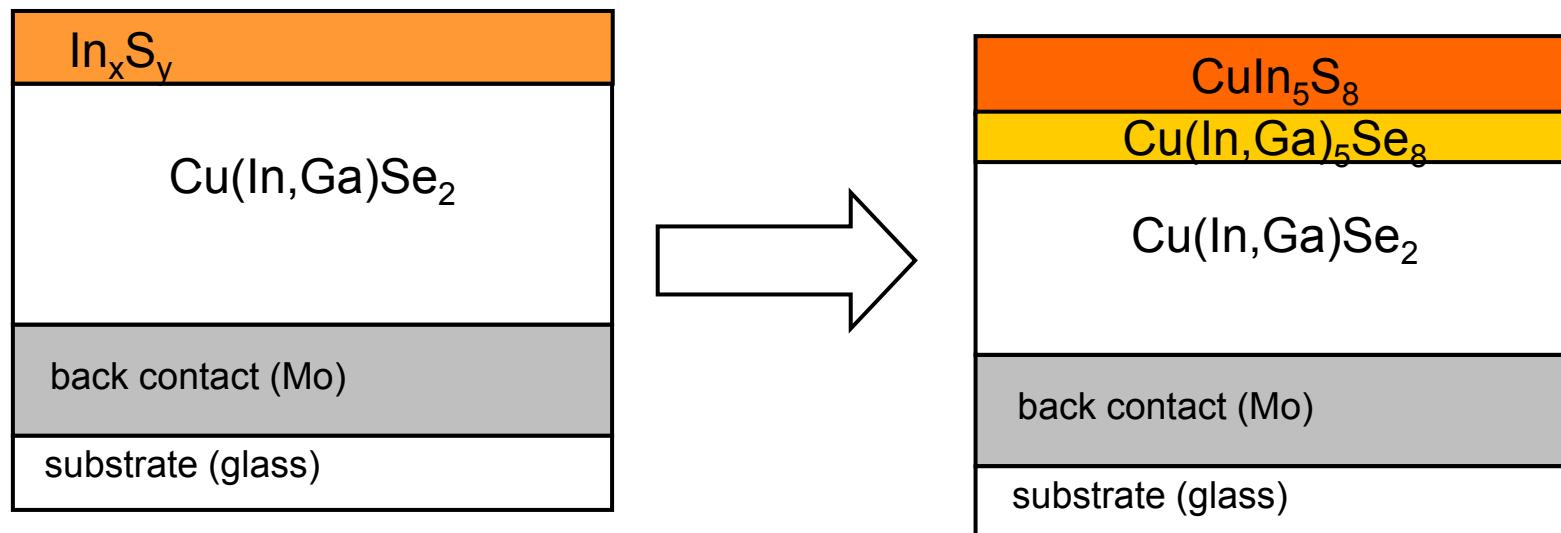
additional:



thin layer

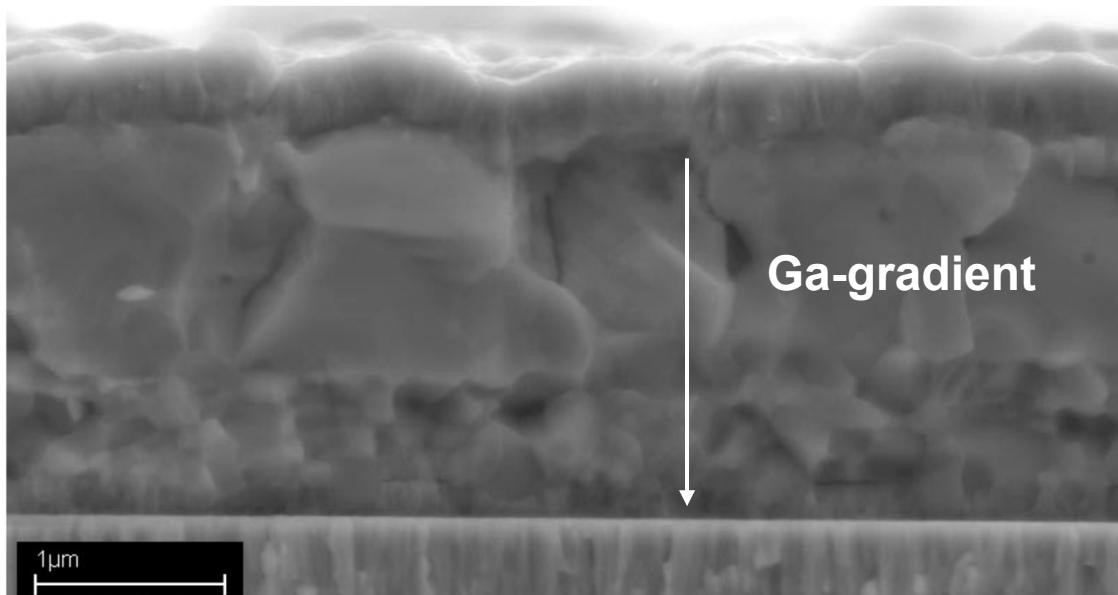
or

small crystallites



## Example: depth-resolved phase analysis (semi-quantitative)

### Ga-gradient in Cu(In,Ga)Se<sub>2</sub> absorber layers



cross-sectional SEM image

ZnO window layer

CdS

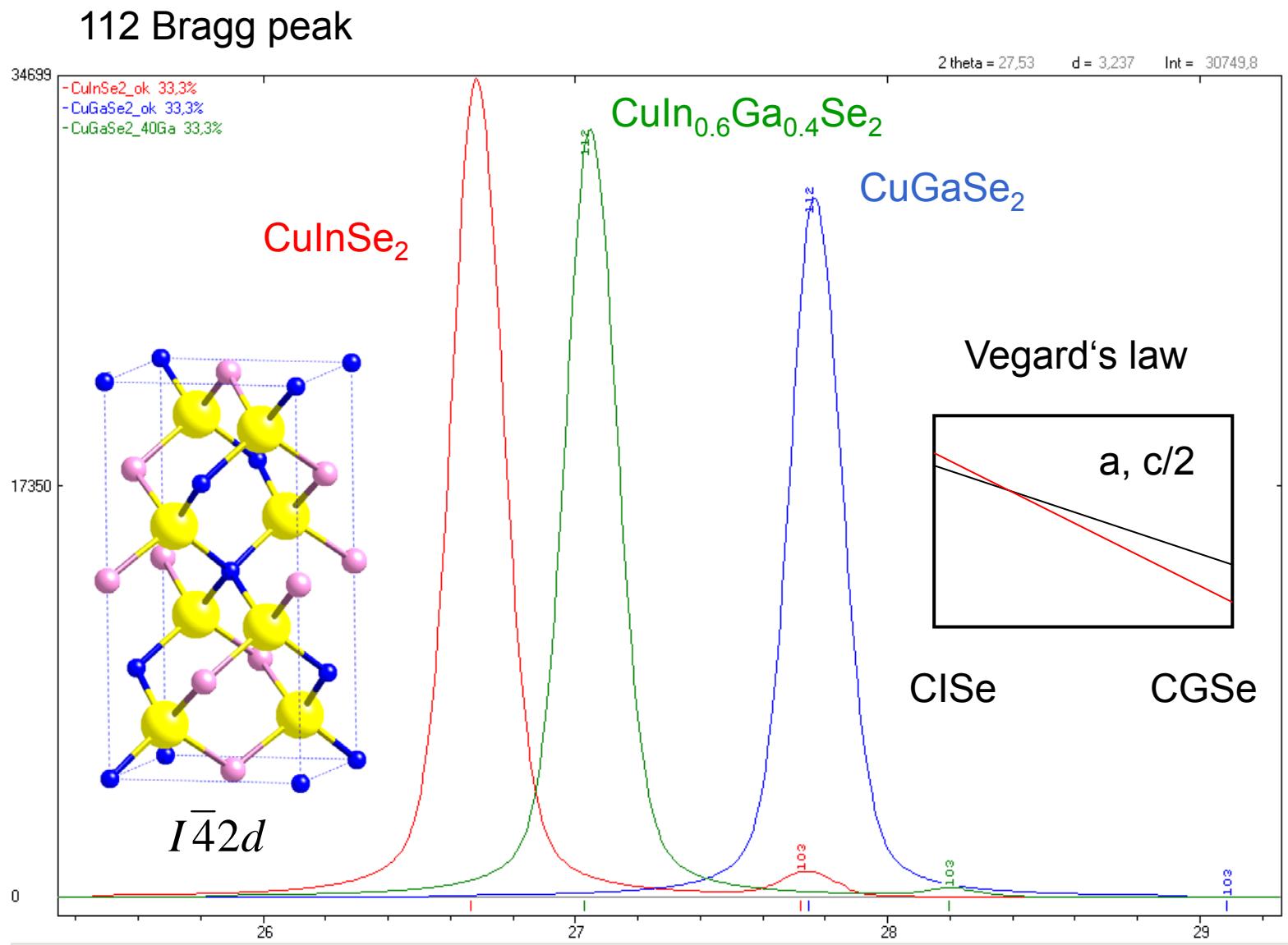
CIGSe absorber

Mo back contact

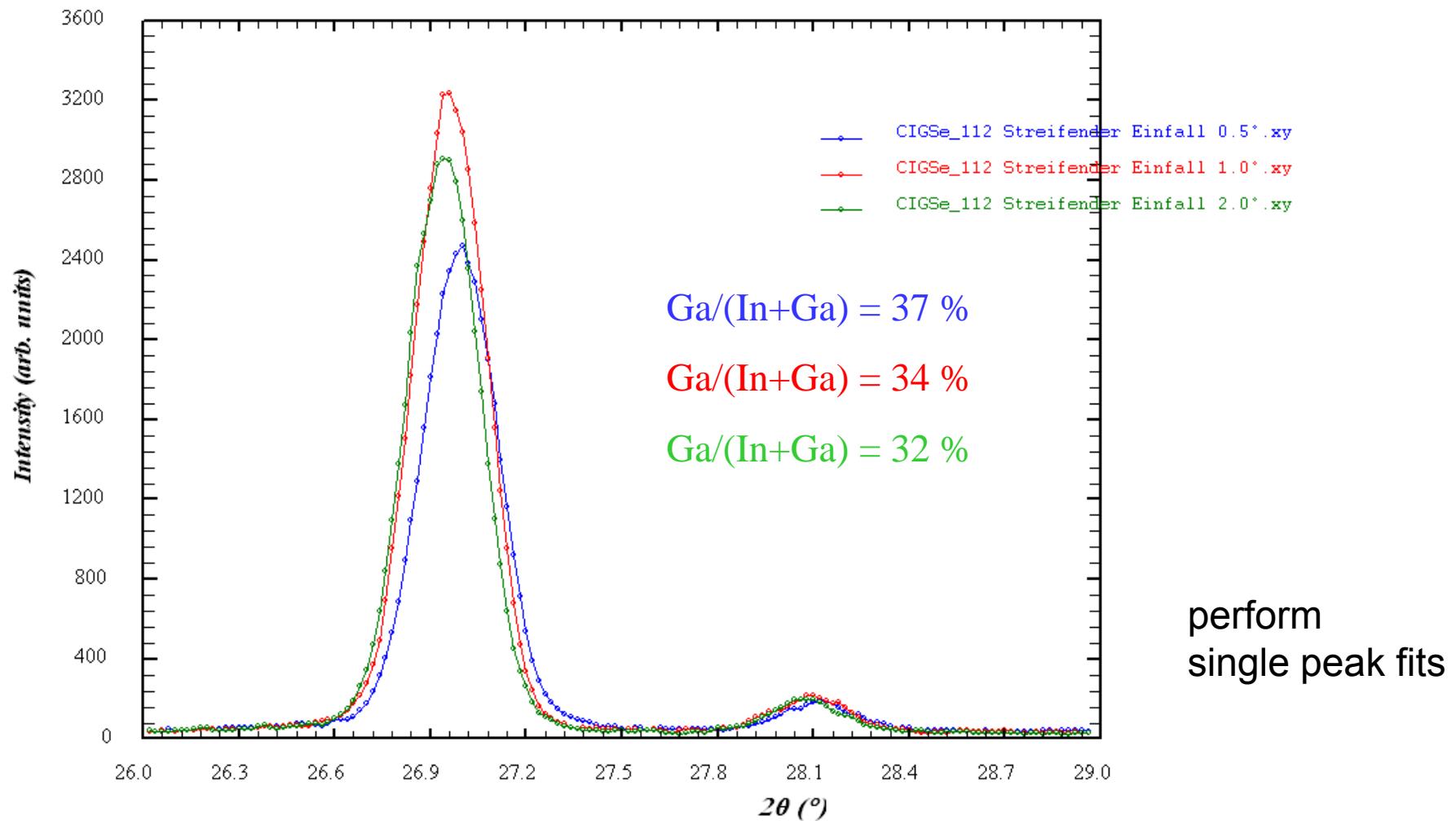
glass substrate

Ch. A. Kaufmann, R. Caballero, T. Unold, R. Hesse S. Schorr, M. Nichterwitz,  
H.-W. Schock, Sol. Energy Mat. & Sol. Cells (2008)

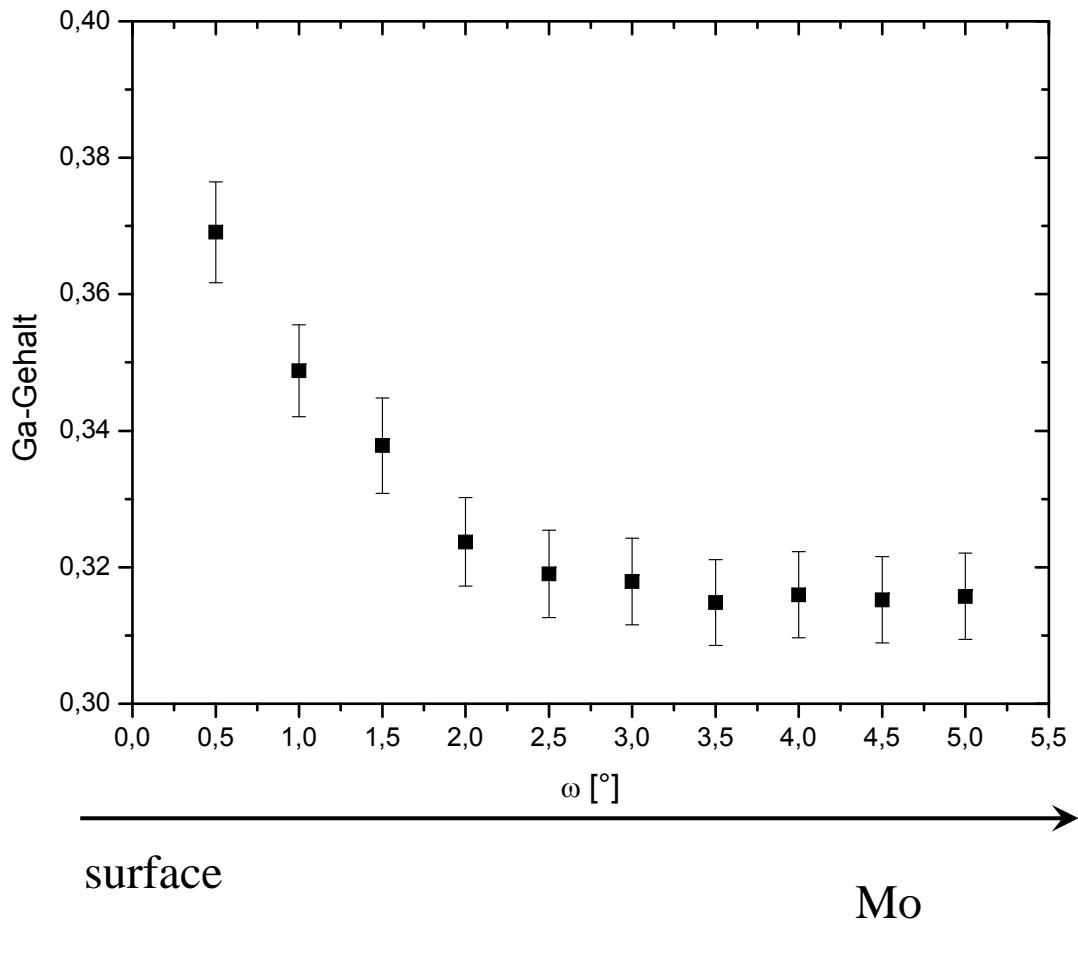
# Simulated powder pattern: Cu(In,Ga)Se<sub>2</sub>



# depth profiles of CIGSe thin films



# depth profiles of CIGSe thin films



Bragg equation:

$$\lambda = 2d_{hkl} \sin \theta$$

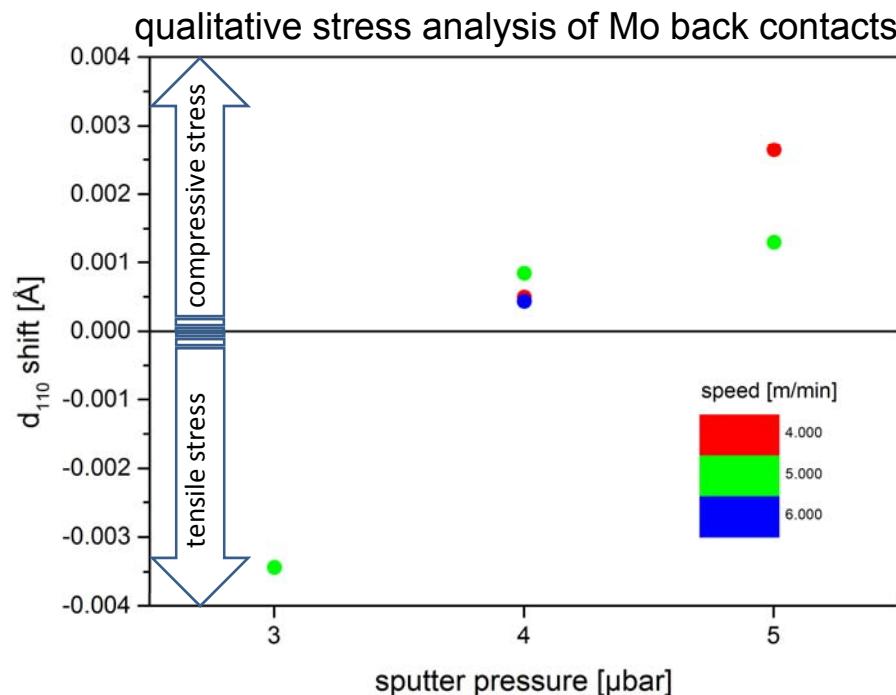
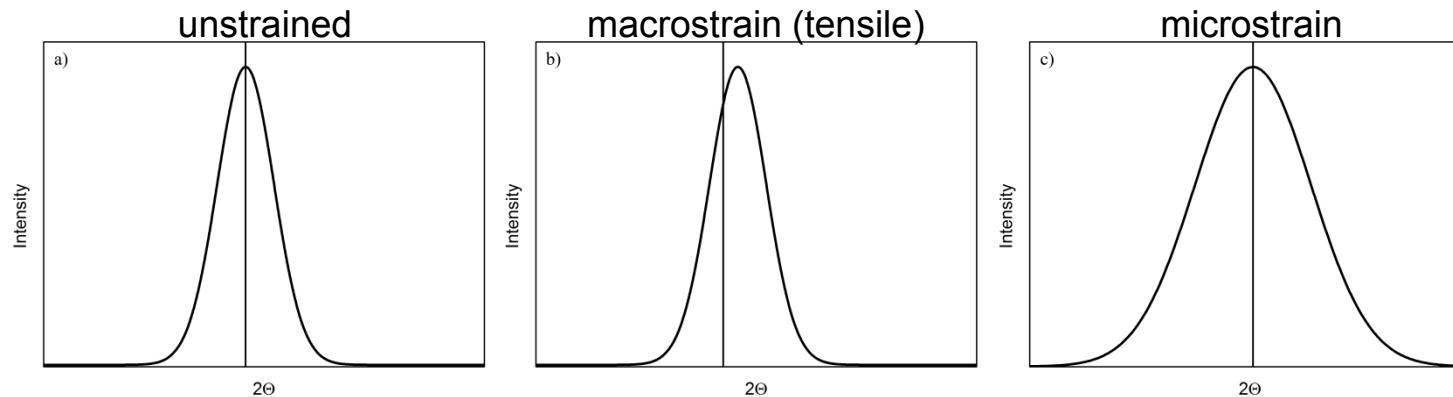
tetragonal crystal system:

$$\frac{1}{d_{hkl}^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2}$$

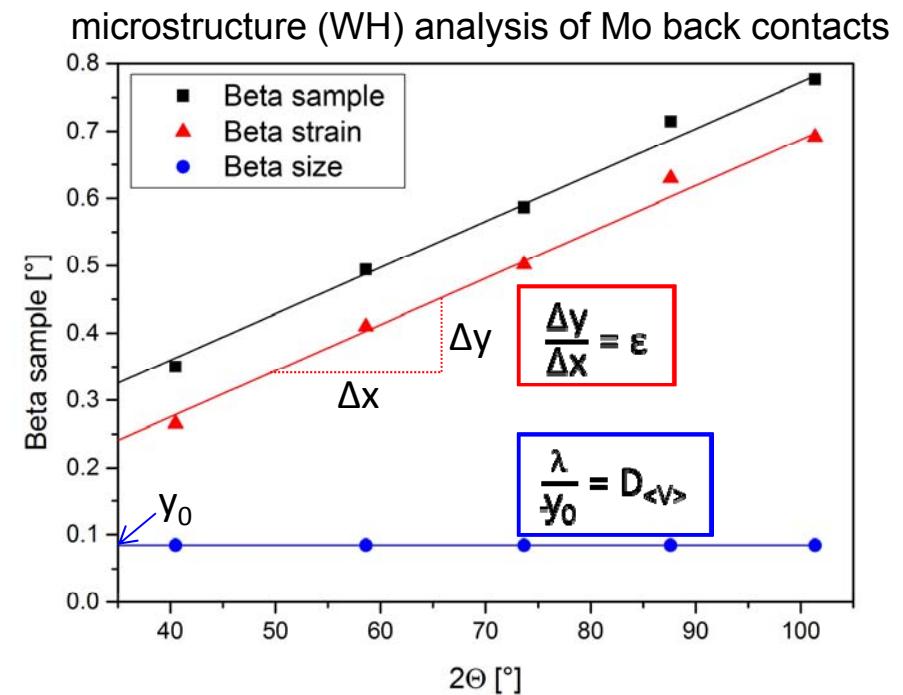
Vegard's law:

$$a_{AB} = a_A(1 - x_B) + a_B x_B$$

# macrostrain and microstrain in thin films



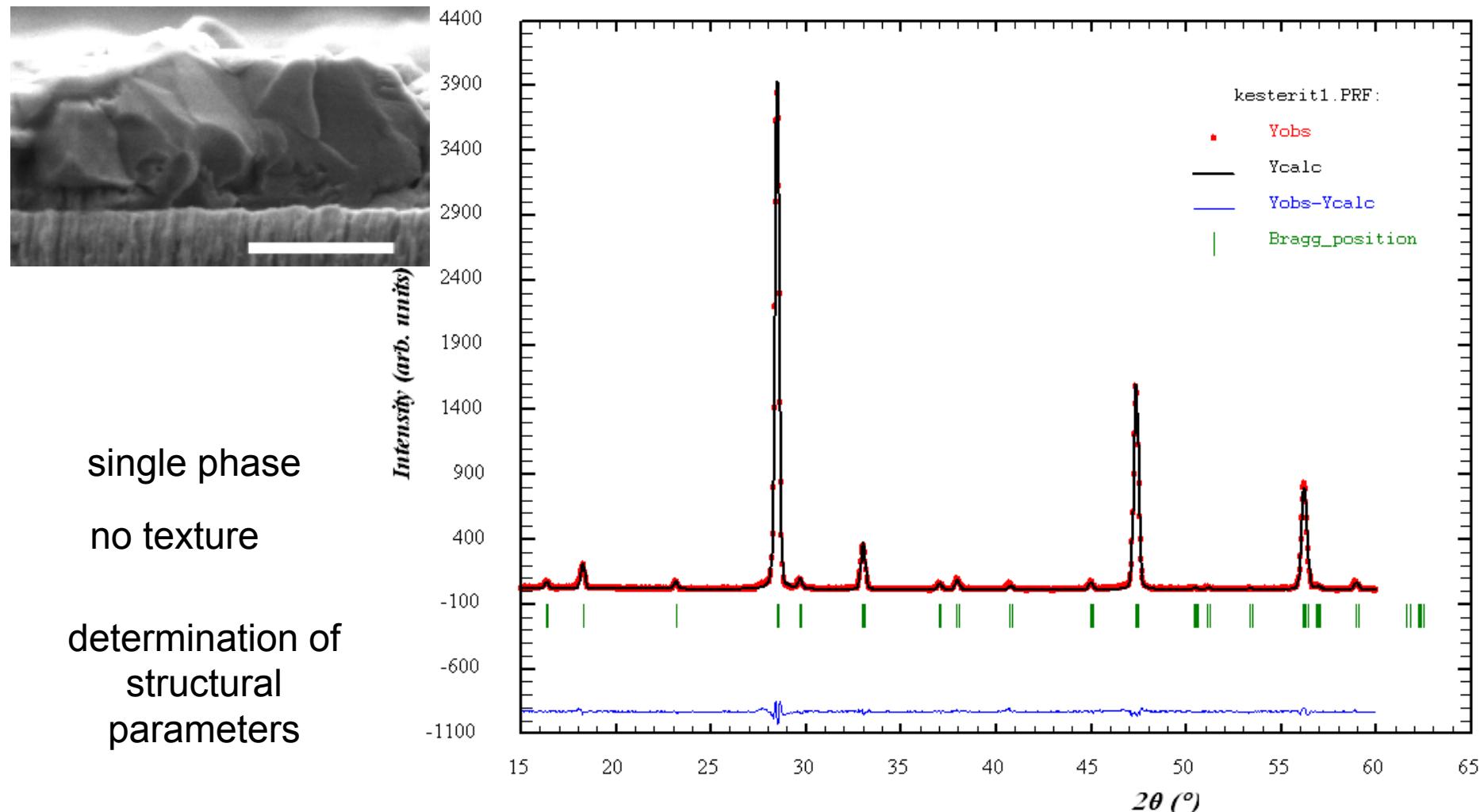
→ shift in peak position reveals stress regime



→ based on individual peak profile parameters  
 → separation of size and strain broadening

# Rietveld refinement of GIXRD data

$\text{Cu}_2\text{ZnSnS}_4$  thin film grown by co-evaporation

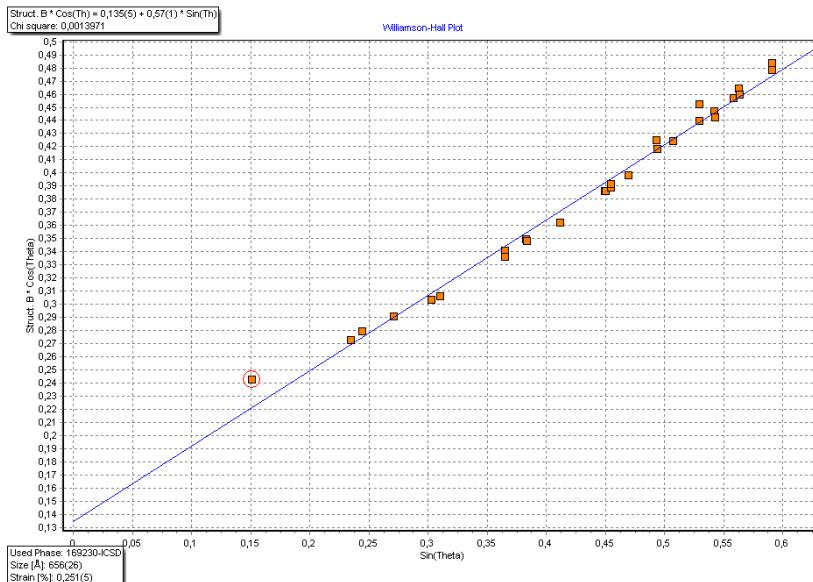


B. A. Schubert, B. Marsen, S. Cinque, T. Unold, R. Klenk, S. Schorr, H.-W. Schock, Progress in Photovoltaics: Research and Application (2010)

# microstructure analysis of thin films

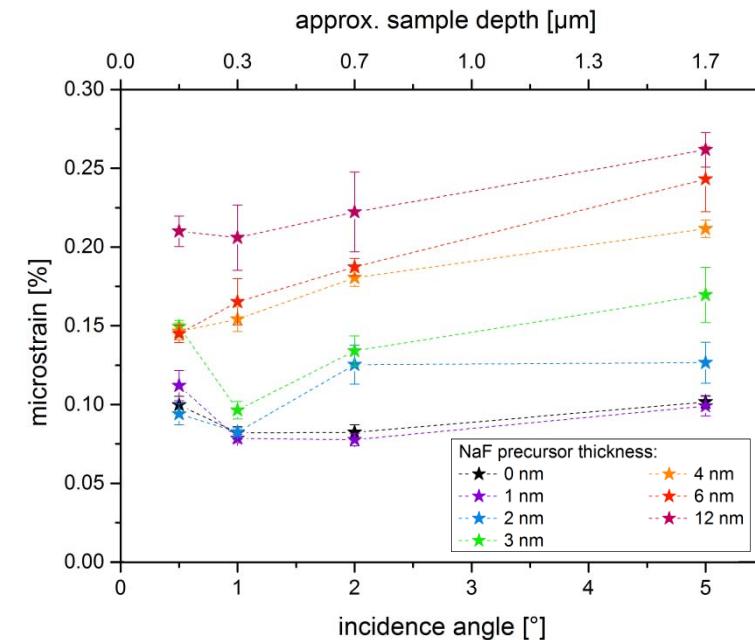
- microstructure analysis of CISe thin film absorber layers
- broadening of integral peak breadths  $\beta$  used to deduce microstrain and domain size
- depth-resolved characterization possible by varying incidence angles

Williamson-Hall analysis of CISe using pseudo-Voigt profile function to obtain  $\beta$   
(done with Highscore Plus, PANalytical)



→ linear equation used to obtain size and strain

microstructure analysis of CISe using Thompson-Cox-Hastings pseudo-Voigt profile function to obtain  $\beta$   
(done with Fullprof Suite software package)



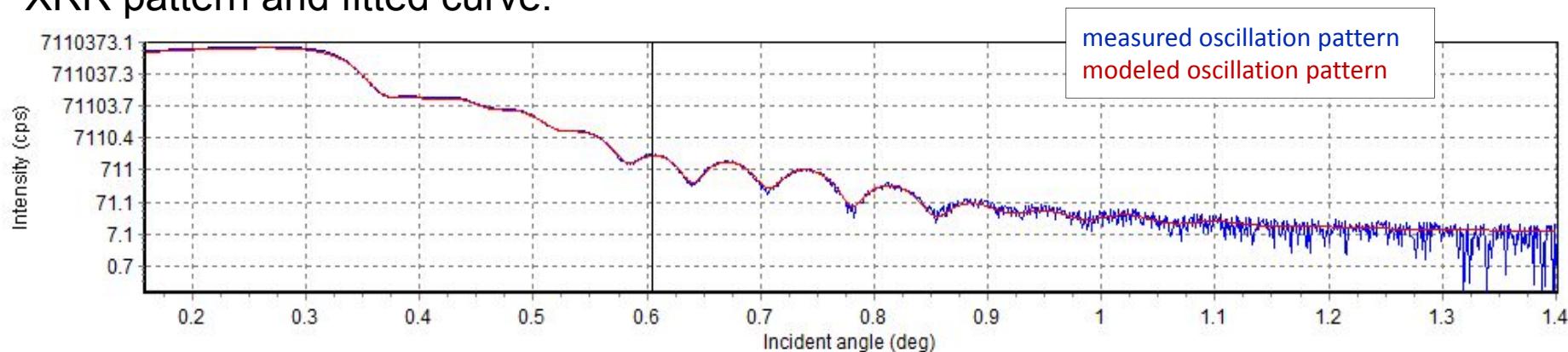
→ size and strain directly calculated from corresponding profile parameters

# X-ray reflectivity (XRR)

estimation of thickness, density and roughness of thin films

- X-ray reflectometry is based on varying reflectivities of X-rays when traversing interfaces between dissimilar media (differing optical constants)
- resulting interference fringes allow modeling of thickness, density and roughness of thin layers
- higher contrasts in optical constants (for multi-layer systems) cause stronger oscillations
- layer thickness is inferred by the period of the oscillations

XRR pattern and fitted curve:



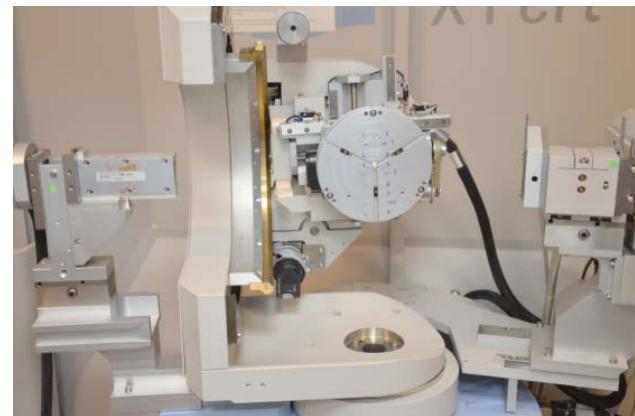
Results from modeling:

Layer	Layer Description	Density (g/cm <sup>3</sup> )	Thickness (nm)	Roughness (nm)
2, 0	DensityOnly, Fe <sub>3</sub> O <sub>4</sub>	5.18	12.845	2.173
1, 0	DensityOnly, CoO	6.45	44.143	1.862
Substrate	DensityOnly, SrTiO <sub>3</sub>	5.1	600000	0.993

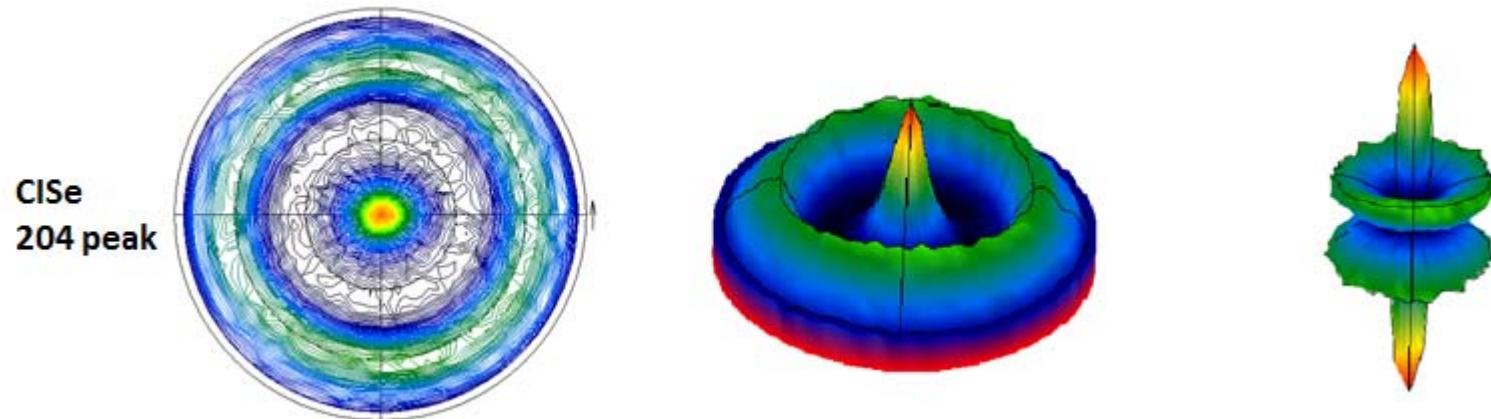
# Instrumentation @ LMC

## PANalytical MRD for texture analysis:

- X-ray lens for high intensive parallel beam
- Eulerian cradle for 3D sample orientation
- Xe single counter
- *X'Pert Texture* to create pole figures and orientation distribution functions (ODF)



Pole figures recorded on  $\text{CuInSe}_2$  (CISe) chalcopyrite-type thin film absorber layer

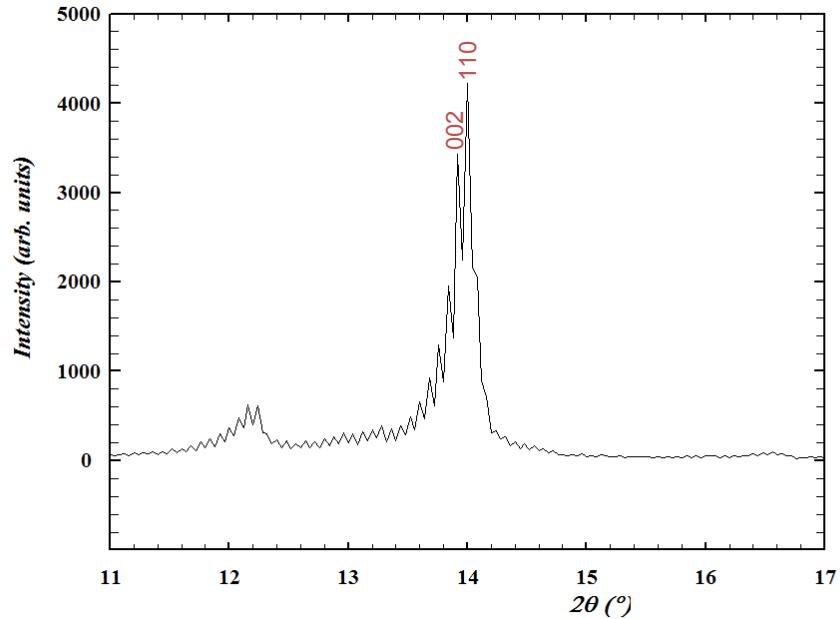
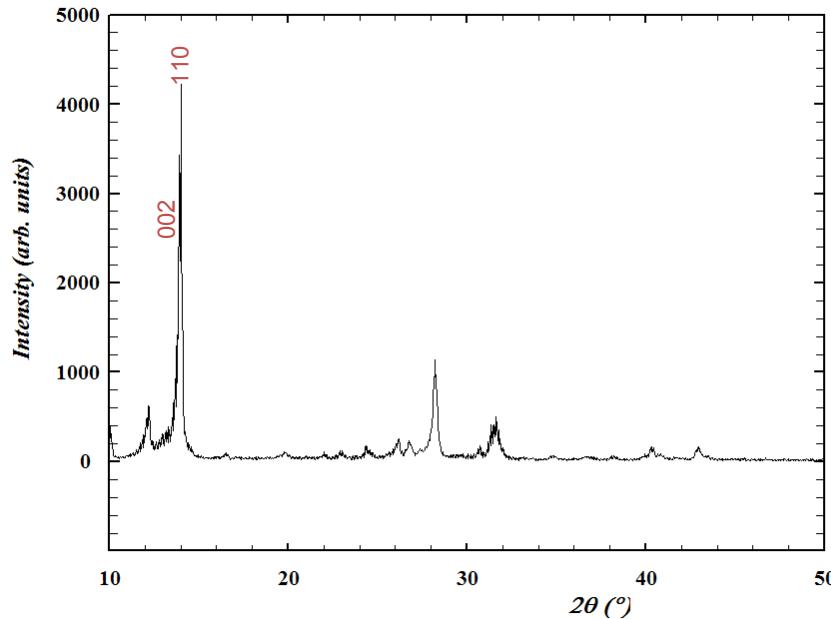


Orientation distribution functions (ODF) shown as contour plot and 3D plot for CISe Bragg peaks 112 and 204



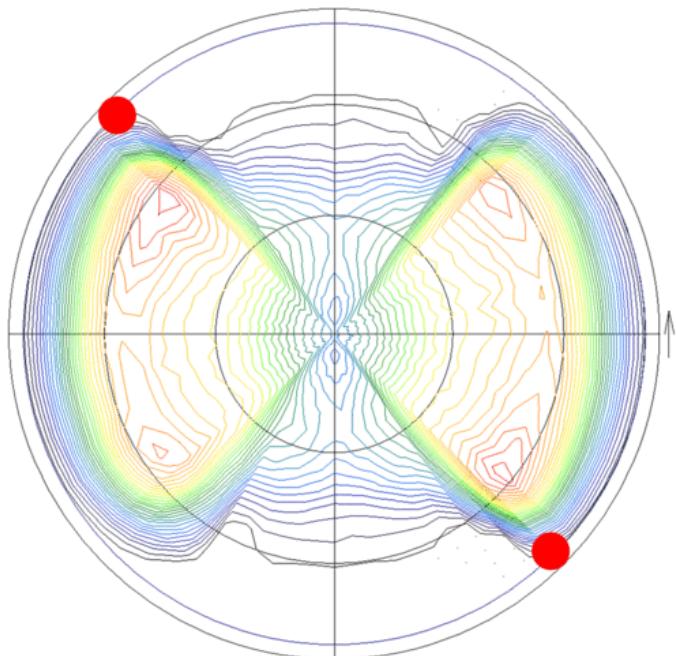
# texture of thin films

hybrid perovskite MAFACsPb(I<sub>x</sub>Br<sub>1-x</sub>)<sub>3</sub> on glass substrate (C. Rehermann)



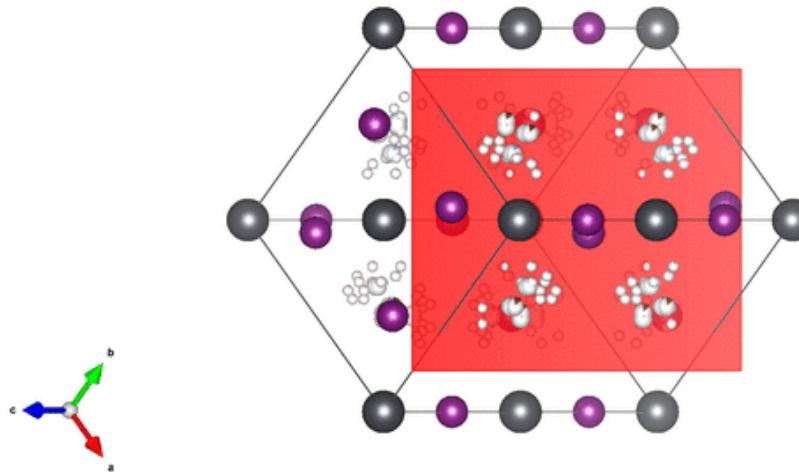
- accidental sample rotation during GIXRD measurement
- „sawtooth“ pattern due to highly textured thin film

# texture of thin films



110 pole figure @  $2\theta = 14.0756^\circ$

→ nearly epitaxial thin film



### Panalytical MRD for epitaxy analysis and micro-diffraction

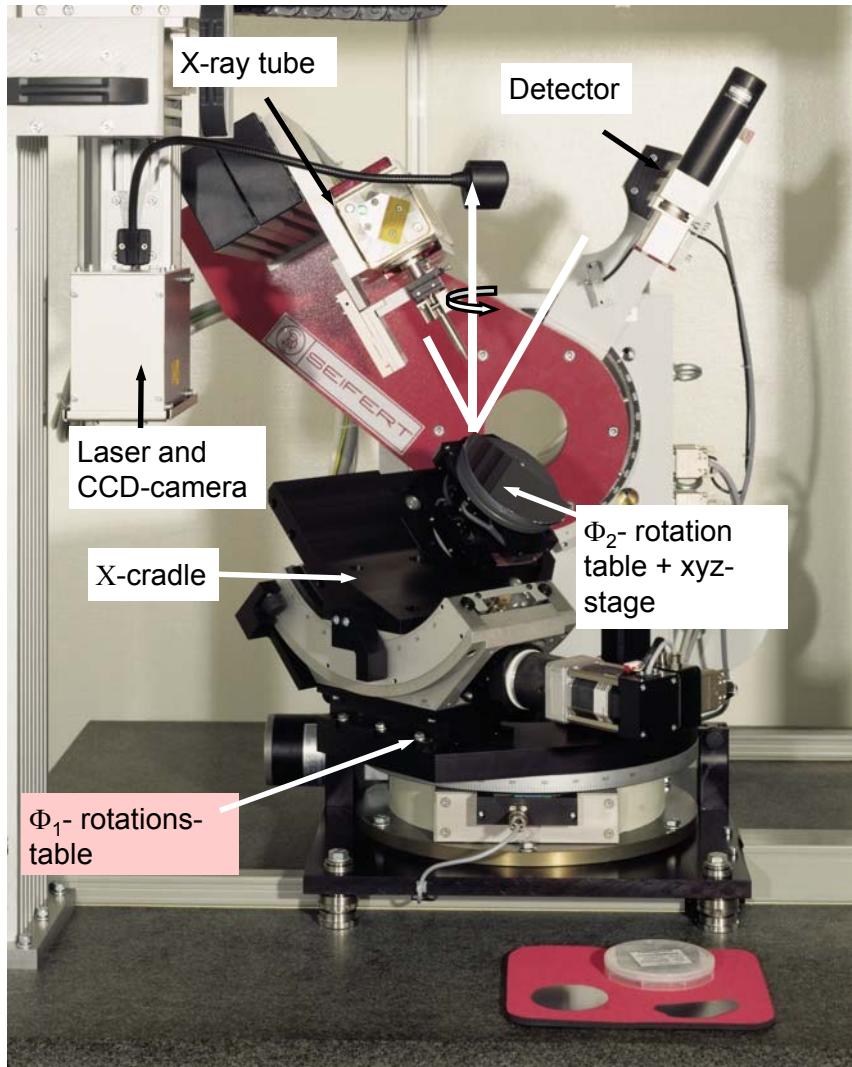


#### Monocapillary 230 x 0,1

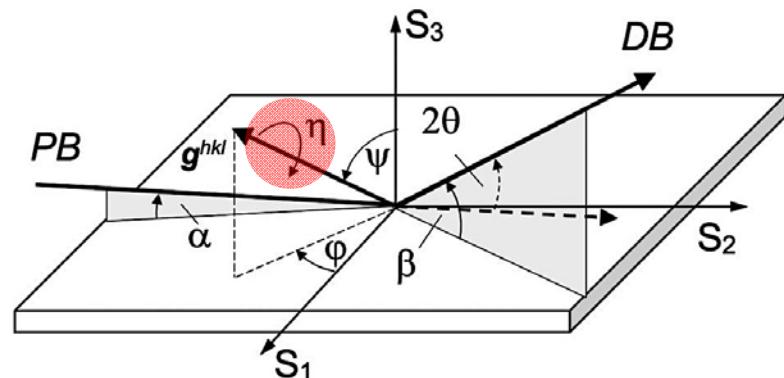
- length 230 mm
- thickness 0,1 mm
- divergence 0,3°

# Instrumentation @ WCRC

## The 5-axes diffractometer ETA for surface gradient analysis



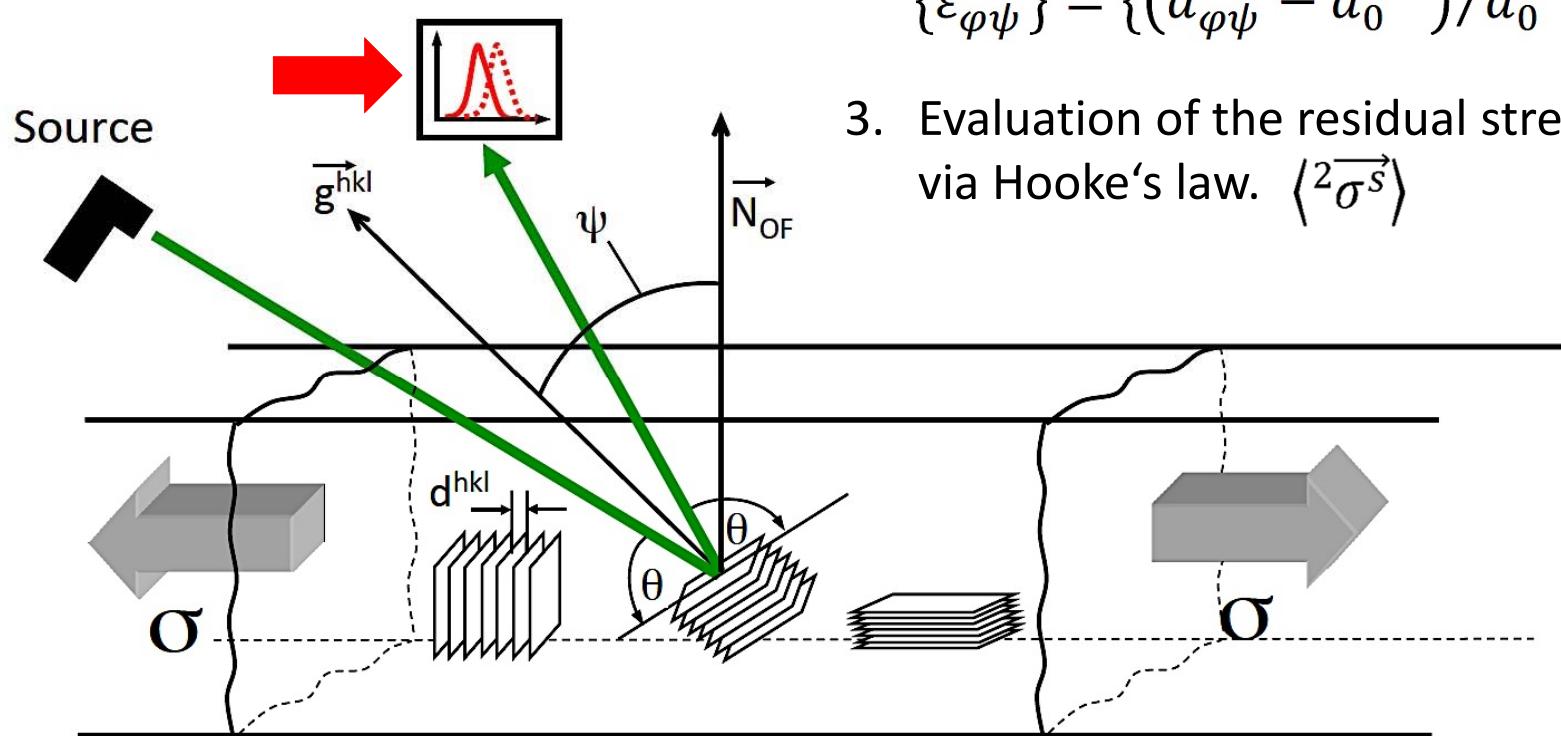
- direct sample rotation around the scattering vector
- polycapillary optics and soller + secondary monochromator for thin film analysis



# Principle of residual stress analysis by diffraction methods

$\downarrow$   

$$\{\varepsilon_{\varphi\psi}^{hkl}\} = \left\{ \begin{array}{l} -\cot\theta^{hkl} \Delta\theta^{hkl} \\ -\Delta E^{hkl} / E^{hkl} \end{array} \right\} \rightarrow \langle \overline{\sigma^s} \rangle$$
  
angle-/energy-dispersive
 $\uparrow$



1. Measurement of the diffraction line shift for various orientations  $(\varphi, \psi)$

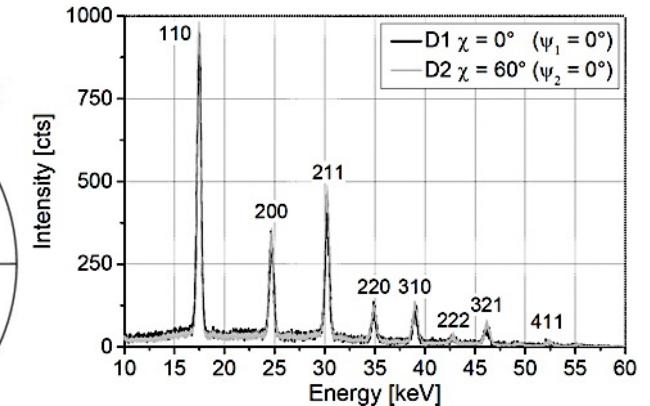
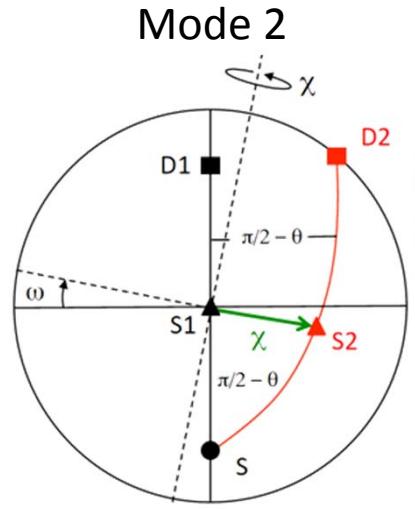
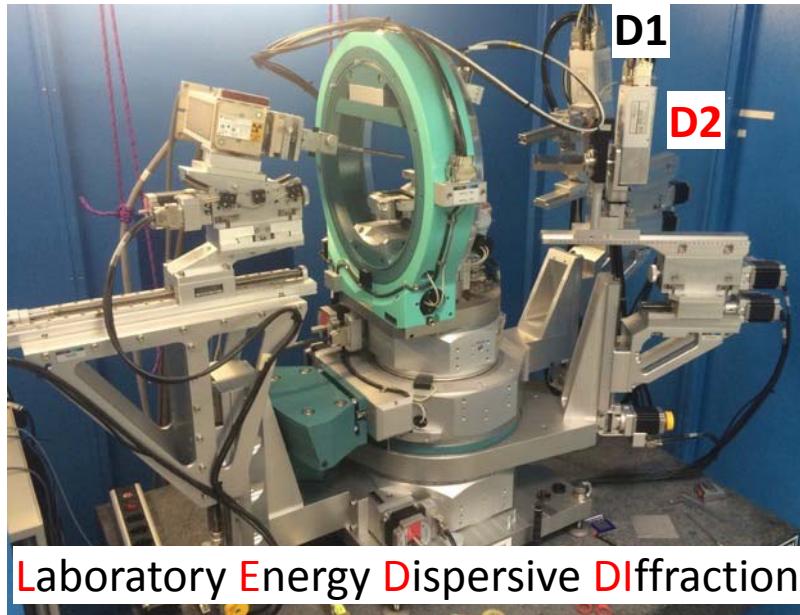
2. Evaluation of the lattice strain

$$\{\varepsilon_{\varphi\psi}^{hkl}\} = \{(d_{\varphi\psi}^{hkl} - d_0^{hkl})/d_0^{hkl}\}$$

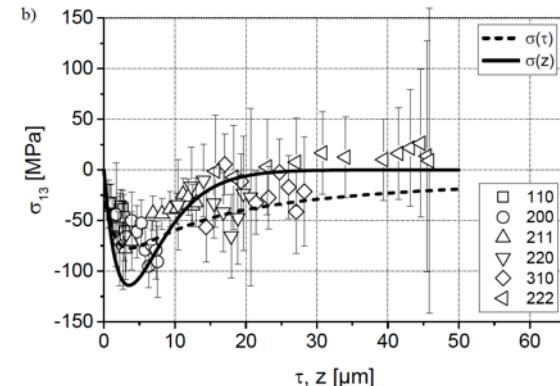
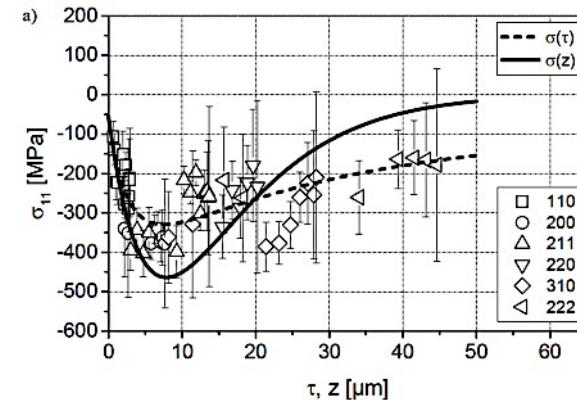
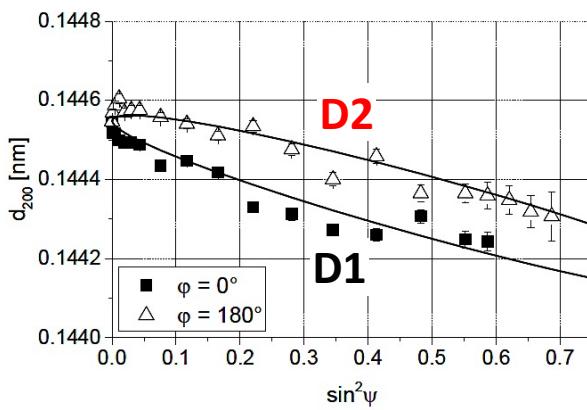
3. Evaluation of the residual stress tensor via Hooke's law.  $\langle \overline{\sigma^s} \rangle$

# Instrumentation @ WCRC

## The energy-dispersive 8-circle diffractometer LEDDI



simultaneous data acquisition with two detectors



in- and out-of-plane residual stress depth profiles from a single  $\chi$ -scan

# Full support : Planning and conducting experiments & data evaluation

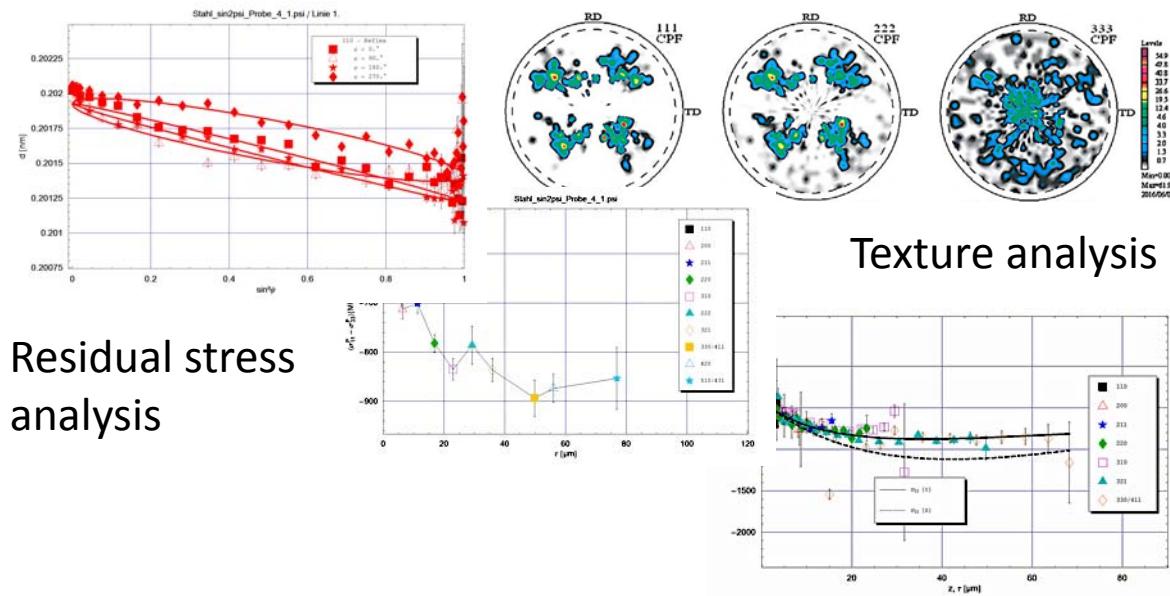
**EDDI-LEDDI**

Ein MATHEMATICA®-Programmsystem zur energiedispersiven Eigenspannungsanalyse

Notebookdatei: EDDI-LEDDI.nb  
Packagedatei: EDDibasicLEDDI.m  
Letzte Änderung: 24. August 2016

Listen mit diffraktionselastischen Konstanten (DEK)  
**EDDI**

Radioaktives Präparat  
Detektor im Labor  
Nützliche Tools ...  
Vorbereitung von ED-Messungen an EDDI  
Auswertung von ED-Beugungsspektren  
Linienlagenkorrekturen  
Darstellung von Ergebnissen  
Quantitative Phasenanalyse zweiphasiger Gefüge  
Texturmessungen  
Ermittlung von Eigenspannungen und Eigenspannungstiefenverteilungen



## Residual stress analysis

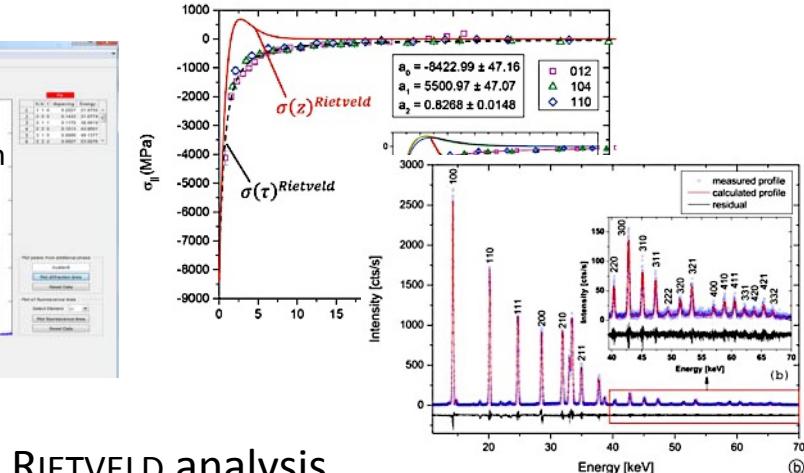
## Texture analysis

**Matlab**

**Script based evaluation program**

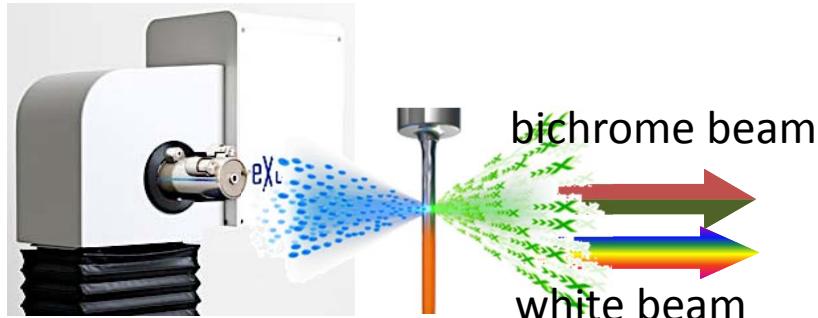
The figure shows the Matlab interface for script-based data evaluation. It includes:

- Editor:** Shows a script named "CreateSample.m" containing MEX code for sample creation.
- Plot Current Measurements:** A plot of Intensity [a.u.] versus Energy [eV] showing multiple sharp peaks.
- Fit results:** A plot of Intensity [a.u.] versus Energy [eV] showing a fit to the experimental data.

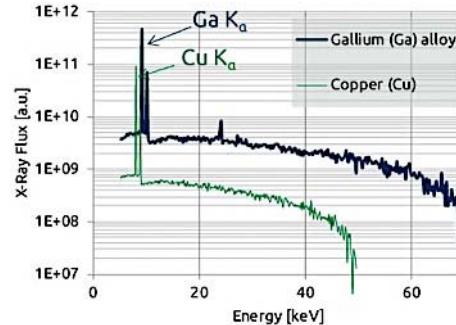


## RIETVELD analysis (stress, microstructure)

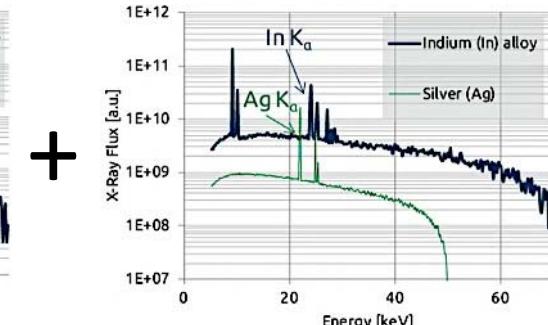
# Synchrotron-like conditions in the lab?



MetalJet X-ray tube

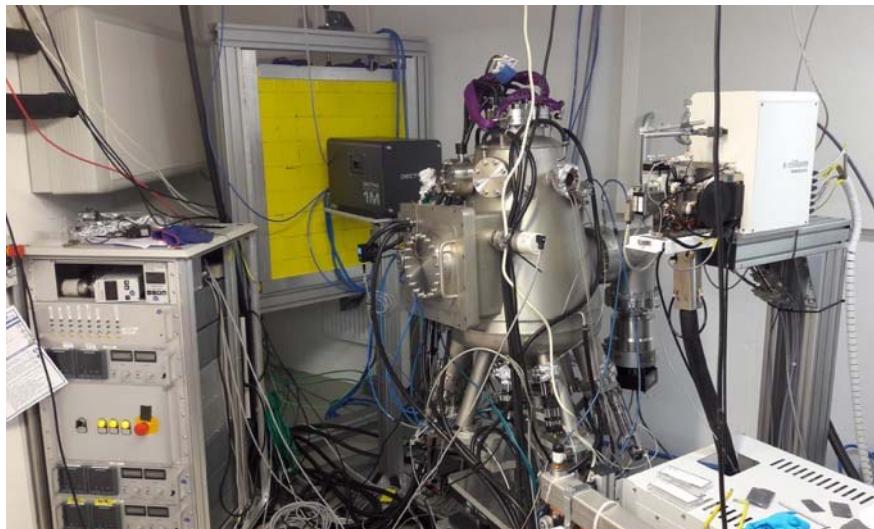


Ga-K $\alpha$ : 9.2 keV



In-K $\alpha$ : 24.2 keV

WCRC/EMIL: 160 kV source

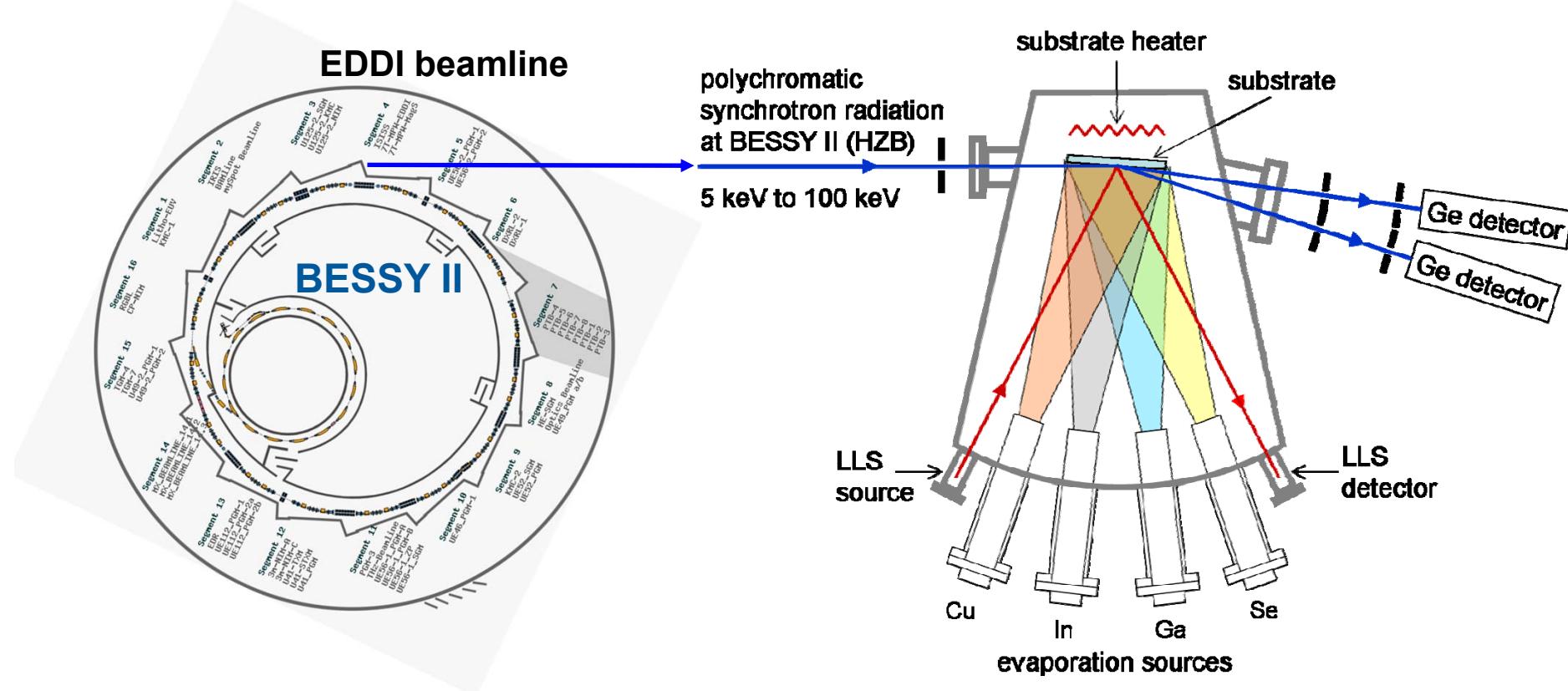


LMC: 70 kV source



# *in situ* EDXRD/XRF during thin film growth

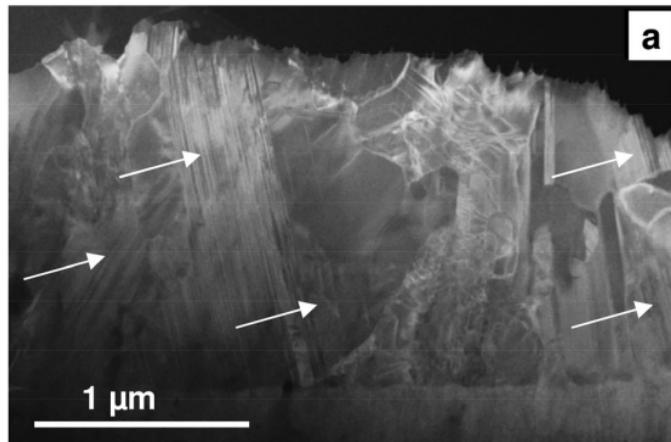
## Energy-dispersive X-ray diffraction and fluorescence (EDXRD/XRF)



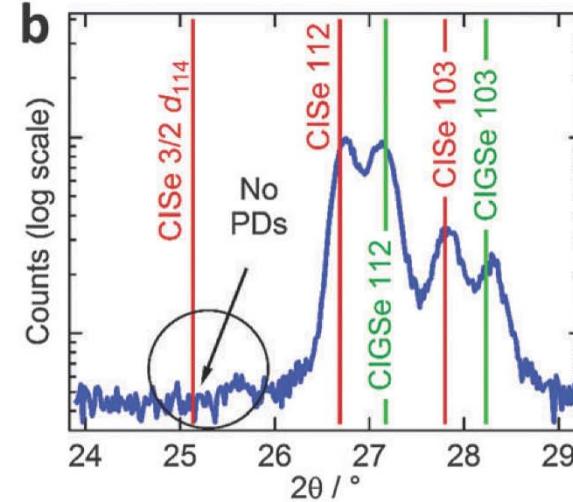
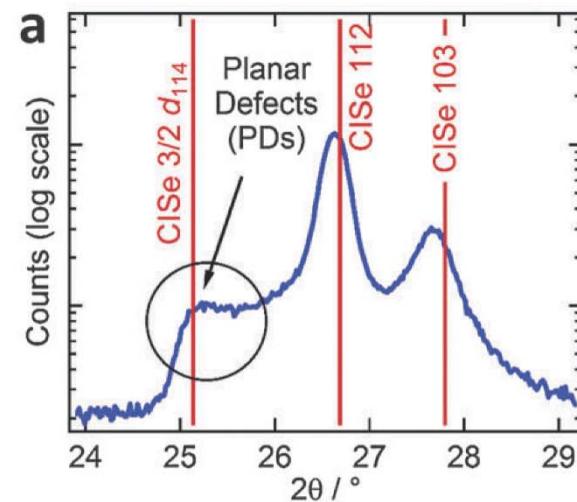
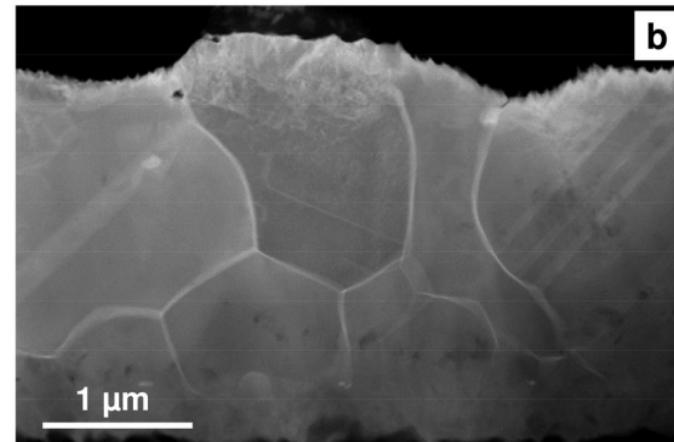
- 3-stage process for Cu(In,Ga)Se<sub>2</sub> thin film growth:
- I) In-Ga-Se co-evaporation
  - II) Cu-Se co-evaporation (Cu-rich stage)
  - III) In-Ga-Se co-evaporation (Cu-poor stage)

# Planar defects in Cu(In,Ga)Se<sub>2</sub> create a diffraction signal

without Cu-rich stage

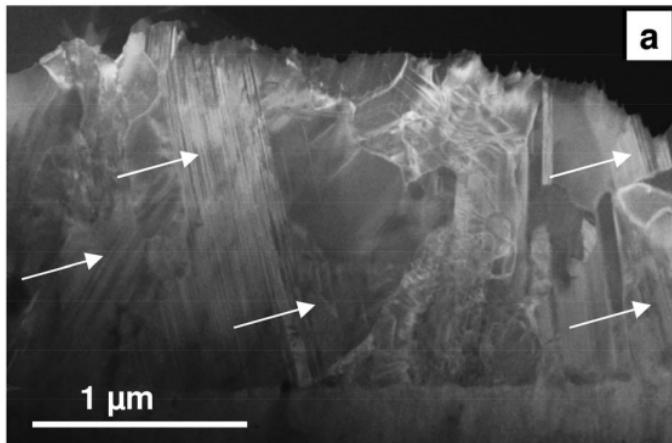


with Cu-rich stage

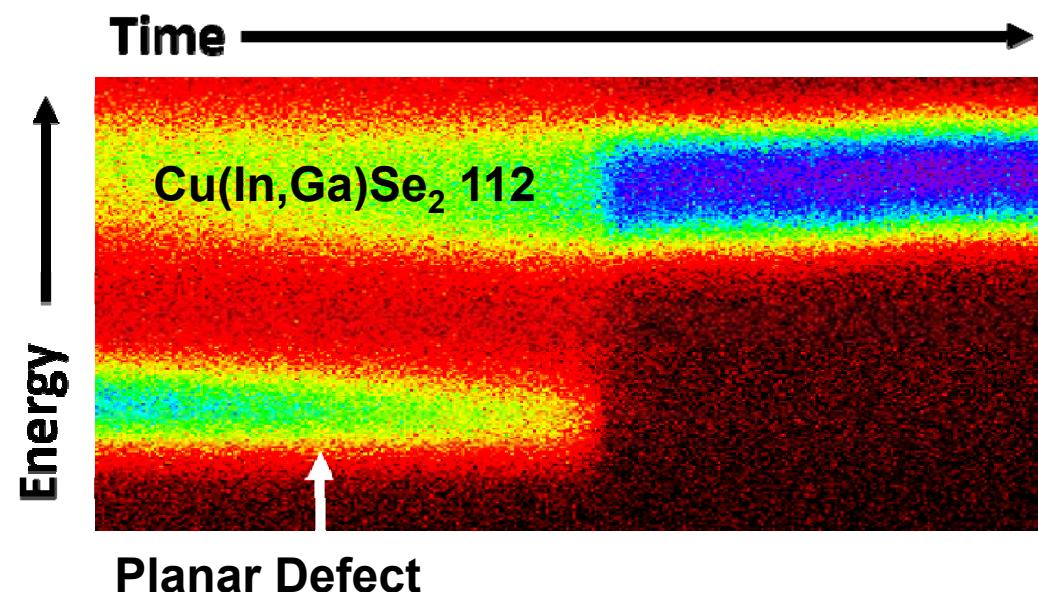
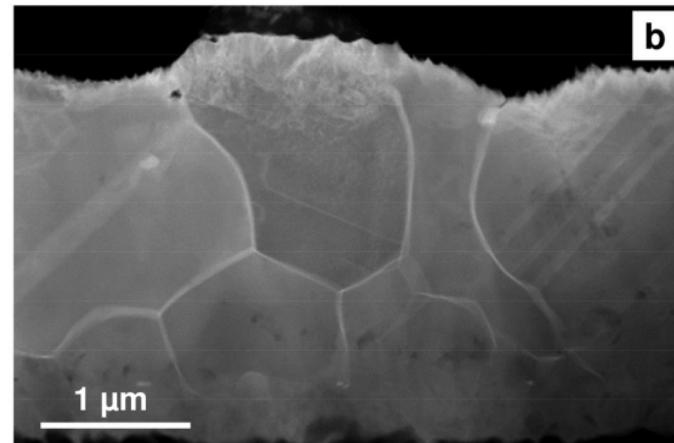


## Monitoring PD annihilation by real-time XRD

without Cu-rich stage

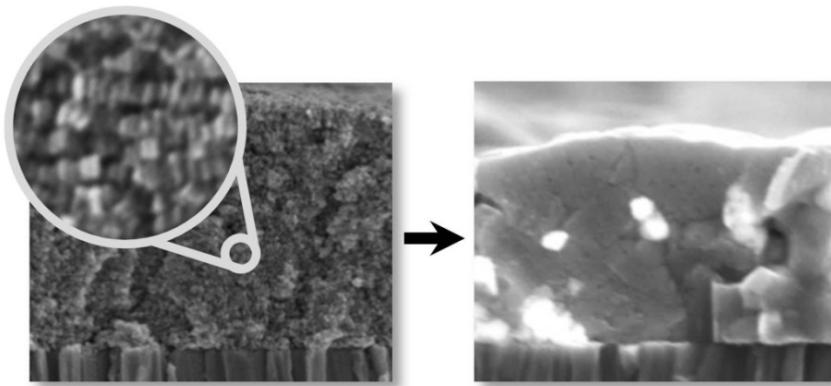


with Cu-rich stage



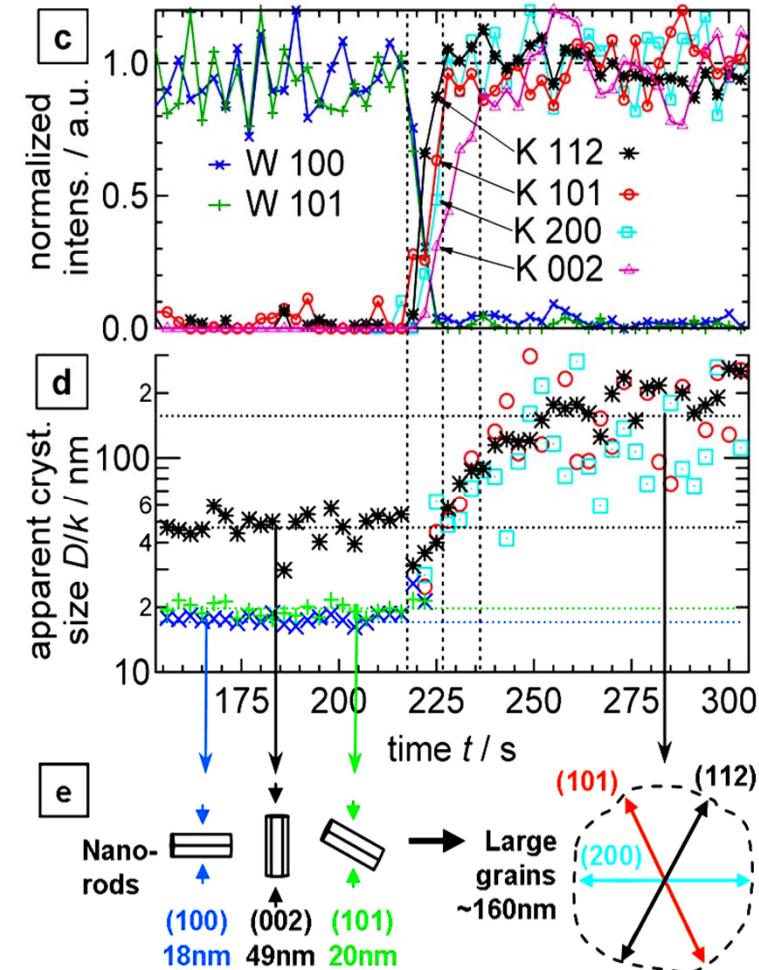
# $\text{Cu}_2\text{ZnSnS}_4$ film formation from wurtzite nanorods

Wurtzite-type  $\rightarrow$  Kesterite-type  
 $\text{Cu}_2\text{ZnSnS}_4$        $\text{Cu}_2\text{ZnSnS}_4$



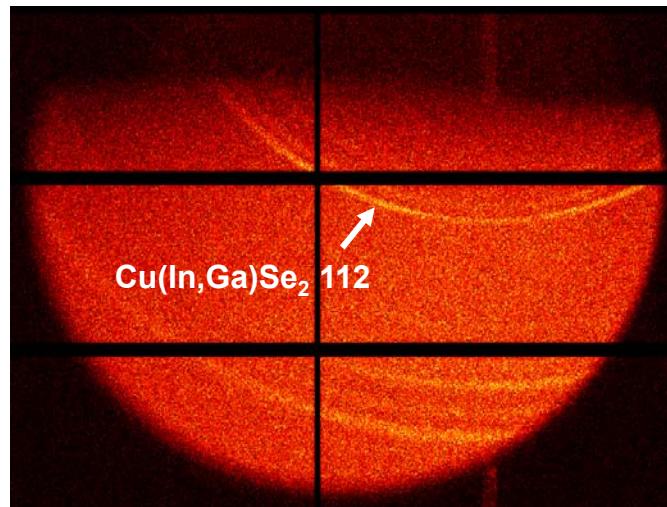
From nanorods to large grains  
within a few seconds!

- correlation of phase formation and domain growth
- phase-transition-driven grain growth

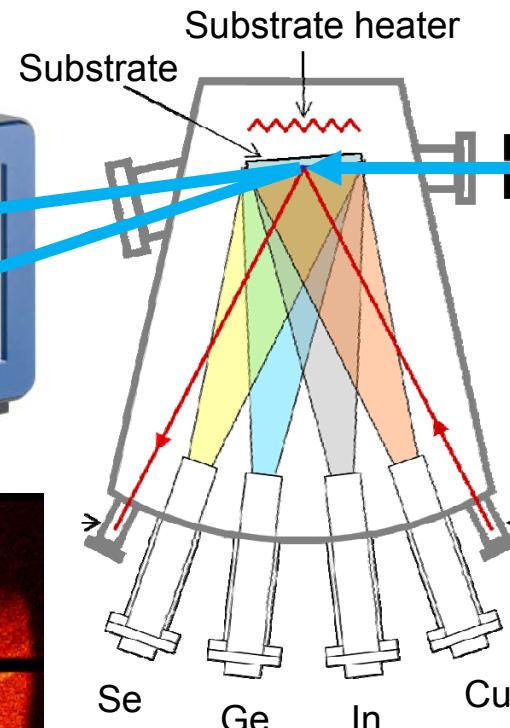


# new *in situ* X-ray laboratory

## 2D photon detection



## thin film growth



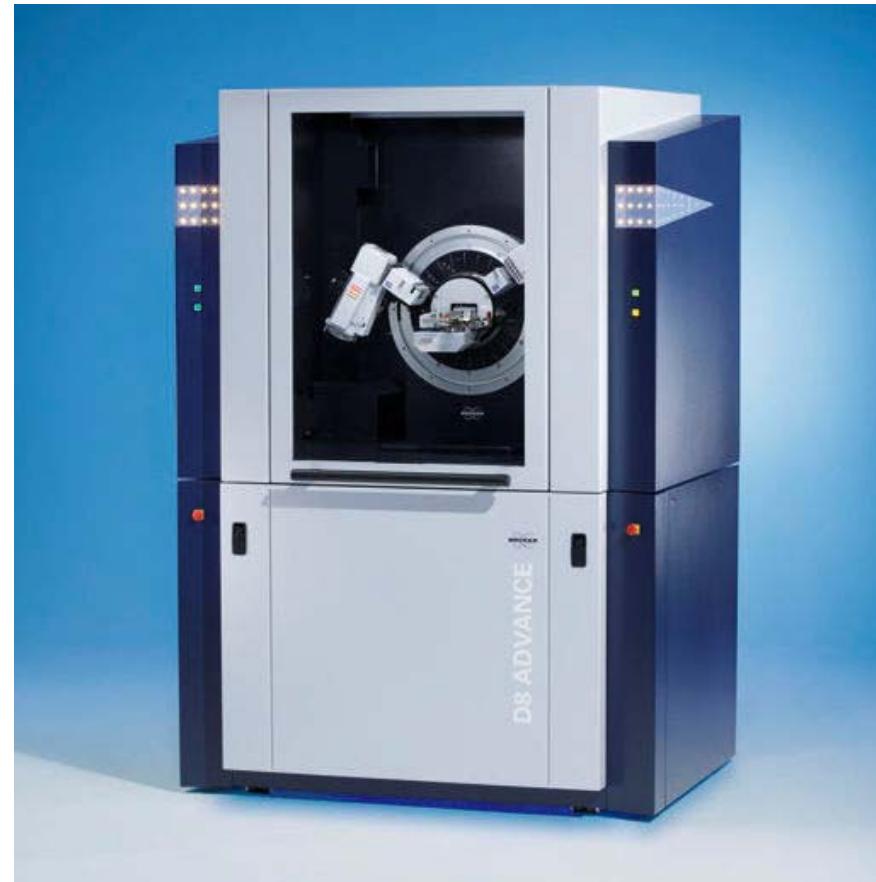
## high-flux X-ray metal-jet source



first in-situ measurement  
during Cu(In,Ga)Se<sub>2</sub> film growth  
with metal-jet X-ray source

## Latest news ...

New X-ray diffractometer for powder diffraction and GIXRD was ordered for WCRC!



course on X-ray diffraction (for PhD students) will be organized next year  
(June 2018) @LMC



## User access – fast and easy!

Scientists from all HZB divisions as well as external users have access to the X-Ray CoreLab.

1st step → each potential user has to register online  
the user has to declare to follow the lab rules

2nd step → booking an instrument of the CoreLab via the online calendar system  
user should give a short description of the planned experiment  
and the samples he/she wants to study

3rd step → check by the scientific lab manager to make sure, that the user has chosen the suitable instrument for his/her problem

4th step → scientific lab manager confirms the booking and the user gets access to the X-Ray CoreLab

LMC → after introduction by the lab manager the user can do the experiments

WCRC → user experiments supported by instrument experts



**Welcome to the X-Ray CoreLab!**

**Thank you for your attention!**